



The Influence of Substitution of a Phosphorus-Containing Polyol with the Bio-polyol on the Properties of Bio-based PUR/PIR Foams

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Abstract

In this work, effects of incorporating of a phosphorus-containing polyol into rigid polyurethane/polyisocyanurate foams' formulations developed with use of two different bio-based polyols, derived from crude glycerol or liquefied cellulose were examined. The bio-polyol derived from crude glycerol was synthesized via two-step process from crude glycerol and castor oil, whereas the bio-polyol derived from liquefied cellulose was prepared in lignocellulose biomass liquefaction process. Rigid polyurethane/polyisocyanurate foams were prepared by substitution 25, 50, 75 and 100 wt% of a bio-polyol with the phosphorus-containing polyol. Density, compressive modulus, thermal stability and chemical and cellular structure content of the PUR/PIR foams were studied. Cell morphology was also observed by scanning electron microscopy. Furthermore, thermal stability and flammability were investigated. The results show that the phosphorus-based polyol favors some PUR/PIR foams properties such as fire retardancy, what was expected, but also compressive strength behavior.

Introduction

One of many challenges that polyurethane industry is facing nowadays its dependency on petrochemical products, having two main raw materials, e.g. polyols and isocyanates of petroleum origin. As it was shown in many published works, renewable materials can often substitute their petrochemical analogs with little to none negative influence on final products' properties. With increasing emphasis being put on waste disposal issues and usage of renewable biomass sources, polyurethane-based materials get more attention and a higher number of possible applications [1, 2]. The advantages of these materials include their low cost, easy availability, and renewability, better biodegradability, as

well as attention-drawing properties related to the plant oil structure [3–5]. A wide range of plant oils is used, namely sunflower oil [6], rapeseed [7], palm [8, 9], castor [4, 10, 11] and soybean [12–15] oils. The number of publications on the synthesis of polyurethanes with the use of plant-based polyols is constantly growing. Petrović et al. [13, 16, 17] obtained polyurethanes using various polyols based on vegetable oils and, he has shown that the thermal stability of urethane groups is influenced by the orderliness of hydroxyl groups. Polyurethanes obtained using polyols with primary hydroxyl groups are more stable than those with secondary ones. PUR derived from castor oil have also higher thermal stability compared to polyurethane materials based on soy oil. Among the various waste components used to obtain PUR foams, the raw glycerol, a by-product from the production of biodiesel, generated on a scale of about 1 million tons per year in Europe alone [18, 19] is especially worth mentioning. Due to its low price, it can be considered as an interesting and economically efficient alternative to petrol-based materials, causing various research groups to investigate its use in bio polyol synthesis [20–22].

From bio-based polyols foamed PUR materials were also obtained, but the study of their properties was often limited and included only basic mechanical tests [23, 24]. Hu et al. [12] studied the feasibility of using crude glycerol to liquefy soybean straw for the production of bio-polyols and polyurethane foams. The obtained bio-polyols showed a hydroxyl

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number of 440–540 mg KOH g⁻¹ and the produced foams showed densities of 33–37 kg m⁻³ with compressive strength from 148 to 227 kPa.

However, polyurethanes are characterized by high flammability, which is accompanied by the emission of toxic gases and large amounts of smoke emitted during combustion. The extremely important factor determining the development of the polymer materials industry is the success in efforts to eliminate toxic compounds and reduce flammability [25, 26]. Safety requirements for polymer materials in the scope of reaction to fire and their fire resistance are now increasingly stringent. In order to increase the fire safety and to reduce flammability of polyurethane materials, flame retardants are applied. In concern for the environment and human safety, the use of halogen flame retardants has been abandoned. The challenge for flame retardant systems is to be non-harmful to people and the environment [27, 28]. The most commonly used commercially available flame retardants for polyurethane materials include triphenylphosphate [29], phosphoric trichloride with bisphenol A diphosphate [30], aniline and 9,10-dihydro-9-oxa-10-phosphaphenanthrene-10-oxide [31, 32], ammonium polyphosphate [33, 34] and tris(1-chloro-2-propyl)phosphate [35, 36]. Phosphoric polyols belonging to the group of reactive flame retardants of polyurethane foams are noteworthy in particular. Organophosphorus compounds favorably affect the charring and char formation process rendering them very effective due to the fact that they produce during burning a stable form of poly(meta-phosphoric acid), which forms an insulating layer on the surface of the polymer. This modification, however, does affect the properties of the modified materials. Problems related to effective flame retardation of polymers bring this area to intensive research and searching for innovative solutions.

The presented study evaluates the potential use for the synthesis of rigid polyurethane/polyisocyanurate foams of two bio-polyols, one based on crude glycerol and castor oil and one obtained by use raw glycerol to liquefy cellulose. The bio-polyol in the foams' formulation was then replaced from 0 to 100% by weight with a commercial phosphoric polyol. The chemical structure (FTIR tests) and cell structure (SEM analysis), mechanical (compression tests, dynamic mechanical analysis) and thermal properties (thermogravimetric analysis), photo-oxidative and thermo-oxidative aging and fire behavior of obtained rigid (PUR/PIR) foams were investigated.

Experimental

Materials

The bio-based polyol designated LC (like liquefied cellulose) was synthesized by liquefaction of cellulose with the

mixture of purified crude glycerol. Cellulose in the form of cotton fiber with a water content of 5.4% was obtained from VITA Rzeszow (Poland). Refined crude glycerol was acquired from Bio-Chem Sp. z o.o. (Poland). The water content of the crude glycerol was lower than 0.5 wt% and its density was 1.26 g cm⁻³. The liquefaction process was catalyzed by 95% aqueous solution of sulfuric acid (VI) obtained from Avantor Performance Materials Poland SA. Detailed description of synthesis and properties of bio-polyol LC was presented by Kosmela et al. [37].

The bio-based polyol designated CG (like crude glycerol) was synthesized from crude glycerol and castor oil. Crude glycerol was acquired from Bio-Chem Sp. z o.o. (Poland), while castor oil was acquired from Pro-Lab (Poland). 85% aqueous solution of orthophosphoric acid was used as a catalyst and 50% aqueous solution of sulfuric (VI) acid was used to neutralize reaction products. Both acids were obtained from POCH S.A. (Poland). Particular stages of synthesis were described by Hejna et al. [35].

Rigid PUR/PIR foams were prepared from previously obtained bio-based polyols, noted as LC and CG and commercially available polyol from Clariant company, Exolit OP 560 (E560) a reactive type flame retardant oligomer phosphonate. The properties of the aforementioned polyols are presented in Table 1.

Polymeric methylene diphenyl diisocyanate (pMDI) was supplied by Borsodchem (Hungary). The average functionality of pMDI was ca. 2.8, while the content of free isocyanate groups was 31.5%.

Tin(II) 2-ethylhexanoate [Sn(Oct)₂], Dabco® TMR7—tertiary amine as well as potassium octoate (K15) were used as catalysts. All of those were supplied by Sigma-Aldrich.

Moreover, Tegostab® B 8465, non hydrolysable surfactant obtained from EVONIK Industries AG was chosen. Distilled water and mixture of liquid hydrocarbons Solkane® 365/227 supplied by Solvay Chemicals International SA. were both used as blowing agents.

Preparation of Polyurethane/Polyisocyanurate Foam

Rigid polyurethane/polyisocyanurate foams were obtained from the two-component system, with the NCO/OH ratio

Table 1 Properties of polyols used for the production of rigid polyurethane/polyisocyanurate foams

Polyol	CG	LC	E560
Density (g cm ⁻³)	1.18	2.7	1.20
OH number (mg KOH g ⁻¹)	580	750	450
Viscosity (mPa s)	840	150	500
Water content (%)	0.21	0.63	0.01

of 2:1 using one-step method. Component A was obtained by thorough mixing (rotational speed of 1800 rpm, mixing time 60 s) of the appropriate amounts of Exolit OP 560 and bio-polyol CG or LC, catalysts, surfactants and blowing agents. Phosphorous containing polyol Exolit OP 560 was incorporated as a substitute- in the foam formulation of 25, 50, 75 and 100 parts by weight of a bio-polyol CG or LC. Component B consisted of an isocyanate pMDI. Components A and B were mixed together at the selected mass ratio in at polypropylene cup and the mixture was poured into a mold. Thus obtained foams were annealed in an oven at 60 °C for approx. 24 h, and then stabilized at room temperature for 24 h. Tables 2 and 3 contain the details of foams formulations.

Measurements and Characterization

FT-IR spectrophotometric analysis was performed in order to determine the chemical structure of the obtained polyurethane/polyisocyanurate foams. The analysis was performed at a resolution of 4 cm⁻¹ using a Nicolet 8700 apparatus (Thermo Electron Corporation) equipped with a snap-Gold State II which allows for making measurements in the reflection configuration mode.

The apparent density of PUR/PIR samples was calculated in accordance with PN-EN ISO 845: 2000, as the ratio of the sample weight to the sample volume.

Sol fraction was determined during swelling test. Samples of foams (around 0.2 g) were swollen in xylene for 72 h (room temperature). Sol fraction was calculated as mass difference of samples before swelling (W_1) and after extraction (W_2), according to Eq. (1)

$$\text{Sol fraction} = \frac{W_1 - W_2}{W_1} \times 100\% \quad (1)$$

The compression strength of rigid foams was estimated in accordance with PN-EN ISO 604:2006. The compression

Table 3 Formulation of LC foams

	Foam's symbol				
	LC ₁₀₀	LC ₂₅ E560 ₇₅	LC ₅₀ E560 ₅₀	LC ₇₅ E560 ₂₅	E560 ₁₀₀
LC	100	25	50	75	–
E560	–	75	50	25	100
K15	4.1	4.1	4.1	4.1	2
TMR7	4.1	4.1	4.1	4.1	2
Sn(Oct) ₂	0.4	0.4	0.4	0.4	0.4
Surfactant	2.5	2.5	2.4	2.5	2.5
Water	0.7	0.7	0.7	0.7	0.7
Solkane	34.2	34.2	34.2	34.4	34.4
pMDI	334.2	253.8	283.8	307.8	228.3

test was performed on a Zwick/Roell tensile tester at a constant speed of 10 mm min⁻¹ to 40% deformation.

The thermogravimetric analysis (TGA) was performed on a Netzsch TG 209 apparatus using 5-mg samples within the temperature range 100–600 °C under a nitrogen atmosphere, at a heating rate of 15°C/min.

The cell morphology of polyurethane/polyisocyanurate samples was investigated with a Philips-FEI XL 30 environmental scanning electron microscope (ESEM) using an acceleration of 25 kV. The samples were cut at room temperature, while the observations were performed in wet mode.

The dynamic mechanical analysis was performed using DMA Q800 TA Instruments apparatus. Samples were analyzed in compression mode with a frequency of 1 Hz. Measurements were performed for the temperature range from 25 to 270 °C with heating rate 4 °C min⁻¹. The glass transition temperature (T_g) of PUR/PIR foams was measured by E' onset.

Photo-Oxidative Aging of Rigid PUR/PIR Foams

Foams were subjected to photo-oxidative aging due to ultraviolet radiation (UV-A) according to EN 927-6:2007 standard. Samples were irradiated with UV-A 340 lamps,

Table 2 Formulation of CG foams

	Foam's symbol				
	CG ₁₀₀	CG ₂₅ E560 ₇₅	CG ₅₀ E560 ₅₀	CG ₇₅ E560 ₂₅	E560 ₁₀₀
CG	100	25	50	75	–
E560	–	75	50	25	100
K15	2	2	2	2	2
TMR7	2	2	2	2	2
Sn(Oct) ₂	0.4	0.4	0.4	0.4	0.5
Surfactant	2.5	2.5	2.5	2.5	2.5
Water	0.7	-	0.7	0.7	0.7
Solkane	34.2	34.2	34.2	34.4	34.4
pMDI	253.8	241.0	253.8	266.3	228.3

$\lambda = 360$ nm, which accounts for the energy of ~ 330 kJ mol⁻¹. Such energy is enough to cause dissociation of bonds present in polyurethane/polyisocyanurate backbone and generate free radicals initiating decomposition processes. Aging was performed in an air atmosphere at a constant temperature of 23 °C and relative humidity of 60%.

Thermo-Oxidative Aging of Rigid PUR/PIR Foams

Foams were subjected to thermo-oxidative aging in accordance with ASTM D3574-03 standard. Cylindrical samples were kept at 140 °C for 22 h. After that, samples were cooled down before further analyses.

Results and Discussion

Figure 1 shows the values of apparent density and compressive strength of prepared rigid polyurethane/polyisocyanurate foams are presented. The apparent density of (PUR/PIR) foams usually affects greatly their mechanical strength, due to its increase compressive strength also raises [6, 38].

Compressive strength and apparent density ranged between 105 and 375 kPa and 39–43 kg m⁻³ respectively. A direct correlation between bio-polyol content and both compressive strength and foam density was observed. Similar interrelationships associated with application of bio-polyols in (PUR/PIR) foams were observed before [39].

Polyurethane/polyisocyanurate foams obtained with crude glycerol derived polyol show better mechanical properties which suggests that not only apparent density but also chemical structure and foam morphology have an effect on compressive strength. Structure anisotropy visible on SEM images indicates differences in mechanical properties, which has been noticed in other paper [40].

The FTIR spectra of the analyzed materials are presented in Fig. 2. Symmetrical and the asymmetric stretching vibrations of N-H groups can be observed in the range between 3325 and 3299 cm⁻¹. Signals at 1519–1509 cm⁻¹ correspond to bending vibrations of these groups. Signals at 2928–2900 cm⁻¹ represent asymmetric C-H stretching vibrations in the CH₂ groups present in the aliphatic chains and CH₃ of the terminal groups. The low intensity of the peaks at 2357–280 cm⁻¹ results from the presence of unreacted -N=C=O groups, which indicates that the excess of isocyanate was used relatively to the polyol content [41, 42]. This is related to the aforementioned peaks present in the urethane groups in the range of 1406–1414 cm⁻¹, which have been assigned to the isocyanurate rings derived from the trimerization of isocyanates [43]. The presence of carbonyl C=O bonds in urethane groups was observed at 1718–1707 cm⁻¹. Signals at 1596–1591 cm⁻¹ represent aromatic groups present in ultra-fine polyurethane/polyisocyanurate foams [44]. The tensile vibrations in the range of

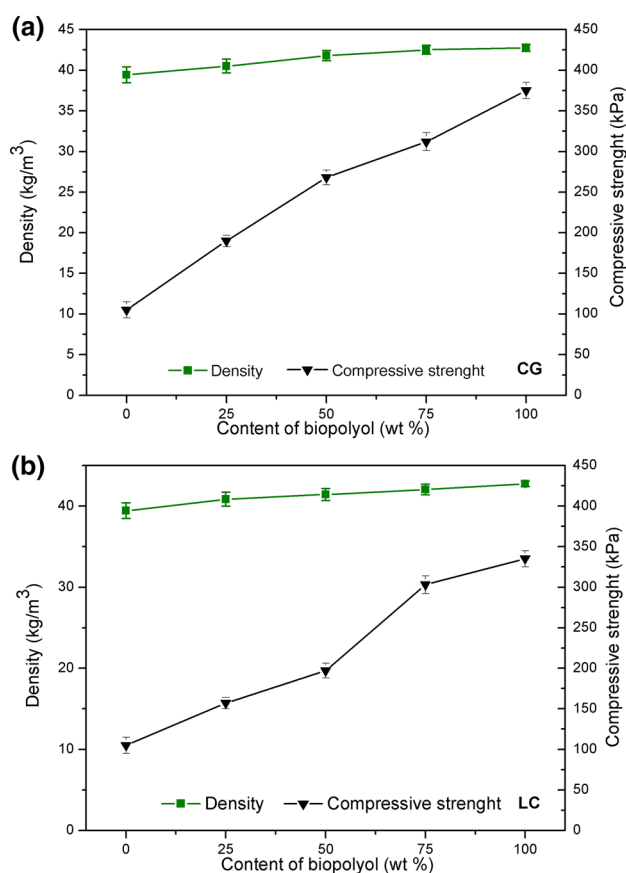


Fig. 1 The effect of bio-polyol content on compressive strength and density of PUR/PIR foams for **a** CG and **b** LC

1221–1211 cm⁻¹ corresponded to C–N groups. The signals at 1062–1072 cm⁻¹ were assigned to the C–O groups present in the flexible segments in polyurethane chains [45]. The 967–962 cm⁻¹ signals correspond to the asymmetric vibrations of the P–O–C group that have been observed in materials containing more than 50% of the E560 polyol [46]. Detailed analysis of shifts attributed to different ratios of polyols used in rigid polyurethane/polyisocyanurate materials can be found in Tables 4 and 5.

The hydrogen bonding index (R) and the content of rigid polyurethane/polyisocyanurate segments may be used to calculate phase separation degree (DSP) from the FTIR results [45]. The indicator (R) was analyzed based on the intensity of the multiplet bands derived from the carbonyl groups present in the urethane and urea formations. For this purpose, a baseline for carbonyl vibration (1770–1630 cm⁻¹) was determined. Using the Gaussian curve fitting method applying Origin 6.0 software, the intensity of these bands was estimated.

The hydrogen bond ratio was calculated using the following equation:

$$R = \frac{A_1 + A_2}{A_3 + A_4} \quad (2)$$

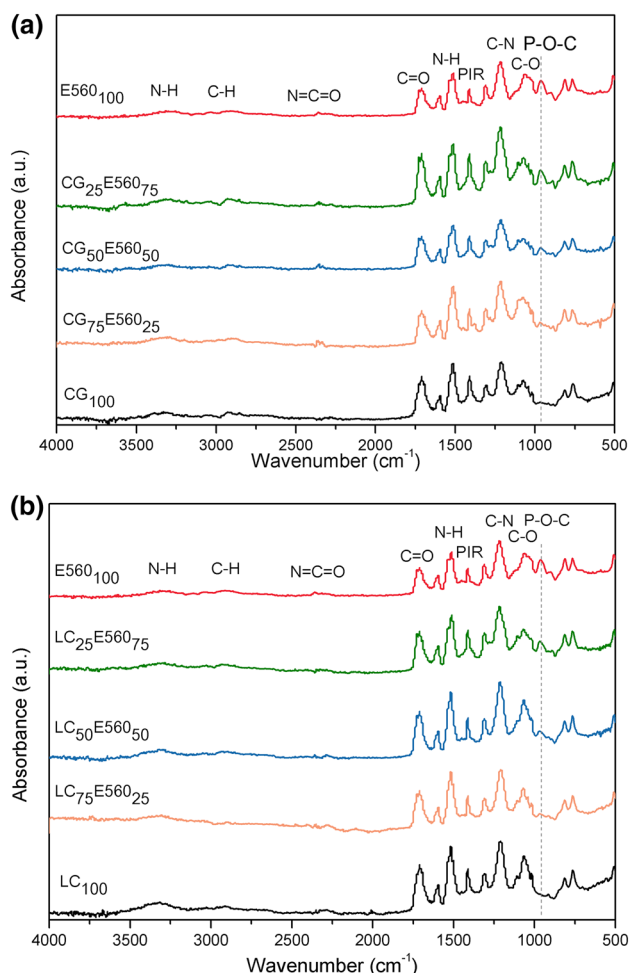


Fig. 2 FTIR spectra of prepared PUR/PIR foams **a** CG and **b** LC

having A_1 , A_2 as a range of bands corresponding to hydrogen bonded carbonyl bonding in urea groups ($1640\text{--}1686\text{ cm}^{-1}$) and urethane groups ($1705\text{--}1724\text{ cm}^{-1}$); A_3 , A_4 as a range of bands corresponding to carbonyl not bound by hydrogen

in urea groups ($1690\text{--}1702\text{ cm}^{-1}$) and urethane groups ($1732\text{--}1760\text{ cm}^{-1}$).

Based on the hydrogen bonding index (R) value the phase separation degree (DSP) parameter has been calculated using the following equation:

$$DSP = \frac{R}{R + 1} \quad (3)$$

The data from Table 6 indicate that phase separation depends on the content of bio-polyol in the test material. The highest hydrogen bonding ratio was obtained for $CG_{75}E560_{25}$ ($R = 1.68$). The introduction of CG biopolymer resulted in the formation of more hydrogen bonds compared to the foam using $E560_{100}$. These changes may be due to the plasticizing effect of the phosphorus polyol. The higher crosslinking degree is associated with an increase in the hardness of the hard segments, and this has resulted in a rise in compressive strength in the case of materials with high bio-polyol content [7].

Parameters describing the cell structure of the analyzed materials are presented in Table 7. All parameters were determined from the microscope images (SEM) shown in Fig. 3. The foams' formulations were modified by varying the amount of Solkane® blowing agent to ensure uniform apparent density. Increasing bio-polyol CG or LC content has reduced the amount of physical blowing agent required to provide the same level of volume expansion during foaming, which is a great advantage in economic and technological terms.

In order to determine the shape parameters like aspect ratio (AR) and roundness (R) of the pores, the computer program fitted the ellipses to the foam cells. The parameters were determined based on the corresponding mathematical equations:

$$AR = \frac{L_l}{L_s} \quad (4)$$

$$R = 4 \times \frac{A}{\pi \times L_l^2} \quad (5)$$

Table 4 Analysis of signals in FTIR spectroscopy of CG materials

CG ₁₀₀ Wavenumber (cm ⁻¹)	CG ₇₅ E560 ₂₅	CG ₅₀ E560 ₅₀	CG ₂₅ E560 ₇₅	E560 ₁₀₀	Bond (vibration)
3325	3303	3299	3314	3300	N-H (stretching)
2928	2900	2920	2920	2908	C-H (asymmetric stretching)
2280	2363	2357	2357	2357	N=C=O (stretching)
1711	1707	1709	1708	1716	C=O (stretching)
1593	1592	1593	1592	1596	Ar-H (deformation)
1512	1509	1512	1509	1510	N-H (bending)
1410	1406	1410	1409	1414	PIR (deformation)
1211	1218	1213	1215	1221	C-N (stretching)
1068	1069	1072	1068	1062	C-O (stretching)
—	—	964	965	962	P-O-C (asymmetric stretching)

Table 5 Analysis of signals in FTIR spectroscopy of LC materials

LC ₁₀₀ Wavenumber (cm ⁻¹)	LC ₇₅ E560 ₂₅	LC ₅₀ E560 ₅₀	LC ₂₅ E560 ₇₅	E560 ₁₀₀	Bond (vibration)
3314	3306	3314	3300	3300	N–H (stretching)
2908	2900	2920	2928	2908	C–H (asymmetric stretching)
2280	2268	2363	2361	2357	N=C=O (stretching)
1712	1707	1711	1718	1716	C=O (stretching)
1595	1591	1595	1595	1596	Ar–H (deformation)
1516	1516	1519	1514	1510	N–H (bending)
1409	1407	1412	1412	1414	PIR (deformation)
1211	1211	1214	1217	1221	C–N (stretching)
1065	1068	1066	1069	1062	C–O (stretching)
–	–	967	966	962	P–O–C (asymmetric stretching)

Table 6 Hydrogen bonding index (R) and phase separation degree (DPS) of tested PUR/PIR samples

Sample	R	DSP (%)
CG ₁₀₀	1.57	61
CG ₇₅ E560 ₂₅	1.68	63
CG ₅₀ E560 ₅₀	1.49	60
CG ₂₅ E560 ₇₅	1.48	60
E560 ₁₀₀	1.53	60
LC ₁₀₀	1.54	61
LC ₇₅ E560 ₂₅	1.53	60
LC ₅₀ E560 ₅₀	1.50	60
LC ₂₅ E560 ₇₅	1.52	60

where L_l and L_s are the respective lengths of the major and minor axes of an ellipse, and A stands for the square area of the ellipse.

For perfectly round pores, the values of AR and R are equal to 1. It has been observed that the introduction of bio-polyols into foam systems does not significantly affect the shape of cells with an AR of 1.6 and R equal to 0.7. This demonstrates the uniform evaporation of the blowing agent during the foaming process. The introduction of CG and

LC to foams resulted in a reduction in the pore diameter, as can be seen in Fig. 3. This is due to the fact that bio-polyol affects the cell nucleation process in the initial phase of foams' formation. Also, the amount of blowing agent used increases the cell size [47].

Table 7 also shows the content of the sol fraction of PUR/PIR rigid foams. Increasing content of CG and LC resulted in a reduction of sol fraction, which is related to the increased density of crosslinking of material [48].

The data presented in Table 7 indicate an increase in the glass transition temperature (T_g) with the increasing content of bio-polyol in the polyols mixture. This is confirmed by the results obtained during the swell equilibration and by tensile strength tests. Similar results from the introduction of bio-polyol to rigid PUR/PIR foams were also reported in another study [49]. The authors linked T_g growth with the reduced elasticity of the bio-polyol chains as compared to the commercial polyol due to an increased number of higher order hydroxyl groups which increased the crosslinking density of the material.

The results of the thermogravimetric analysis of the tested PUR/PIR foams are presented in Table 8. The temperature

Table 7 Parameters describing the cellular structure of the obtained materials

Foam symbol	Properties				T_g E' (°C)
	Average cell diameter (μm)	Cell aspect ratio	Cell roundness	Sol fraction (wt%)	
CG ₁₀₀	145	1.56	0.71	3.84	225
CG ₇₅ E560 ₂₅	156	1.47	0.73	5.03	191
CG ₅₀ E560 ₅₀	183	1.48	0.73	6.06	185
CG ₂₅ E560 ₇₅	208	1.52	0.71	7.00	172
E560 ₁₀₀	242	1.64	0.70	4.80	203
LC ₁₀₀	181	1.55	0.69	1.65	207
LC ₇₅ E560 ₂₅	194	1.58	0.70	2.29	184
LC ₅₀ E560 ₅₀	199	1.67	0.67	4.91	182
LC ₂₅ E560 ₇₅	214	1.63	0.71	7.11	150

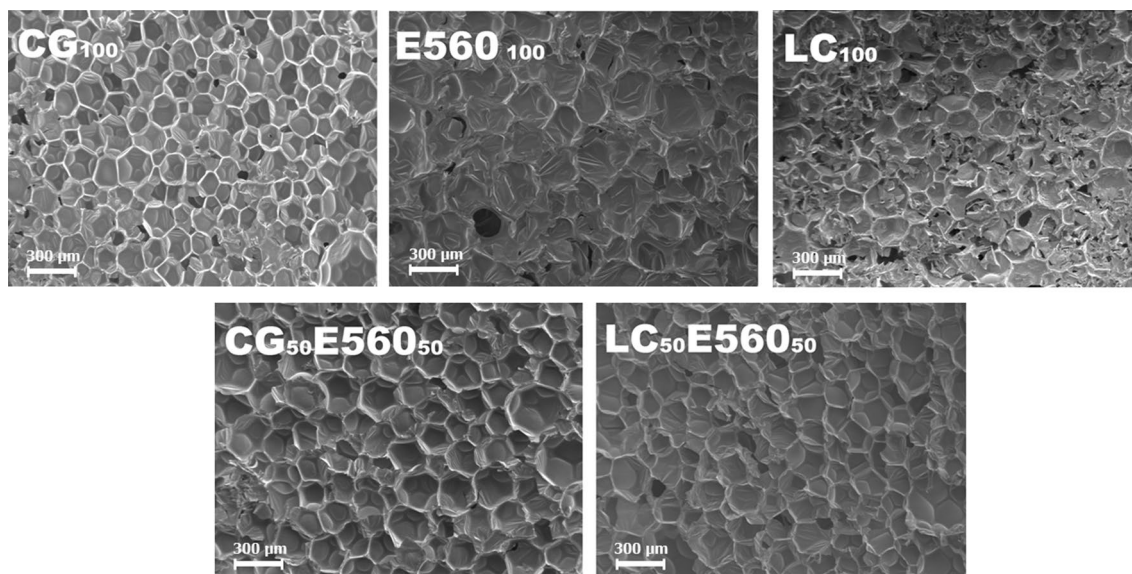


Fig. 3 SEM microscope images of rigid PUR/PIR foams

values for the beginning of the thermal degradation and individual values of the sample weight loss and the temperature positions of the peaks present on the differential thermogravimetry (DTG) are presented. In addition, Fig. 4 shows the DTG curves of the investigated foams.

Thermal degradation of polyurethanes is a very complex process due to the diversity of its structure. Introduction of bio-polyol to rigid PUR/PIR foams has led to an increase in the temperature of early stage of thermal degradation. Figure 4 and Table 8. show that the phosphorus contained in E560 polyol accelerates degradation by shifting $T_{\max 1}$ and $T_{\max 2}$ to lower temperatures while simultaneously stabilizing the second degradation [46]. The first one corresponded to the cleavage of the urethane-urea linkages and the second degradation step to the degradation of the polyol. Then, about 300 °C, the degradation of rigid polyurethane

segments begins, which consists in the breakdown of biuret, allophanate and urethane bonds leading to the formation of low molecular amines, intermediates and carbon dioxide [8, 50–52].

Replacing the phosphorus polyol through CG and LC resulted in a shift of the peaks on the DTG curves to higher temperatures, which confirms the increased thermal stability of the foams. This may also suggest a decrease in the amount of volatile compounds and an improvement in the thermal resistance of rigid segments in the obtained materials.

Also very important from the application point of view is the flammability of rigid PUR/PIR foams used as thermal insulation materials because it is directly related to the safety of health and human life. In order to determine the flammability of the foams obtained, they were tested using the UL 94 flammability test.

Table 8 Characteristics of thermal degradation in rigid polyurethane/polyisocyanurate foams

Sample	Onset of degradation (°C)	Mass loss (%)		$T_{\max 1}$ (°C)	$T_{\max 2}$ (°C)	$T_{\max 3}$ (°C)	$T_{\max 4}$ (°C)
		5	10				
Temperature (°C)							
CG ₁₀₀	280.4	191.8	242.2	181.3	309.4	411.5	470.4
CG ₇₅ E560 ₂₅	272.0	207.8	247.0	233.4	327.1	–	–
CG ₅₀ E560 ₅₀	239.0	216.4	236.0	233.6	322.7	–	–
CG ₂₅ E560 ₇₅	231.9	210.6	227.8	227.4	320.5	415.5	–
E560 ₁₀₀	246.2	248.1	262.2	272.3	318.5	385.7	–
LC ₁₀₀	230.5	185.2	260.1	172.4	248.8	329.9	397.9
LC ₇₅ E560 ₂₅	227.7	219.6	249.8	276.1	328.5	372.2	–
LC ₅₀ E560 ₅₀	227.2	220.6	235.2	234.4	326.3	417.5	–
LC ₂₅ E560 ₇₅	222.0	217.7	230.8	230.5	320.4	415.2	–

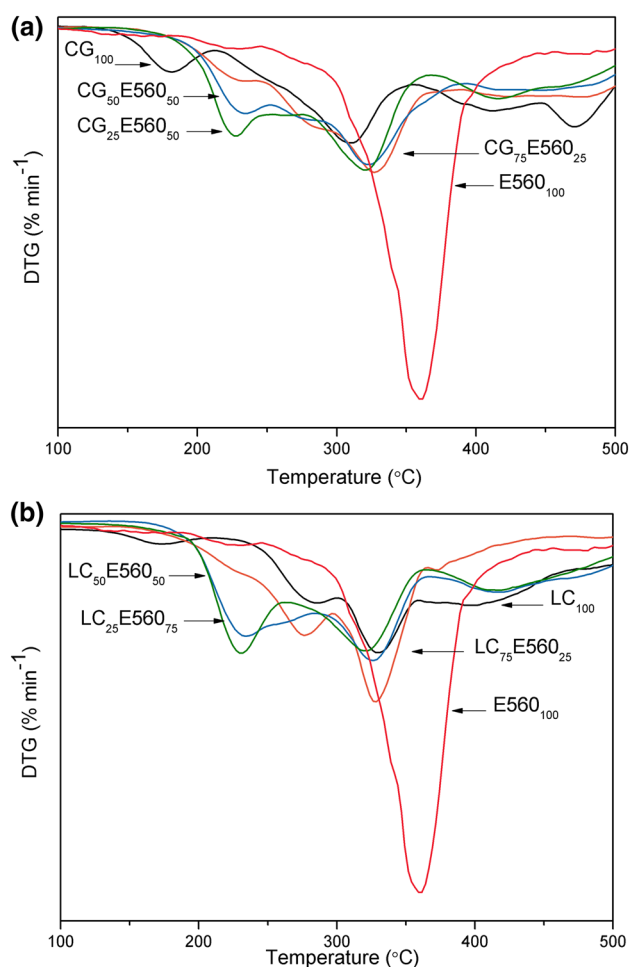


Fig. 4 DTG curves of the prepared PUR/PIR foams **a** CG and **b** LC

As shown in Table 9, the CG100 and LC100 do not meet the flammability class according to UL 94 HB and V-0 due to the high flame range. Containing 25% of the weight of the phosphoric polyol resulted in a change in the combustion of the material compared to the sample, which should be considered as an advantage of this modification.

The accelerated aging process was also used to better understand the effect of the rigid polyurethane/polyisocyanurate foam degradation process on mechanical properties and apparent density as shown in Figs. 5 and 6. Evident differences in compressive strength and apparent density were observed for materials subjected to thermo-oxidative aging. The CG₅₀E560₅₀ foam exhibits a 10% reduction in mechanical strength in the case of photo-aging and a 50% reduction in thermo-oxidative aging. Similar results were obtained for LC₅₀E560₅₀ foam with a compression strength of 15% for photodegradation and 27% for thermo-oxidative aging. The decrease in mechanical properties has also been described by Leuteritz et al. [53].

Table 9 Parameters describing the flammability of the prepared rigid PUR/PIR foams

Sample	UL 94 HB	UL 94 V	Flame range (mm)
CG ₁₀₀	–	–	112
CG ₇₅ E560 ₂₅	HB-A	V-0	30 ± 2
CG ₅₀ E560 ₅₀	HB-A	V-0	20 ± 1
CG ₂₅ E560 ₇₅	HB-A	V-0	12 ± 2
E560 ₁₀₀	HB-A	V-0	11 ± 3
LC ₁₀₀	–	–	124
LC ₇₅ E560 ₂₅	HB-A	V-0	21 ± 3
LC ₅₀ E560 ₅₀	HB-A	V-0	16 ± 2
LC ₂₅ E560 ₇₅	HB-A	V-0	12 ± 2

Figure 7. shows FTIR spectra of rigid PUR/PIR foams subjected to thermal- and photo-oxidation processes to determine the effect of these processes on the chemical structure of the materials analyzed. The results are similar to those reported by other researchers [54]. Photo-oxidative aging produces a new band near 1770 cm^{-1} which indicates the formation of carbonyl groups by oxidation. The band between 2900 and 2928 cm^{-1} comes from CH_2 and is connected to chemical reactions involving these groups, under the influence of UV radiation and elevated temperatures [55]. There is also a change in the intensity and shape of the band at 3299–3325 cm^{-1} (N-H stretching) which indicates the breakdown of urethane groups by UV radiation. This phenomenon has also been confirmed by Romanova et al. [56]. Significant changes in the FTIR spectrum are more evident for photo-oxidative aged materials. Differences in the intensity of spectra show changes in the area of carbonyl groups (1800–1600 cm^{-1}) due to UV-induced oxidation. The range in this area is wider, indicating the emergence of new structures containing C=O bond.

Conclusions

Rigid polyurethane/polyisocyanurate foams were prepared by substitution 25, 50, 75 and 100 wt% of a bio-polyol with the phosphorus-containing polyol. The bio-polyols which were used for production foams were synthesized via two-step process from crude glycerol and castor oil and lignocellulose biomass liquefaction process. The hydroxyl numbers of polyols were respectively: crude glycerol (CG) $L_{\text{OH}} = 560 \text{ mg KOH/g}$, lignocellulose biomass (LC) $L_{\text{OH}} = 750 \text{ mg KOH g}^{-1}$ and phosphorus-containing polyol (E560) $L_{\text{OH}} = 450 \text{ mg KOH g}^{-1}$. Formulations of foams were adjusted to provide a similar apparent density of analyzed materials. The compressive strength of foams decreased with increasing amounts phosphorus-containing polyol reaching

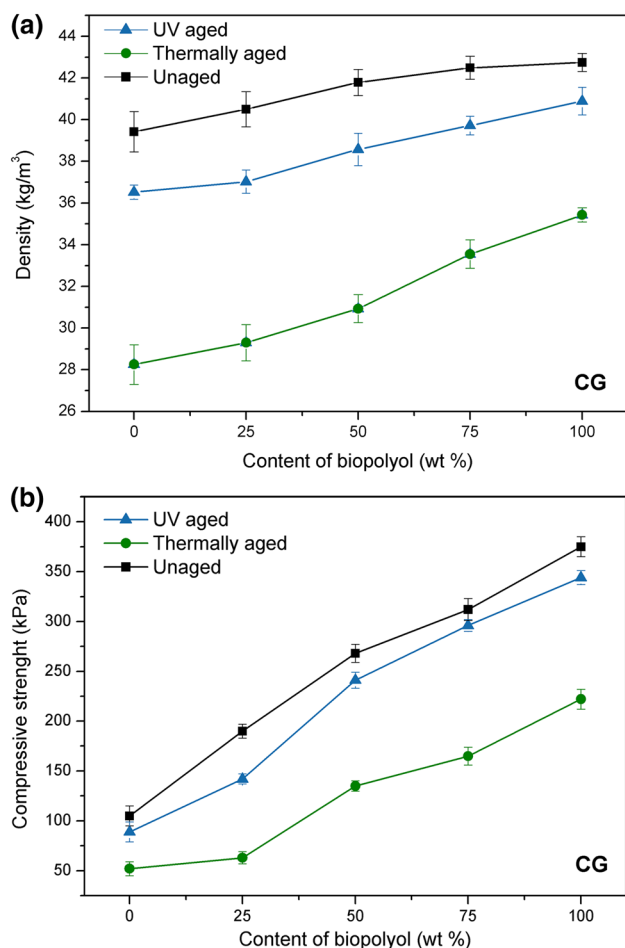


Fig. 5 Effect of bio-polyol content **a** on density and **b** compressive strength of PUR/PIR and CG foams

a value of 105 kPa for foams containing only phosphorus polyol. The SEM photos of the surface of rigid (PUR/PIR) foams showed increased of cell size with the addition of E560 polyol and changes in cell aspect ratio and cell roundness. Replacement of the bio-polyol by phosphorus-containing polyol resulted in higher of thermal stability of foams. 25% of weight addition of phosphoric polyol renders these materials capable of fulfilling the flammability classes HB-1 and V-0. The impact of thermo-oxidative and photo-oxidative aging on the structure and properties of rigid polyurethane/polyisocyanurate foams was also investigated. The apparent density and mechanical properties of phosphorus-containing polyol based PUR/PIR decreased

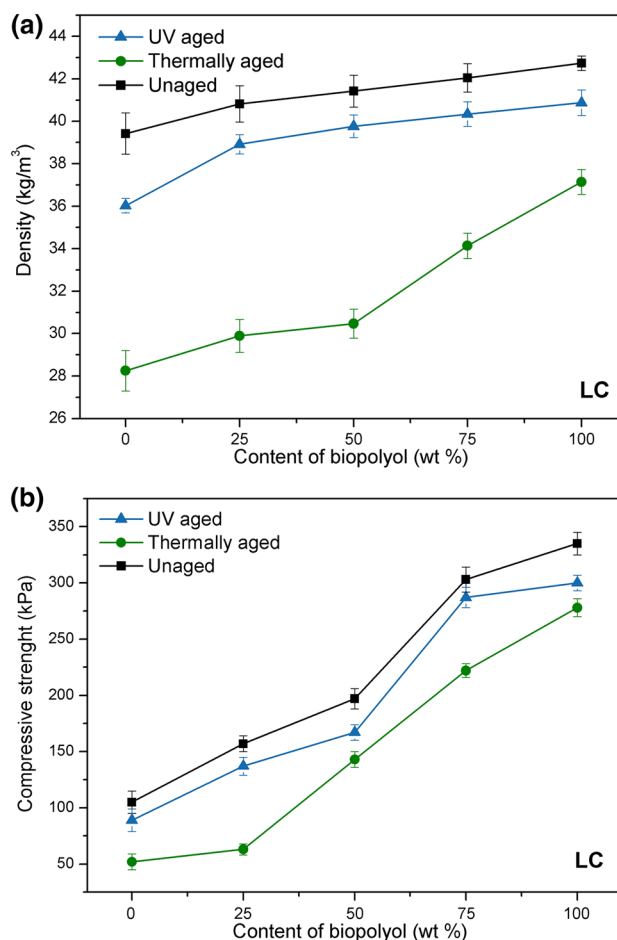


Fig. 6 Effect of bio-polyol content **a** on density and **b** compressive strength of PUR/PIR and LC foams

obviously with aging process time. Bigger changes in the apparent density and compressive strength were observed for thermo-oxidative aging.

Summarizing, this work confirms that two types of bio-polyols can be successfully applied in the production of polyurethane/polyisocyanurate foams. Foams produced with the use of obtained bio-polyols are characterized by acceptable mechanical and their properties. The result shows also that substitution 25 wt% bio-polyols by phosphorus-containing polyol caused the foams fulfilling the flammability classes.

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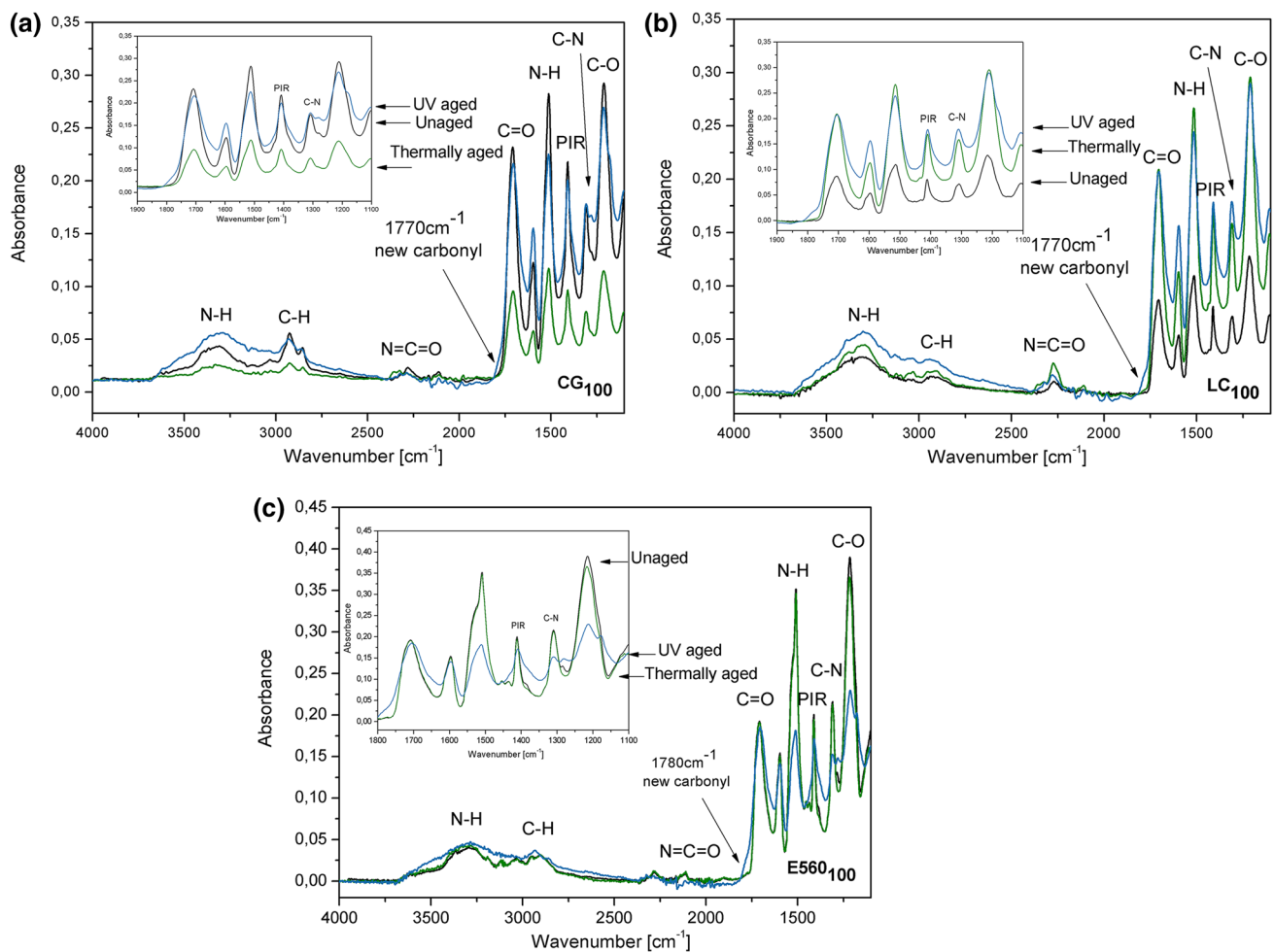


Fig. 7 FTIR spectra of prepared PUR/PIR foams **a** CG₁₀₀, **b** LC₁₀₀ and **c** E560₁₀₀

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