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# Electrical Characterization of Diamond/Boron Doped Diamond Nanostructures for Use in Harsh Environment Applications

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**Abstract.** The polycrystalline boron doped diamond (BDD) shows stable electrical properties and high tolerance for harsh environments (e.g. high temperature or aggressive chemical compounds) comparing to other materials used in semiconductor devices. In this study authors have designed electronic devices fabricated from non-intentionally (NiD) films and highly boron doped diamond structures. Presented semiconductor devices consist of highly boron doped structures grown on NiD diamond films. Fabricated structures were analyzed by electrical measurements for use in harsh environment applications. Moreover, the boron-doping level and influence of oxygen content on chemical composition of diamond films were particularly investigated. Microwave Plasma Enhanced Chemical Vapour Deposition (MW PE CVD) has been used for thin diamond films growth. Non-intentionally doped diamond (0 ppm [B]/[C]) films have been deposited on the Si/SiO<sub>2</sub> wafers with different content of carbon, boron and oxygen in the gas phase. Then, the shape of the highly doped diamond structures were obtained by pyrolysis of SiO<sub>2</sub> on NiD film and standard lithography process. The highly doped structures were obtained for different growth time and [B]/[C] ratio (4000 - 10000 ppm). The narrowest distance between two highly doped structures was 5 μm. The standard Ti/Au ohmic contacts were deposited using physical vapour deposition for electrical characterization of NiD/BDD devices. The influence of diffusion boron from highly doped diamond into non-doped/low-doped diamond film was investigated. Surface morphology of designed structures was analyzed by Scanning Electron Microscope and optical microscope. The resistivity of the NiD and film was studied using four-point probe measurements also DC studies were done.

## 1. Introduction

Diamond shows unique electrical, physical, chemical and mechanical properties [1–3]. Due to its properties has many applications in electronics [2–7], MEMS [8–11], bio-sensors [12], may also act as an electromechanical sensor [13–15], be used for microfluidics [16] or be applied in neuron electrodes [11]. The chemical inertness of the polycrystalline boron doped diamond (BDD) combined with its high thermal conductivity, mechanical strength and electrical conductivity, makes it an attractive material for harsh environment applications.



The characterization of the electrical properties is of importance in order to be able to control and utilize diamond as an electronic material. In order to be able to characterize diamond thin films a new electronic devices have been designed. This devices have been fabricated from non-intentionally doped (NiD) diamond thin films and highly boron doped structures.

The NiD films and BDD structures has been grown by Microwave Plasma Enhanced Chemical Vapor Deposition (MW PE CVD). NiD films have been deposited on the Si/SiO<sub>2</sub> wafers. For the production diamond layers on Si/SiO<sub>2</sub> wafers, nucleation is necessary [1,3,17–19]. The most common used nucleation method is mechanical seeding of the substrate in water solution [19,20] or DMSO [21] with the use of ultrasound bath [22,23]. There are also known methods in which a thin film with nuclei is applied on a substrate [3]. In this methods the nanodiamond powder is placed in polyvinyl alcohol (PVA) matrix [24] or photoresist [25]. Based on diamond suspension technique in polymer PVA mixture [26] has been used for nucleation. This method is easy to obtain and in the PVA are elements only appearing during diamond synthesis by microwave plasma (H, C, O).

Highly boron doped structures has been fabricated by growing BDD structures on the NiD layer by method called second grown [27–29]. The second grown was successfully used for diamond growth [16,30]. For obtain shape of highly doped diamond structures (boxes in 4 size with different distance between them) a pyrolysis of SiO<sub>2</sub> on NiD film and standard lithography process has been used. Deposition and etching SiO<sub>2</sub> were exanimated by Smirnov et al [31].

In this paper authors investigated and showed influence of diffusion boron from highly doped diamond into non-doped/low-doped diamond film. The influence of diffusion has been shown by comparison resistivity of non-doped diamond layer measured by using four-point probe and V/I characteristics of non-doped/boron doped diamond structures. Also, morphology of designed structure was analyzed by Scanning Electron Microscope (SEM) and optical microscope.

## 2. Experimental

### 2.1. Growth of structure

Before growth diamond Si/SiO<sub>2</sub> (400nm thick SiO<sub>2</sub> by wet thermal oxidation ) wafers have been rinsed in acetone and 2-isopropanol for 10 minutes in ultrasonic bath. After cleaning substrates have been dried in nitrogen blow. For nucleation PVA mixture with nanodiamonds was spin-coated on SiO<sub>2</sub> surface. The method of obtaining PVA mixture was described elsewhere [26,32]. NiD diamond layers 400nm thick were grown by  $\mu$ PA CVD system (SEKI 5250). Before process movable stage with samples was heated to 700°C. Temperature during deposition where increase by plasma heating up to 1000°C. The mixture of gases CH<sub>4</sub>, H<sub>2</sub> and B<sub>2</sub>H<sub>6</sub> was used to growth diamond. The NiD layer was grown with flow of gases H<sub>2</sub> - 298.5 sccm, CH<sub>4</sub> - 3 sccm, O<sub>2</sub> - 1 sccm . Gas pressure in the chamber were around 50 torr, and plasma were activated by microwaves 2.45 GHz at power 1300 W.

The SiO<sub>2</sub> mask on NiD diamond layer was obtained, for that at first, a layer of SiO<sub>2</sub> and thickness 1.5  $\mu$ m was produced by pyrolytic method. In to the system (PCVD: Oxford Plasmalab System 80 Plus) a mixture of Si<sub>2</sub>H<sub>4</sub> and N<sub>2</sub>O was inserted in the ratio 1:5 with entire flow of 1200 sccm. SiO<sub>2</sub> deposition process was carried out in the temperature of 350°C, pressure of 1.5 torr and microwave power of 80W. The SiO<sub>2</sub> growth rate with those parameters is 169nm/min. . Next, on the NiD layer the photoresist was spin coated and exposed to UV light for 25s (Karl Suss Model MA-6.). In the next step, SiO<sub>2</sub> was etched by buffered oxide etch (BOE) 6:1.

Etched samples go back again to  $\mu$ PACVD system for one hour secondary growth of highly doped diamond structures. The structures were grown with 4000 and 10000 ppm [B]/[C] ratio. After second growth of diamond the Ohmic (Ti/Au) contacts were deposited on diamond structures by thermal evaporator (Temescal BJD 1800).

## 2.2. Analysis of structures

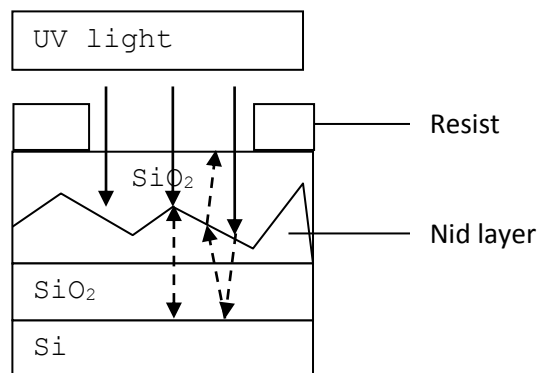
The NiD thin films were analysed by spectroscopic ellipsometry and 4-point probe. For determining properties of continuous diamond layer on Si/SiO<sub>2</sub> substrate was used Jobin-Yvon UVISSEL ellipsometer (HORIBA Jobin-Yvon Inc.). The investigated wavelength range was 260–830 nm. The experiments were performed at room temperature using an angle of incidence fixed at 70°. Ellipsometric fitting was based on a four-phase optical model roughness/diamond/SiO<sub>2</sub>/Si-wafer), Ellipsometry was also used for thickness analysis of SiO<sub>2</sub>.

The morphology of obtained highly doped diamond structures (size of crystals and continuity) was determined by optical microscope (mikroLAB MMT 120 BT) and scanning electron microscope (SEM Hitachi S-3400N). Apart from that, SEM was used to analyze the highly doped structures, to assess smoothness of the edge lines, density of diamond grains occurrence within the areas of unintentional growth. Smoothness of the diamond phase edge was determined from SEM picture after graphic processing in GIMP programme (v. 2.6.11, GIMP development team) and a curve separating diamond and non-diamond phase was determined. As the y-axis was used a line parallel to expected direction of the edge line and contacting in a specific distance the smallest value of the curve. Next, the area under the curve was integrated and divided a segment length on which it was measured. The result was averaged for 10 segments of length 30 μm.

The surface resistivity of non-doped diamond layer was measured by 4-point probe connected to Keithley 2400 and computer with dedicated software. The conductivity between structures were measured by metallic needles on micromanipulators also connected to the Keithley 2400. The 4 point probe were used to analysis resistivity of diamond layer. In case when the film thickness is much smaller than the distance between the probes, the surface resistivity can be calculated from Eq:

$$\rho = \frac{\pi}{\ln(2)} \cdot \frac{V}{I} \cdot h \cdot f_2$$

where  $V$  is voltage measured on the internal probes,  $I$  is current applied to the external probes,  $h$  is the thickness of a layer and  $f_2$  is correction factor [33]. The size of samples (20 mm x 20 mm) and spacing between probes (1.5 mm) gives value on correction factor around 0.97 [34].



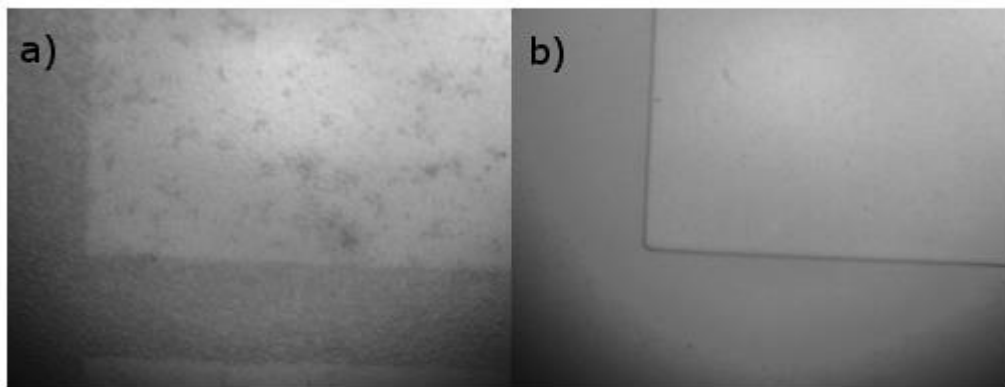
**Figure 1.** Imaging of contact UV light to not perpendicular surface of diamond.

## 3. Results and discussion

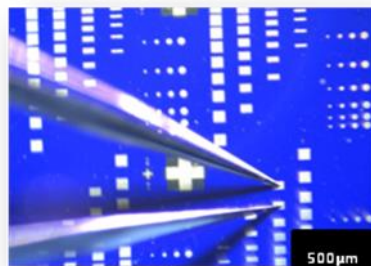
In the experiment the highly doped diamond structures in shape of pads with minimal size of 50 μm were obtained. The NiD diamond layers were used as a substrate for second growth were obtained on Si/SiO<sub>2</sub> wafer with density of diamond nuclei 10<sup>10</sup>/cm<sup>2</sup>. The thickness of diamond film was measured by ellipsometry and was 300 nm which results growth rate of 5 nm/min.



To fabricate the SiO<sub>2</sub> mask for creating diamond structures was necessary to create SiO<sub>2</sub> layer on diamond. For the next step of production structures the 1.5 μm SiO<sub>2</sub> has been grown. For fabrication mask typical process of photolithography has been done. In reason of changed reflection by diamond from the background (figure 1) (compared to bare Si/SiO<sub>2</sub>) to get high quality smooth mask from SiO<sub>2</sub> was necessary to change times exposition to UV light. Additional layers and also not flat diamond layer provides to longer way of the light and different intensity of exposition on some areas on sample. To decrease this negative effects it's possible to increase the time of exposition to UV light. Good effect was obtained when Si/SiO<sub>2</sub>/diamond/ SiO<sub>2</sub>/photoresist sample was 3 times longer exposed to UV light than normally Si/SiO<sub>2</sub> structure. The differences of SiO<sub>2</sub> edged after short can be observed on figure 2.



**Figure 2.** The shape of SiO<sub>2</sub> layer after etching with a) standard time of exposition to UV light (7 s) b) 3 times longer (21 s) exposition to UV light.



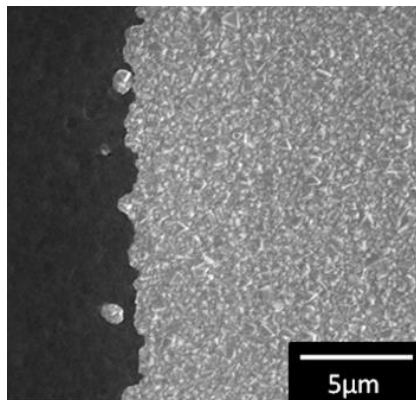
**Figure 3.** Obtained pads with cover Ti/Au picture from optical microscope.

Created and measured highly doped structures were in the shape of the pads in four sizes (50 x 200 μm, 100 x 200 μm, 150 x 200 μm and 200 x 200 μm). The pads of one size were put in the row and distance between them were increasing. The distance between next pads was starting from 5 μm from edge to edge goes by 10, 15, 20, 40, 60, 80, 100, 150 up to 200 μm (figure 3).

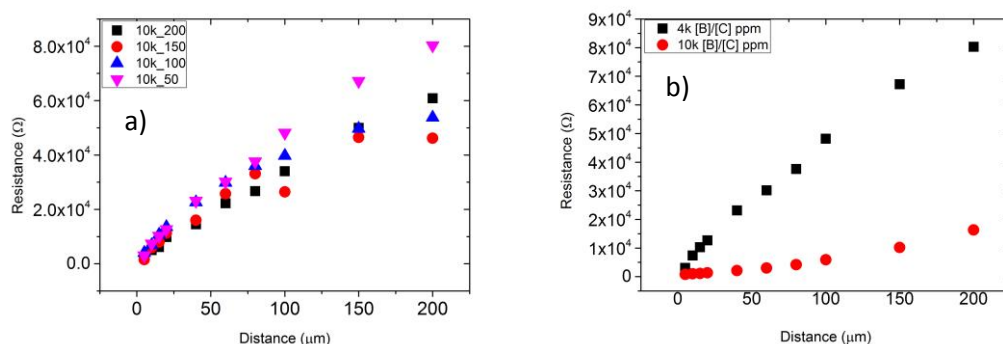
The SEM observations after NiD overgrowth by highly doped diamond structures show high quality edges. The amount of nuclei on the area of unintentional growth was close to 0. The minimal amount of crystals which were grown on black area in figure 4 have minor influence on the isolation between structures. Also the average smoothness of edge line was only 170 nm with maximum inclination of 500nm.

The Ohmic contacts has been used to get better contact to diamond layer in both cases to measured the surface resistivity of NiD layer and to measured contact between highly boron doped structures. The measured value of voltage at the minimum range of current were over the scale of measurement system shows that the material is an isolator.

The measured I/V characteristics have been done between every neighboring pads. The analyzed resistance between all pads of the same size shows almost linear changing resistivity to the distance



**Figure 4.** SEM of observed edge line with one of the crystals on non intentionally phase growth.



**Figure 5.** (a) Comparison of resistance between different size pads (b) Resistivity comparison for two different level of dopant at pads of size 50 x 200  $\mu\text{m}$ .

what was suspected. For small distances value of the resistance of all size pads it's almost that same  $6.5 \times 10^3 \Omega \pm 10^3$  for 10  $\mu\text{m}$ . The analysis the resistivity when distance is 150  $\mu\text{m}$  the differences between pads are bigger and there are  $5 \times 10^4 \Omega \pm 1.5 \times 10^4$ . But it is impossible to recognize the influence of pads size on the resistivity (see figure 5). When compare all results it is possible to observe that using 10000 ppm [B]/[C] level of doping substrate you get better conductivity between the structures than when using 4000 ppm [B]/[C] in gas phase. The 4000 ppm level of doping has always 5 times bigger resistivity than 10000 ppm. It shows that some number of dopant migrate to the non-doped layer provides electrical contact between structures. This effect can be used by controlling the level of dopant migrating into NiD layer or low resistivity layer to build semiconductor devices.

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