

# The Effect of High Molecular Weight Bio-based Diamine Derivative of Dimerized Fatty Acids Obtained from Vegetable Oils on the Structure, Morphology and Selected Properties of Poly(ether-urethane-urea)s

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**Abstract** In this work, the effect of the high molecular weight bio-based diamine on the chemical structure and selected properties of poly(ether-urethane-urea)s has been investigated. The ether-urethane prepolymer was cured using 1,4-butanediol and/or bio-based diamine. Mentioned chain extenders were used separately or in the mixture, and their different molecular weight and chemical structure resulted in obtaining materials with diversified mechanical performance. The presence of specific chemical groups (i.e. urethane and urea groups) was confirmed by FTIR method. For the synthesized poly(ether-urethane-urea)s morphology and fracture mechanism, thermo-mechanical properties and mechanical properties were determined and discussed. Results confirmed that bio-based diamine acts as soft segments, and this is connected with changing of mechanical and thermo-mechanical properties of prepared partially bio-based poly(ether-urethane-urea)s. The increasing content of bio-based diamine resulted in increasing of tensile modulus and decreasing of tensile strength and elongation at break, and this is resulted from chemical structure of bio-based diamine (i.e. presence of aliphatic side chains).

**Keywords** Poly(ether-urethane-urea) · Bio-based diamine · Chemical structure, morphology and properties

## Introduction

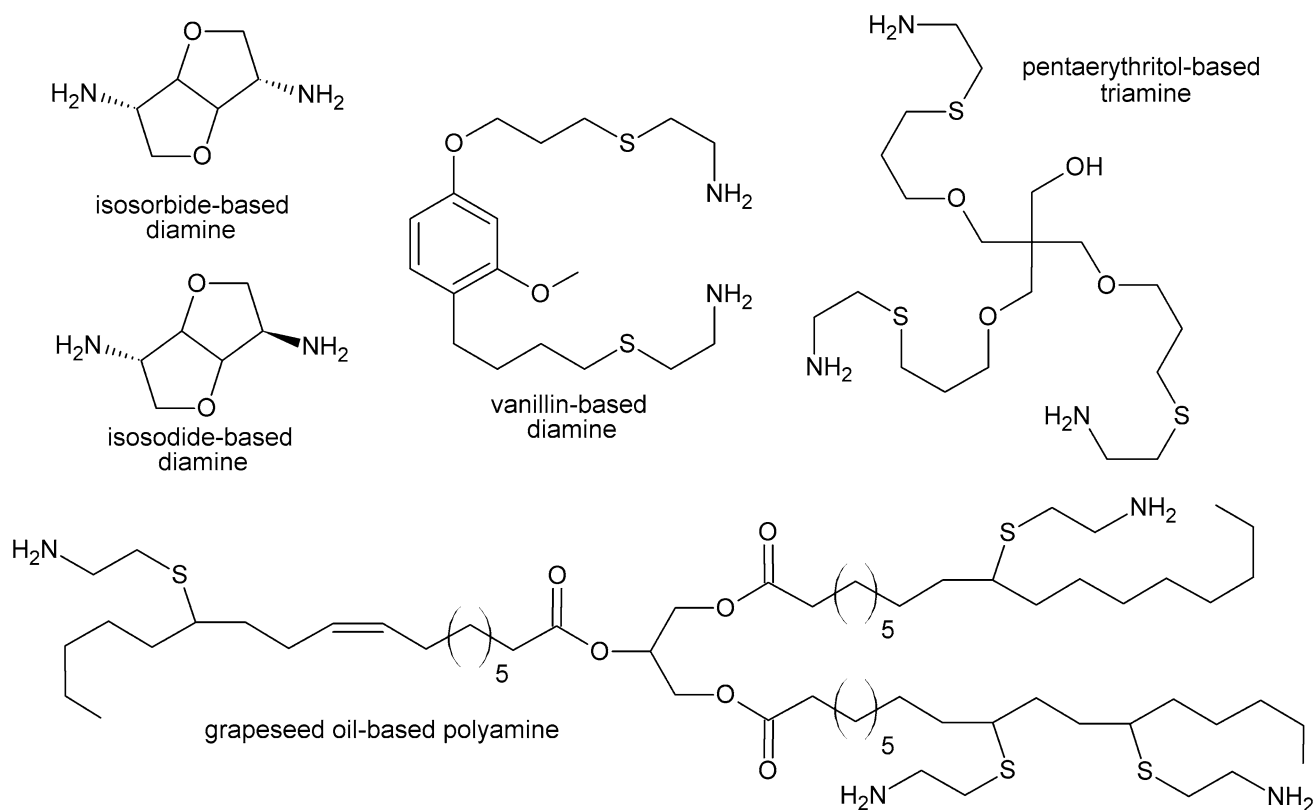
Modern chemistry and technology of polyurethanes is focused on using renewable resources connected with industrial plants, e.g. polyols based on vegetable oils [1–4] glycerol [5] or sorbitol [6], bio-based chain extenders, e.g. corn sugar-based 1,3-propanediol [3, 7, 8], and bio-based diisocyanates synthesized using fatty acids dimers [9] or lysine [10] in the synthesis. In accordance to the literature, the special place is occupied by vegetable oils and their derivatives (e.g. fatty acid dimers) which can be transformed to di- or poly-functional polyols, which are useful in the synthesis of e.g. polyurethanes foams [2, 11, 12], thermoplastic polyurethane elastomers [3] or waterborne polyurethane dispersions [13].

In the area of polyureas, poly(urethane-urea)s and non-isocyanate polyurethanes the main important issue is synthesis of bio-based diamines. Amine-functional compounds can be also derived from natural resources, as e.g. cardanol [14], vanillin [15], pentaerythritol [16], isosorbide [17], isosodide [18], dimers of fatty acids [19, 20] and vegetable oils [21]. Examples of bio-based diamines are presented in Fig. 1. The modification of bio-based starting molecules can be easily realized by thiol-ene coupling i.e. reaction between unsaturated monomers and aminothiols [16, 21]. Mentioned bio-based diamines are especially applied as: a curing agent for epoxy resins [14, 16], cyclic carbonate intermediates (components for the synthesis of non-isocyanate polyurethanes) [15, 19] and as components in the synthesis of polyamides [17] and polyimides [18].

Priamine 1071 produced by Croda (Netherlands) used in this study as a curing agent is bio-based and high purity building block with 100% renewable carbon. Mentioned diamine, in the terms of chemical structure is an amine derivative of dimerized fatty acids. The fatty acids were

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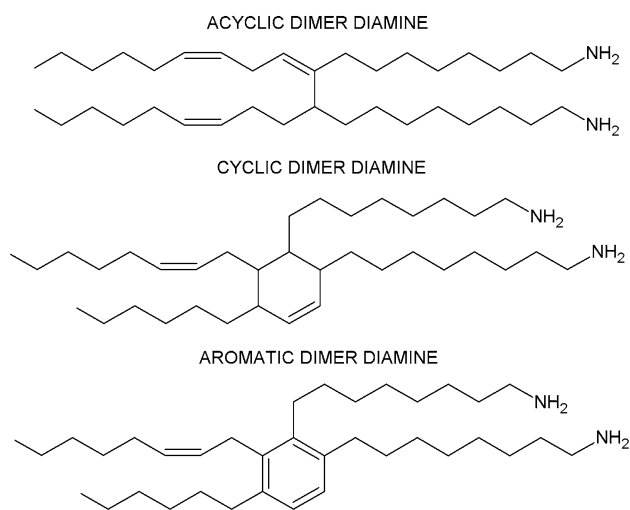


**Fig. 1** Examples of bio-based diamines

obtained from vegetable oil (e.g. soybean oil) by chemical conversion, and Croda produce several different fatty acid dimers from e.g. soybean and sunflower oil. Priamine 1071 was designed as curing agent for epoxy systems. Examples of possible structures (acyclic, cyclic and aromatic) of mentioned bio-based diamine are presented in Fig. 2.

Our earlier work [22] was focused on the poly(ester-urethane-urea)s synthesized using a biobased diamine (PRIAMINE 1071) as a chain extender and chemical structure, mechanical and thermo-mechanical properties of obtained poly(ester-urethane-urea)s were determined. In the chemistry and technology of polymers it is very important to investigate the influence of the selected monomer (chain extender) on the properties of different systems (consisted of different isocyanates and polyols). The interaction between the bio-based diamine moieties and ester- and ether-based polyol main chain is different due to their different chemical structure.

The main aim of this work was to synthesize the poly(urethane-urea)s using a bio-based diamine (obtained from fatty acid dimers) as a single curing agent or in the mixture with 1,4-butanediol. Chemical structure, morphology, thermo-mechanical properties and tensile properties of obtained poly(ether-urethane-urea)s were determined and discussed.



**Fig. 2** Possible chemical structures of amine derivative of dimerized fatty acids

## Experimental

### Synthesis and Description of Poly(ether-urethane-urea)s

The synthesis of poly(ether-urethane-urea)s was realized by two-step method. In the first step ether-urethane prepolymer

was synthesized through a reaction between polytetrahydrofuran (PolyTHF, Overlac, Poland;  $M_w = 2000$  g/mol,  $L_{OH} = 56$  mgKOH/g) and 4,4'-diphenylmethane diisocyanate (MDI, Borsodchem, Hungary;  $M_w = 250,25$  g/mol). The molar ratio of components PolyTHF:MDI was equal 1:2, and the reaction was performed in bulk. Reaction was realized at 80 °C, for 2 h, under vacuum. In the next step, prepared prepolymer was extended with using: 1,4-butanediol (BDO, BRENNTAG, Poland;  $M_w = 90,12$  g/mol), bio-based diamine (PRIAMINE 1071, Croda, Netherlands;  $M_w = 548$  g/mol;  $L_{NH_2} = 205$  mg KOH/g) or mixture of both components. The molar ratio of isocyanate to hydroxyl or/and primary amine groups during chain extending step was 1:1. Formulations of used chain extender agents were presented in Table 1.

The chain extending of prepolymer was realized using Brabender mixer (100 °C, 80 rpm, 6 min), due to high viscosity of reaction mixture, resulted from high reaction rate of isocyanate groups with primary amine groups. After that obtained reaction mass was transferred into metal mould (dimension of mould cavity: 120 mm × 110 mm × 2 mm) and pressed using a hydraulic press at 100 °C, under pressure 5 MPa for 15 min and finally seasoned at 100 °C for 24 h to complete curing process in laboratory oven. Synthesis of ether-urethane prepolymer and poly(urethane-urea)s is shown in Fig. 3.

The content of hard segments (HS), soft segments (SS) and bio-based components was presented in Table 2. In the case of prepared poly(ether-urethane-urea), coded as PEtUM-1.0BA, soft segments are main constituents (83.6 wt%) in the structure and consist of high molecular weight bio-based diamine moieties (PRIAMINE 1071), and high molecular weight linear ether-based polyol (PolyTHF). Hard segments are made up of 4,4'-diphenylmethane diisocyanate (MDI) rest. In the case of materials synthesized using low molecular weight diol i.e. 1,4-butanediol

[as a single chain extender (PEtU) or as a part of chain extenders mixture (PEtUM-0.25BA, PEtUM-0.5BA and PEtUM-0.75BA)] hard segments are composed of 4,4'-diphenylmethane diisocyanate moieties and 1,4-butanediol rest. The difference in molecular weight between 1,4-butanediol and used bio-based diamine results in higher content of hard segments in the case of PEtU in comparison to PEtUM-1.0BA, due to soft nature of high molecular weight bio-based diamine.

## Testing Methods

The presence of chemical groups (e.g. urethane and urea groups) in the chemical structure obtained poly(ether-urethane) and poly(ether-urethane-urea)s was confirmed by Fourier Transform Infrared Spectroscopy (FTIR) using Nicolet 8700 FTIR Spectrophotometer (Thermo Electron Co.), and 64 scans was taken between 500 and 4000  $cm^{-1}$ , with the resolution of 4  $cm^{-1}$ .

Chemical structure of bio-based diamine (Priamine 1071) and selected poly(ether-urethane-urea)s was investigated by Hydrogen Nuclear Magnetic Resonance Spectroscopy ( $^1H$  NMR). Spectra were recorded at the room temperature using a Varian Mercury Vx spectrometer operating at frequency of 400 MHz and applying  $CDCl_3$  as a solvent.

The surface morphology of selected bio-based poly(ether-urethane-urea)s was analyzed using atomic force microscopy (AFM, Nanosurf EasyScan II) in tapping mode. The AFM micrographs analysis was performed in Gwyddion program.

Dynamic Mechanical Thermal Analysis (DMTA) was performed with using DMA Q800 Analyzer (TA Instruments). Measurements were realized in the temperature range from -100 to 100 °C, at an operating frequency of 10 Hz, with a heating rate of 4 °C/min. The samples were 3 mm thick, 10 mm wide and 40 mm long. Storage modulus, loss modulus and damping factor vs. temperature were determined.

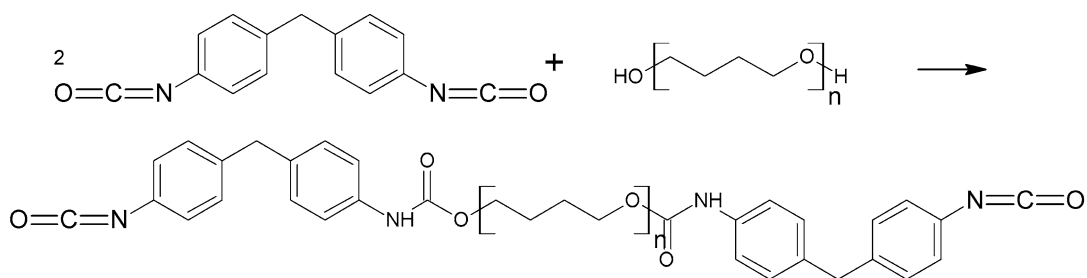
Thermal characterization of poly(urethane-urea) was carried out using a DSC 204 F1 Phoenix Analyzer. At first, the sample was heated at a rate of 20 °C/min from -80 to 200 °C. In the first cycle, the thermal history of the sample was erased. It was then cooled down to -80 °C at a cooling rate of 20 °C/min, and then heated up to 200 °C at a rate of 20 °C/min. The measurements was performed under a nitrogen atmosphere.

Thermogravimetric analysis (TGA) of the selected samples was carried out using NETZSCH TG 209F3 analyzer. The sample has the mass ca. 5 mg and the temperature of measurement was ranged from 40 to 600 °C with a heating rate of 20 °C/min under nitrogen atmosphere.

**Table 1** Formulations of prepared polyurethane and poly(urethane-urea)s

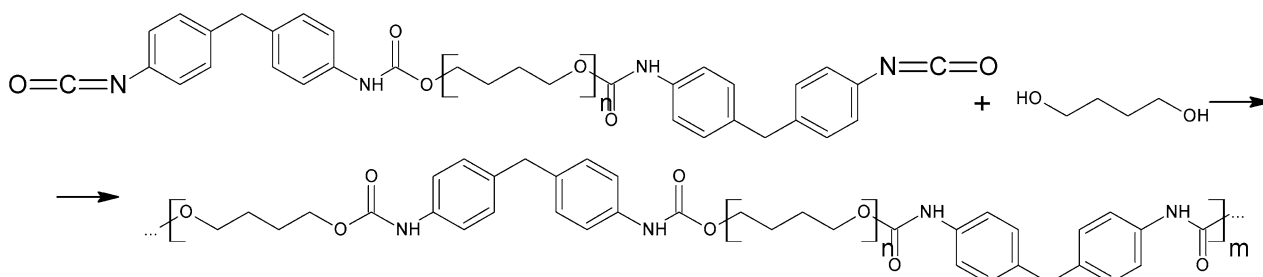
Polymer	Synthesis—step I		Synthesis—step II	
	PolyTHF (mol)	MDI (mol)	BDO (mol)	PRIAMINE (mol)
PEtU Poly(etherurethane)	1	2	1	0
PEtUM-0.25BA Poly(etherurethane-urea)	1	2	0.75	0.25
PEtUM-0.5BA Poly(etherurethane-urea)	1	2	0.5	0.5
PEtUM-0.75BA Poly(etherurethane-urea)	1	2	0.25	0.75
PEtUM-1BA Poly(etherurethane-urea)	1	2	0	1

### STEP I - Synthesis of ether-urethane prepolymer

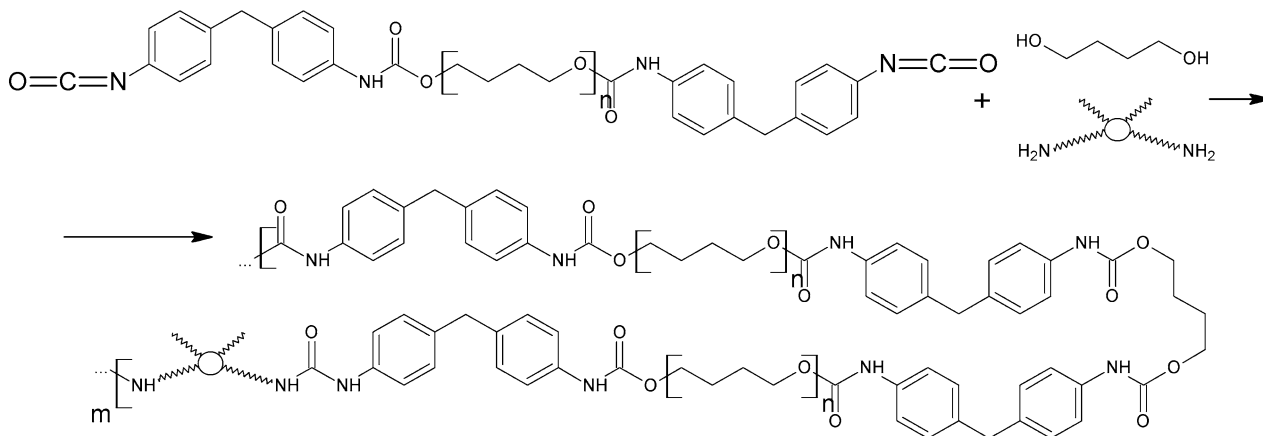


### STEP II - Synthesis of poly(ether-urethane) and poly(ether-urethane-urea)s

#### Example II-1 Synthesis of poly(ether-urethane) - Reference Sample PEtU



#### Example II-2 Synthesis of poly(ether-urethane-urea) - Sample PEtUM-0.5BA



#### Example II-3 Synthesis of poly(ether-urethane-urea) - Sample PEtUM-1.0BA

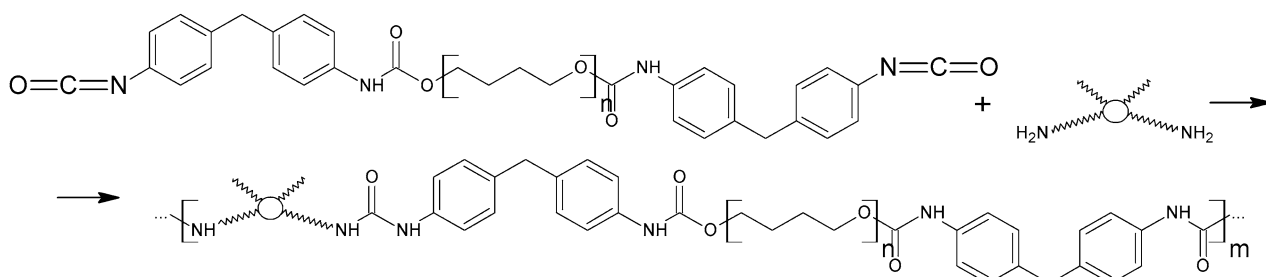


Fig. 3 Synthesis of poly(ether-urethane) and poly(ether-urethane-urea)s

**Table 2** Hard segments, soft segments and bio-based content in the prepared polyurethane and poly(urethane-urea)s

Polymer	Hard segments content	Soft segments content	Bio-based content
	HS (wt%)	SS (wt%)	BIO (wt%)
PEtU	22.8	77.2	–
Poly(etherurethane)			
PEtUM-0.25BA	21.0	79.0	5.1
Poly(etherurethane-urea)			
PEtUM-0.5BA	19.3	80.7	9.7
Poly(etherurethane-urea)			
PEtUM-0.75BA	17.8	82.2	14.0
Poly(etherurethane-urea)			
PEtUM-1BA	16.4	83.6	18.0
Poly(etherurethane-urea)			

Static tensile properties (tensile strength and elongation at break) were measured with using Universal Testing Machine Zwick/Roell Z020 with cross-head speed equal 100 mm/min. Static tensile tests were realized according to ISO 527-2. The test samples were prepared in a standard dumbbell shape (with the width around 6 mm and thickness around 2 mm). Recorded results are averages of five independent measurements.

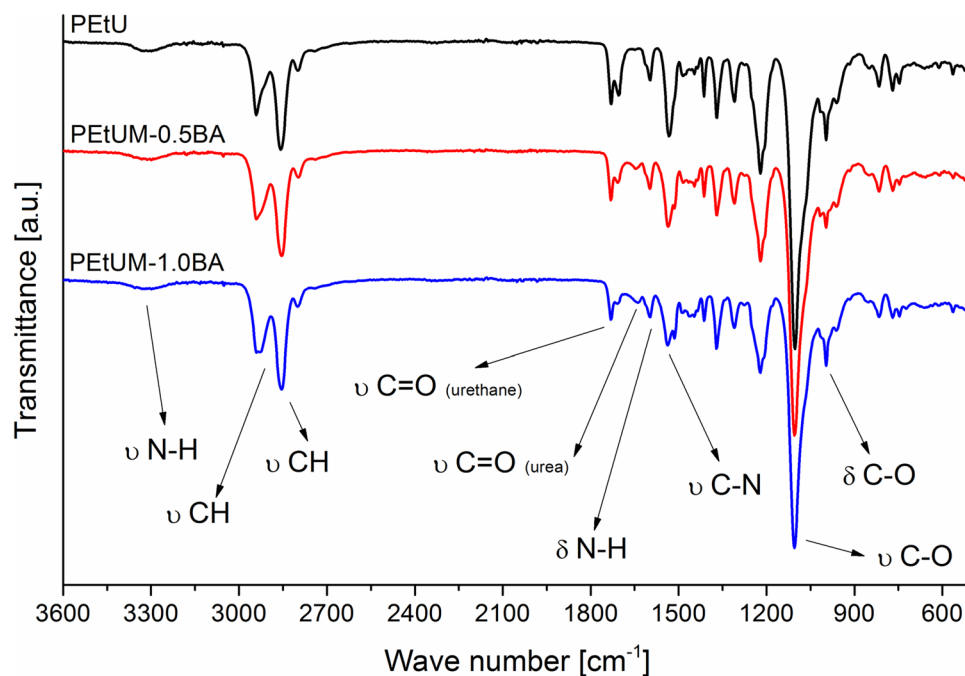
Fracture surfaces after static tensile test were analyzed using desktop scanning electron microscope Phenom G2 PRO (Phenom-World) with accelerating voltage of 5 kV.

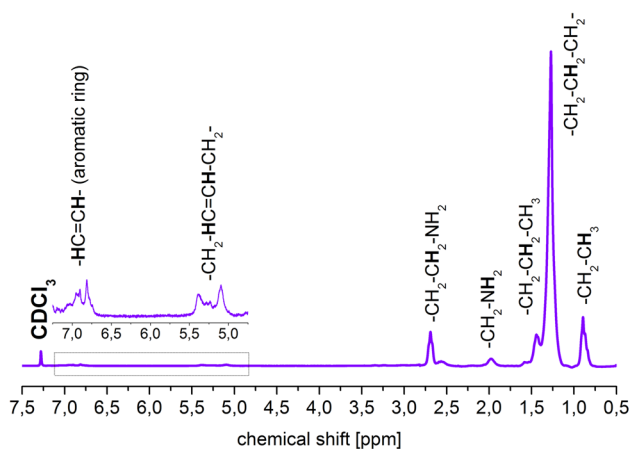
## Results and Discussion

Analysis of FTIR spectra (Fig. 4) confirm presence of typical chemical groups in the chemical structure of obtained poly(ether-urethane) and poly(ether-urethane-urea)s. The most important bands related to characteristic vibrations of chemical groups are: stretching vibrations of carbonyl group in urethane groups ( $\nu$  C=O at  $1725\text{ cm}^{-1}$ ), stretching vibrations of carbonyl group in urea group ( $\nu$  C=O at  $1625\text{ cm}^{-1}$ ), stretching vibrations of C–N in amide group [ $\nu$  C(O)NH at  $1525\text{--}1535\text{ cm}^{-1}$ ], stretching ( $\nu$  C–O at  $1165$  and  $1138\text{ cm}^{-1}$ ) and bending ( $\delta$  C–O at  $1078\text{ cm}^{-1}$ ) vibrations of C–O–C in urethane bonds and ether bonds (introduced by ether-based polyol). Higher amount of fatty acid dimer diamine in the mixture of chain extenders results in increase of peaks intensity for stretching vibration of carbonyl group in urea bond and stretching vibrations (ca.  $2950$ ,  $2920$  and  $2870\text{ cm}^{-1}$ ) of methyl and methylene groups. As can be seen in Fig. 2 the bio-based diamine consist of aliphatic main chain and aliphatic side chains, so increasing of C–H stretching vibrations intensity is obvious.

The chemical structure of bio-based diamine PRIAMINE 1071 (Fig. 5) and the two selected poly(ether-urethane-urea)s, i.e. PEtUU-0.5BA and PEtUU-1BA, was investigated using  $^1\text{H}$  NMR spectroscopy (Fig. 6.).

**Fig. 4** FTIR spectra of prepared poly(ether-urethane) PEtU and poly(ether-urethane-urea)s i.e. PEtUM-0.5BA and PEtUM-1.0BA





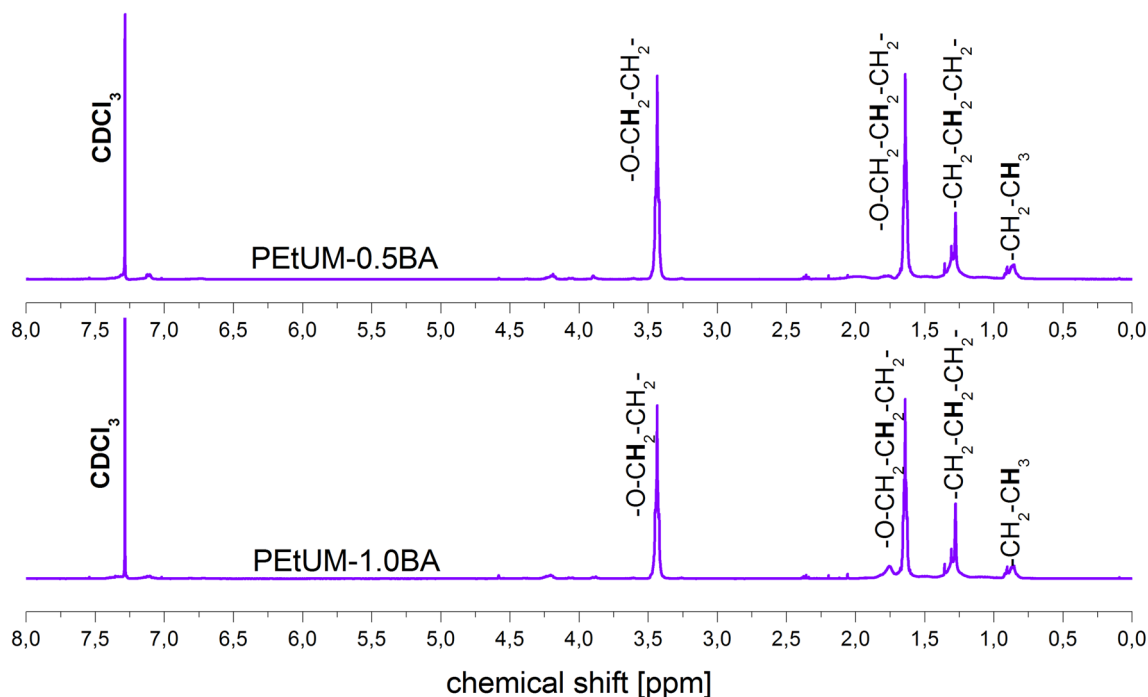
**Fig. 5**  $^1\text{H}$  NMR spectrum of bio-based diamine PRIAMINE 1071

The registered spectra are very similar, and the slight difference in the intensity of signals related to the methylene units ( $-\text{O}-\text{CH}_2-\text{CH}_2-\text{CH}_2-\text{CH}_2-\text{O}-$  and  $-\text{O}-\text{CH}_2-\text{CH}_2-\text{CH}_2-\text{CH}_2-\text{O}-$ ) presented in PolyTHF and methyl ( $\text{H}_3\text{C}-\text{CH}_2-\text{CH}_2-$ ) and methylene ( $-\text{CH}_2-\text{CH}_2-\text{CH}_2-$ ) units in Priamine 1071 are observed. It is resulted from the different amount of both components, which were used for the synthesis of both analyzed materials.

The AFM analysis was used to observe the phase-separated microstructure of prepared poly(ether-urethane-urea)s. Figure 7 shows the AFM (height and three-dimensional)

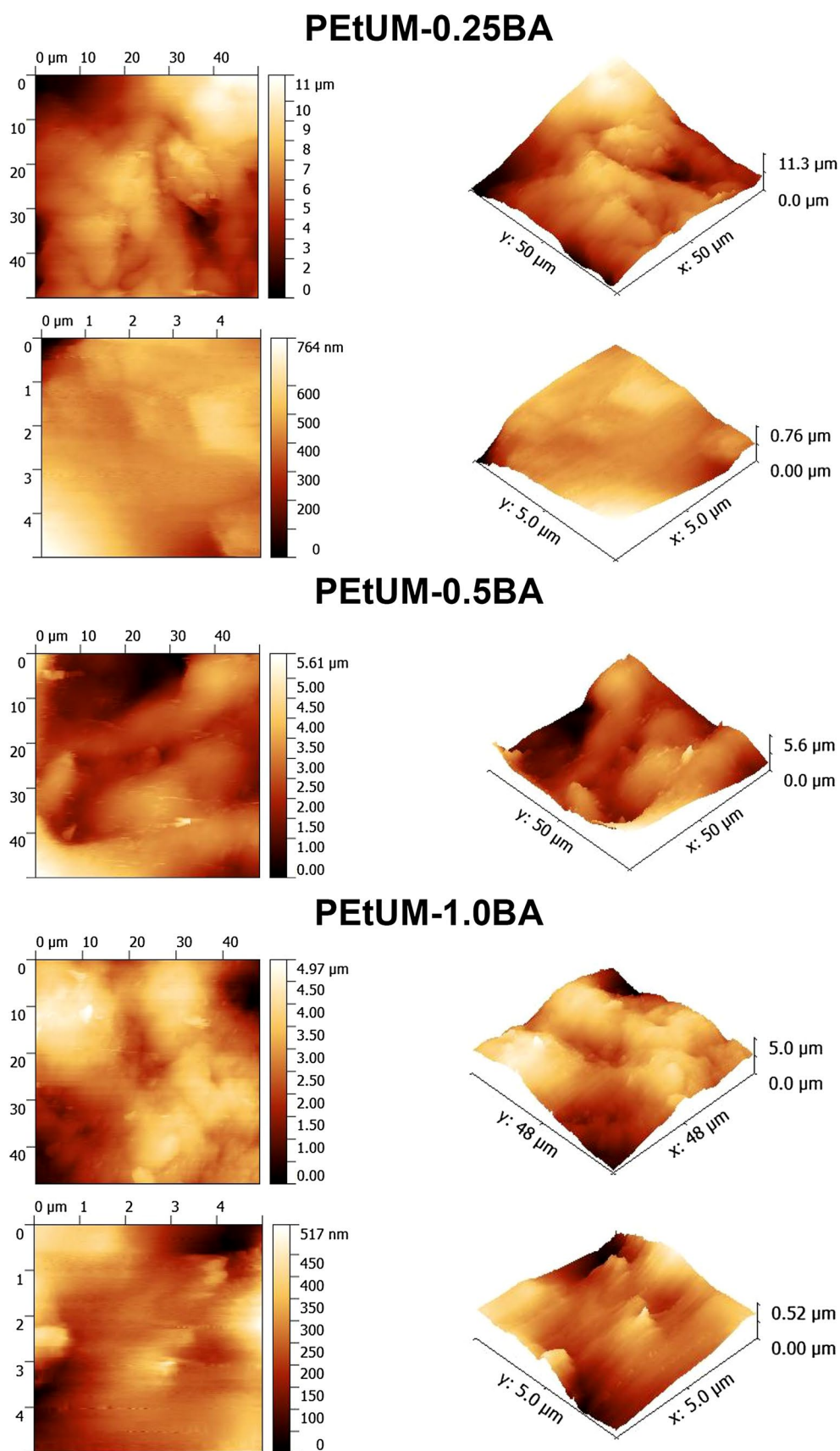
images obtained for PEtUM-0.25BA, PEtUM-0.5BA and PEtUM-1.0BA. It can be seen that prepared poly(urethane-urea)s present a phase separated microstructure, where the light regions are connected with soft segments and the dark corresponded to hard segments. The assignment of soft and hard segments to the light and dark regions in the AFM analysis was also presented by Saralegi et al. [3]. The height of soft regions decreases with increasing content of bio-based diamine in the mixture of chain extenders. Moreover, the surface morphology is more separated for PEtUM-1.0BA than in the case of PEtUM-0.25BA. This can be resulted from higher-content of soft segments based on bio-based diamine, which has higher molecular weight than 1,4-butanediol. The AFM results confirm that presence of bio-based diamine in the structure of poly(urethane-urea)s affecting the soft domain morphology, what can be resulted from different chemical nature of ether-based polyol and aliphatic-based diamine.

Results from dynamic mechanical analysis of poly(ether-urethane-urea)s are shown in Figs. 8, 9, 10. For the all investigated samples, the storage modulus starts decrease around  $-70^\circ\text{C}$ . The decrease in storage modulus and the maximum of damping factor at low temperatures are related to the glass transition temperature of soft segments ( $T_{g\text{SS}}$ ) observed in the region from  $-55.8^\circ\text{C}$  (PEtUM-0.25BA) to  $-48.0^\circ\text{C}$  (PEtUM-1.0BA). Above glass transition of soft segments, storage modulus still decreasing to the melting temperature of soft segments ( $T_{m\text{SS}}$ ) observed in the region

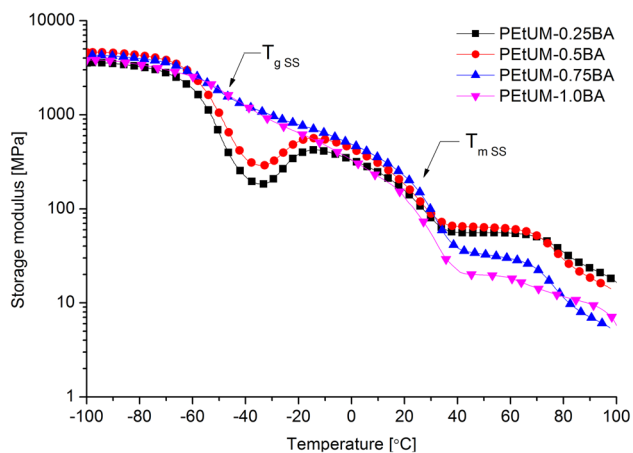


**Fig. 6**  $^1\text{H}$  NMR spectra of selected poly(ether-urethane-urea)s i.e. PEtUM-0.5BA and PEtUM-1.0BA

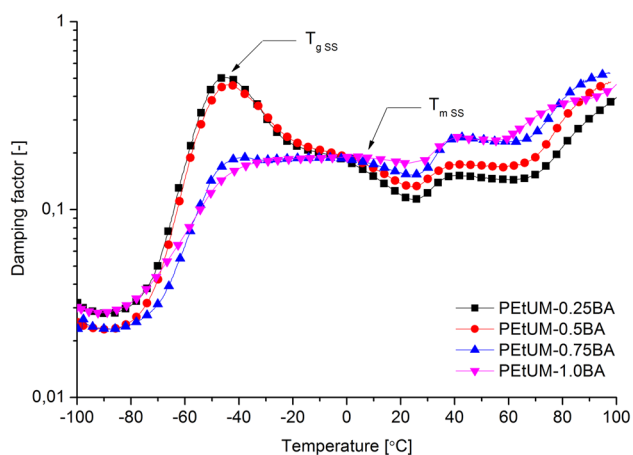




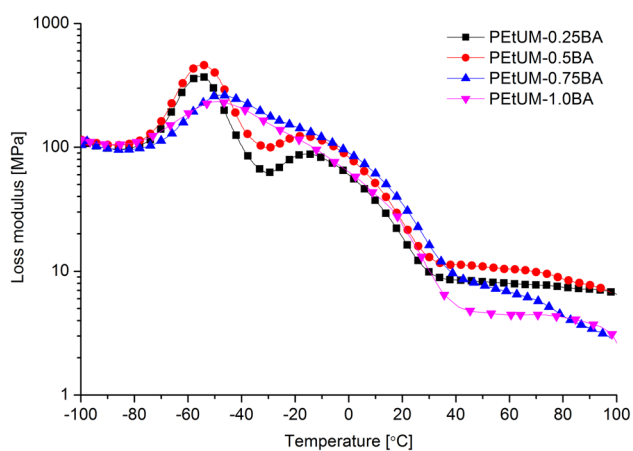
**Fig. 7** Height (*left*) and three-dimensional (*right*) AFM images of selected bio-based poly(ether-urethane-urea)s



**Fig. 8** Storage modulus vs. temperature for obtained poly(ether-urethane-urea)s



**Fig. 9** Loss modulus versus temperature for obtained poly(ether-urethane-urea)s



**Fig. 10** Damping factor versus temperature for obtained poly(ether-urethane-urea)s

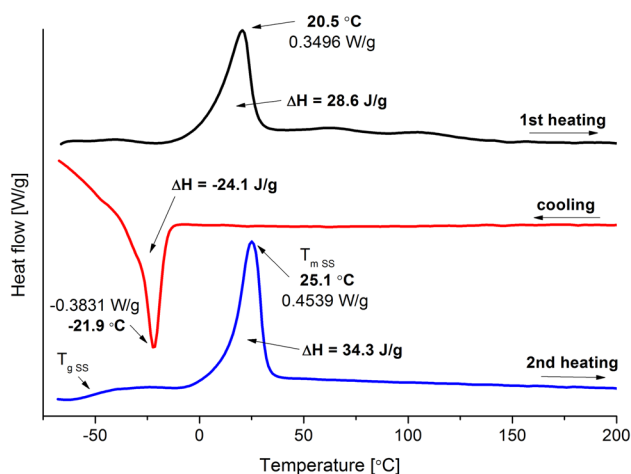
from  $-7.2^{\circ}\text{C}$  (PEtU) to  $5.4^{\circ}\text{C}$  (PEtUM-1.0BA). At this point the rate of decreasing is changing again. In the case of polyurethanes with the high content of soft segments  $T_{m,ss}$  is often observed and similar behavior was reported by other authors. The transition is connected with melting of crystalline ether-based soft segments, which is also observed in the case of poly(ether-urethane)s synthesized using high-molecular weight polyols [3, 23, 24]. The similar phenomena was also observed in our earlier work [22] connected with the synthesis of poly(ester-urethane-urea)s using a biobased diamine (PRIAMINE 1071), where the polyol [ $\alpha,\omega$ -dihydroxy(ethylene-butylene adipate, POLIOS 55/20)] with the high average molecular weight (2000 g/mol) was used. The polyols in previous (POLIOS 55/20) and current (PolyTHF) work have the same average molecular weight but are different in the terms of chemical structure (i.e. ester- and ether-based polyol, respectively).

The improvement in storage modulus in the low temperatures (improvement of stiffness at low temperatures), especially visible in the case of PEtUM-0.75BA and PEtUM-1.0BA, is resulted from soft segment crystallinity. The crystalline regions of the soft domains act as reinforcing factor and result in mechanical integrity of the prepared poly(ether-urethane-urea)s. Similar behavior was observed by Korley et al. [24] in the case of semicrystalline segmented polyurethanes based on poly(ethylene oxide)-poly(propylene oxide)-poly(ethylene oxide) macrodiol, 1,6-hexamethylene diisocyanate and 1,4-butanediol as a chain extender. The increasing of the storage modulus after the glass transition temperature can be also connected with the cold crystallization of the soft segments and the similar behavior was reported in the literature in the case of polyurethanes [25, 26].

Figure 11 shows DSC thermogram obtained for the poly(ether-urethane-urea) synthesized using only bio-based diamine as a curing agent (PEtUM-1.0BA). From the second heating curve two thermal transitions are visible, i.e. glass transition temperature of soft segments ( $T_{g,ss}$ ) at  $-50.9^{\circ}\text{C}$  and melting transition of soft segments ( $T_{m,ss}$ ) at  $25.1^{\circ}\text{C}$ . The board peak of melting transition is related to high content of soft phase (83.6 wt%). The similar values of melting transition temperature and melting enthalpy were observed by Brzeska et al. [27], whose synthesized poly(ether-urethane) using poly(tetramethylene glycol) (2000 g/mol) and 4,4'-diphenylmethane diisocyanates, and 1,4-butanediol as a chain extender, with the molar ratio PTMG:MDI:BDO equal 1:2:1.

The thermal stability of prepared poly(ether-urethane-urea)s was studied using TGA. The results for two selected materials, i.e. PEtUM-0.5BA and PEtUM-1.0BA, are presented in Fig. 12 and in Table 3. The thermal degradation of polyurethanes depends of its chemical structure, and it is known that alkyl-based polyurethanes





**Fig. 11** DSC thermogram of synthesized poly(ether-urethane-urea) with the highest content of bio-based diamine (PEtUM-1.0BA)

exhibit higher thermal stability than aryl-based ones. The first main step, which occurs between 180 and 300 °C, is connected with thermal decomposition of hard segments due to presence of urethane bonds. The second step is connected with decomposition of soft segments [28]. The presence of urea bonds in the structure of poly(urethane-urea)s can improve thermal stability in comparison to similar polyurethanes [29]. The course of the weight loss curves of tested poly(urethane-urea)s are very similar. The small differences (slight higher thermal stability) are connected with higher content of long aliphatic chain bio-based diamine in the structure

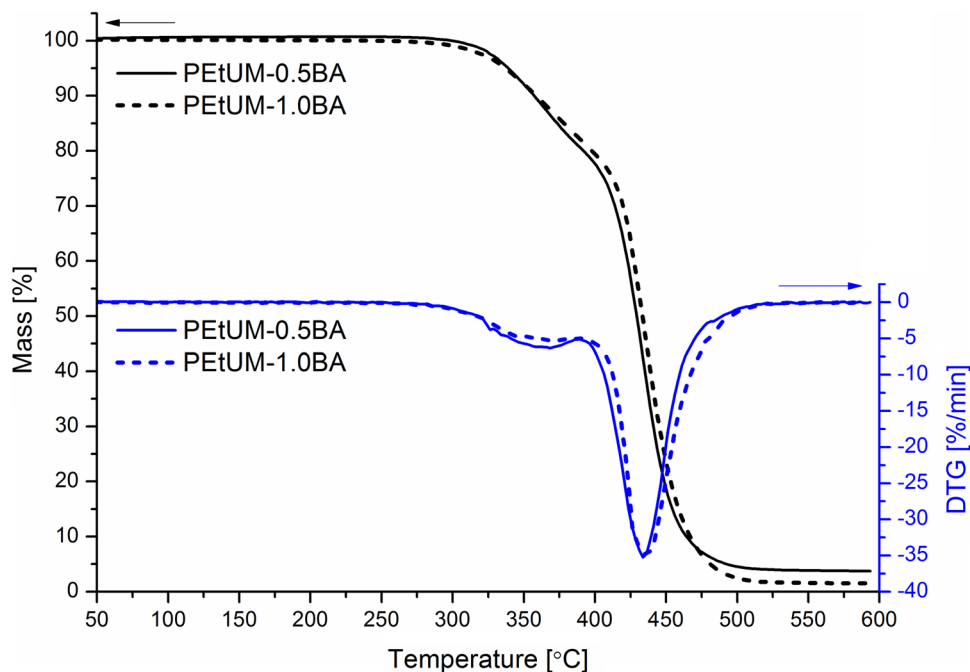
of prepared poly(ether-urethane-urea)s. The corresponding derivative curves consist of the small peak at around 366 °C and the much bigger one at around 435 °C, which are connect with the decomposition of hard and soft segments, respectively. When the degree of phase-separation is insignificant, in the DTG curve only one peak is visible [30].

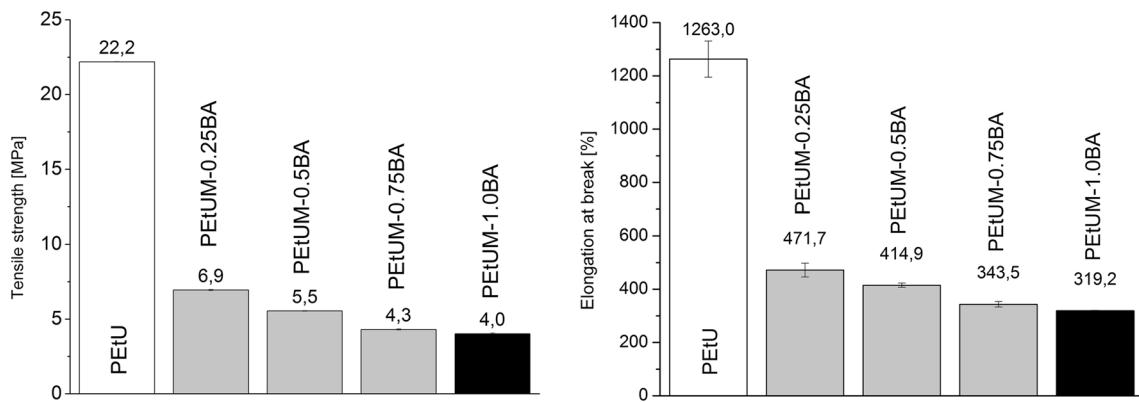
The tensile properties of obtained materials were shown in Figs. 13 and 14. The increasing amount of soft segments in the case of obtained poly(ether-urethane-urea)s is connected with decreasing of tensile strength and elongation at break. It can be also resulted from the structure of applied bio-based diamine. Presence of aliphatic side chains in the structure of bio-based diamine (Fig. 2) can caused reduction of polymer chains movements. Restriction of chains mobility resulted in faster failure of material than in case of poly(ether-urethane) obtained without using diamine (PEtU), which has the highest hard segments content. Similar behavior in the context of tensile strength and elongation at break was observed in our earlier work [22] in the case of poly(ester-urethane-urea)s obtained using 4,4'-diphenylethane diisocyanate,

**Table 3** Thermal stability of selected poly(ether-urethane-urea)s

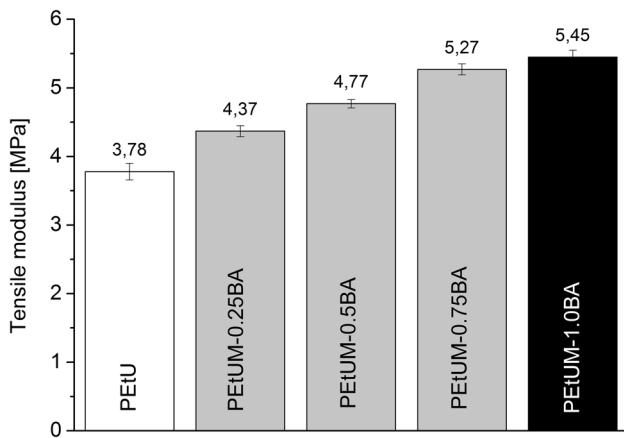
Poly(urethane-urea)	T <sub>5%</sub> (°C)	T <sub>10%</sub> (°C)	T <sub>50%</sub> (°C)	T <sub>max</sub> (°C)
PEtUM-0.25BA	338.6	356.4	427.6	432.1
PEtUM-0.5BA	339.0	357.2	429.3	434.4
PEtUM-0.75BA	338.8	357.4	431.4	434.8
PEtUM-1.0BA	337.0	358.7	433.3	435.7

**Fig. 12** TG and DTG curves of selected poly(ether-urethane-urea)s





**Fig. 13** Tensile strength and elongation at break of prepared poly(ether-urethane) and poly(ether-urethane-urea)s



**Fig. 14** Tensile modulus of prepared poly(ether-urethane) and poly(ether-urethane-urea)s

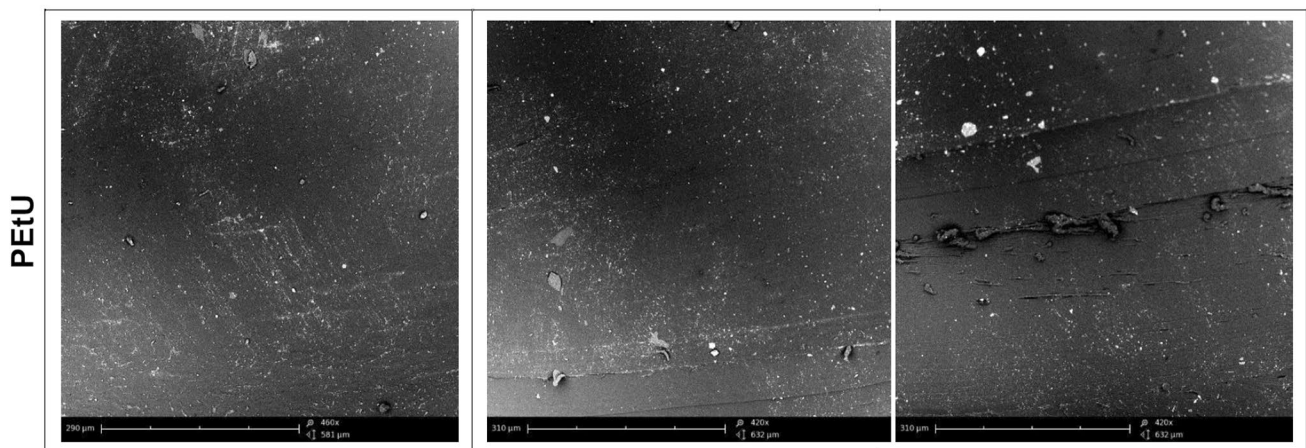
affected by the presence of bio-based diamine moieties in the chemical structure of the prepared polymers. Similar effect of fatty acids dimers on the mechanical behavior (e.g. low tensile strength) was observed by Bueno-Ferrer [31] in the case of biobased thermoplastic polyurethanes obtained from dimer fatty acids. The poly(ether-urethane-urea)s described in this work have higher tensile strength and elongation at break than poly(ester-urethane-urea)s described in our earlier work [22], which is probably resulted with better affinity of bio-based diamine moieties to ether-based soft segments.

The increasing content of soft segments generally is connected with decreasing of tensile strength and increasing of tensile modulus and elongation at break [3]. In the case of prepared poly(ether-urethane-urea)s, the situation is more complicated due to several factors related to applied chain extenders i.e. different molecular weight connected and different chemical structure i.e. different polarity, presence of aliphatic side chains. Presence of long aliphatic side chains in the structure of bio-based

$\alpha,\omega$ -dihydroxy(ethylene-butylene adipate) (POLIOS 55/20), Priamine 1071 and BDO, which suggest that the tensile properties of poly(urea-urethane)s are strongly

**CROSS-SECTION SURFACE**

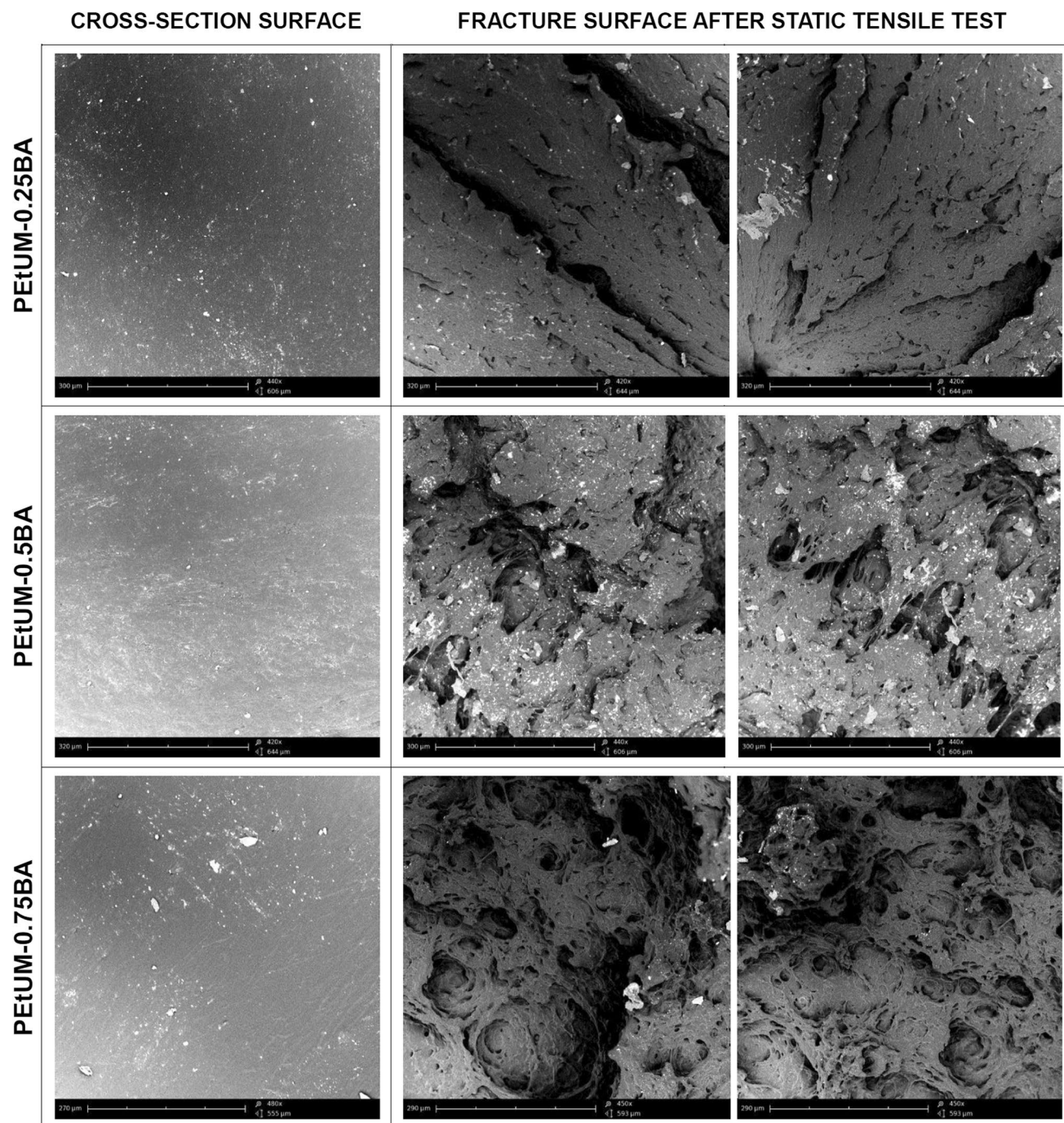
**FRACTURE SURFACE AFTER STATIC TENSILE TEST**



**Fig. 15** Cross-section surface and fracture surface after static tensile test of prepared reference sample, i.e. poly(ether-urethane)

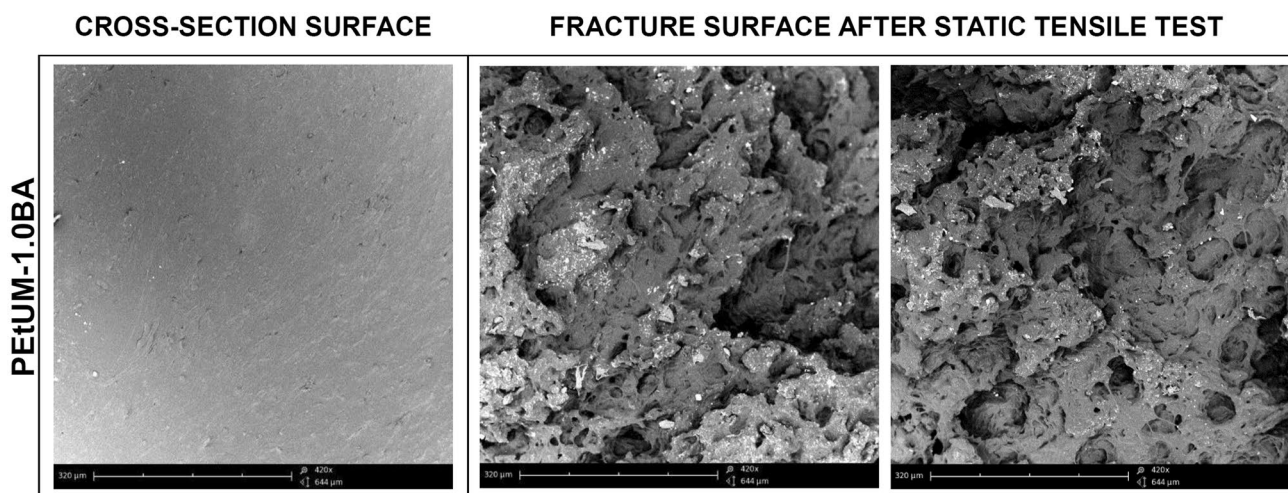
diamine is also connected with increasing of tensile modulus (Fig. 12) observed with increasing of bio-based diamine content. The aliphatic side chains act as steric hindrance during poly(urea-urethane) chains movements and this result in higher stiffness (higher tensile modulus) of prepared bio-based materials.

The morphology of cross section and fracture surface after static tensile test was analyzed for all prepared polymers, and results are showed in Figs. 15, 16, 17. As can be seen the cross-section surfaces for all prepared materials are smooth and voids are absent. The analysis of fracture surface clearly indicate, that using of high molecular weight



**Fig. 16** Cross-section surface and fracture surface after static tensile test of prepared poly(ether-urethane-urea)s synthesized using mixture of curing agents





**Fig. 17** Cross-section surface and fracture surface after static tensile test of prepared poly(ether-urethane-urea)s synthesized only bio-based diamine as a curing agent

bio-based diamine results in increasing of ductile nature of fracture formed during static tensile test. It is resulted from higher soft segments content due to presence of bio-based diamine, composed from long aliphatic main chain and side chains. The fracture surface is more deformed for PETUM-1.0BA in comparison to reference sample PETU.

## Conclusions

Partially bio-based poly(ether-urethane-urea)s were successfully obtained using plant-based monomer. The diamine derivative of dimerized fatty acids (from vegetable oils, e.g. soybean or sunflower oil) was used as a curing agent for ether-urethane prepolymer. The bio-based content in the synthesized poly(urethane-urea)s is in the range from 5.1 wt% (PETUM-0.25BA) to 18.0 wt% (PETUM-1.0BA). Application of different amount of bio-based diamine caused change of soft segment content in the prepared materials, affected tensile properties. Observed behavior, i.e. increasing of tensile modulus connected with decreasing of tensile strength and elongation at break, is connected with the chemical structure of bio-based diamine (long aliphatic main chain with aliphatic side chains), which caused restriction of polymer chains movements. It was confirmed e.g. by results of dynamic mechanical analysis, due to observe increasing the glass transition temperature of soft segments.

Synthesis of poly(urethane-urea)s with using vegetable oil-based monomers (like Priamine 1071 which is an amine derivative of fatty acids dimers) are connected with trends of green chemistry. Suitable selection of

components and their amounts will result in obtain of materials with competitive properties in comparison to petrochemical-based polymers. As was proved in this work, the application of plant-based building block in the synthesis of polymers resulted in obtain materials with diversified properties.

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