ORIGINAL RESEARCH



Enhanced Photoelectrocatalytical Performance of Inorganic-Inorganic Hybrid Consisting BiVO₄, V₂O₅, and Cobalt Hexacyanocobaltate as a Perspective Photoanode for Water Splitting

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Abstract

Thin layers of $BiVO_4/V_2O_5$ were prepared on FTO substrates using pulsed laser deposition technique. The method of cobalt hexacyanocobaltate (Cohcc) synthesis on the $BiVO_4/V_2O_5$ photoanodes consists of cobalt deposition followed by electrochemical oxidation of metallic Co in $K_3[Co(CN)_6]$ aqueous electrolyte. The modified electrodes were tested as photoanodes for water oxidation under simulated sunlight irradiation. Deposited films were characterized using UV-Vis spectroscopy, Raman spectroscopy, and scanning electron microscopy. Since the V_2O_5 is characterized by a narrower energy bandgap than $BiVO_4$, the presence of V_2O_5 shifts absorption edge ($\Delta E = \sim 0.25 \text{ eV}$) of modified films towards lower energies enabling the conversion of a wider range of solar radiation. The formation of heterojunction increases photocurrent of water oxidation measured at 1.2 V vs Ag/AgCl (3 M KCl) to over 1 mA cm⁻², while bare $BiVO_4$ and V_2O_5 exhibit 0.37 and 0.08 mA cm⁻², respectively. On the other hand, the modification of obtained layers with Cohcc shifts onset potential of photocurrent generation into a cathodic direction. As a result, the photocurrent enhancement at a wide range of applied potential was achieved.

Keywords Photoanode · Water splitting · Visible light activity · Pulsed laser deposition

Introduction

The scientific community has a moral obligation to take up efforts to reduce emissions of greenhouse gases. Photoelectrochemical water splitting is one of the most promising methods of hydrogen generation which is of great importance considering climate changes [1]. Sunlight energy conversion to environmentally desired fuel requires efficient photocatalysts. Since 1972, when Fujishima and Honda published their milestone paper [2], many different materials were tested as photoanodes for water photooxidation. Among them is bismuth vanadate, which exhibits high absorption coefficient [3] and a relatively narrow energy bandgap [4]. However, BiVO₄ suffers from poor kinetics of water oxidation and low

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mobility and separation efficiency of charge carriers [5]. There are many strategies of enhancement of the BiVO₄ photocatalytic properties that can minimize the main drawbacks [6]. One of them is based on the formation of heterojunction. It can be nn [7] as well as n-p [8, 9] junction built from BiVO₄ and another semiconductor. The most commonly tested system is based on BiVO₄ and WO₃; however, BiVO₄ has been already combined with, e.g., Bi₄V₂O₁₁ [10], TiO₂ [11], and MnO₂ [12]. This manuscript is focused on the photoelectrochemical properties of BiVO₄/V₂O₅ bulk system. The photocatalytic and antibacterial properties of such a composite have been already studied, e.g., [13, 14]. Recently, BiVO₄/V₂O₅ junction has been characterized as a promising photoanode for water splitting. The various geometries of the junction (e.g., bulk, layer on layer) have been studied [15]. The V₂O₅ presence positively affects charge separation efficiency and light absorption ability due to the narrower energy bandgap. BiVO₄ was additionally doped using W atoms, and it leads to one of the highest photocurrents of water oxidation obtained for BiVO₄-based photoanodes [15]. Other authors claim that the presence of V_2O_5 in BiVO₄/V₂O₅ junction enhances mobility of charge carriers measured as transit time for the migration of photoexcited electrons to counter electrode, what was confirmed using intensity-



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modulated photocurrent spectroscopy [16]. The Bi/BiVO₄/ V₂O₅ composite was obtained by annealing of pure bismuth vanadate powder in Ar/H₂ gas mixture. Obtained powder exhibited photocatalytic properties towards water oxidation reaction in AgNO₃ solution. The films of resulting material were tested as photoanodes, but only for hole scavenger photoelectrooxidation [17]. Significant enhancement of water oxidation photocurrent has been achieved for inverse opal heterostructure (V₂O₅/BiVO₄) modified using NiOOH due to the specific morphology, presence of the heterojunction, and electrode/electrolyte interface modified by cocatalyst [18].

The formation of a junction may positively affect the electron/hole pairs separation on the interface between the components as well as, in some cases, enhancing photoabsorption ability. Nevertheless, such a way of modification does not affect the surface recombination that takes place on the BiVO₄/electrolyte surface. The passivation of recombination centers and enhancement of oxygen evolution kinetics can be achieved via cocatalyst loading. BiVO₄-based photoanodes are commonly modified using oxygen evolution catalysts, i.e., FeOOH/NiOOH [19], cobalt phosphate (CoPi) [20], and cobalt borate [21]; however, the catalytic role of cocatalyst is not the most important. It is claimed that the presence of CoPi on BiVO₄ suppresses surface recombination and does not affect the kinetics of water oxidation [22]. However, appropriate cocatalyst on the top of BiVO₄ may affect both the kinetics of O₂ evolution reaction and inhibits surface recombination [23]. Thus, the discussion about the role of oxygen evolution cocatalysts in photoelectrochemical water splitting is still open.

Prussian blue analogues (PBA) are interesting group of oxygen evolution catalysts. Generally, the cobalt center coordinated by the nitrogen atoms from the cyanide group in PBA is essential to obtain efficient electrocatalyst [24]. However, the enhancement of photocurrent generation of water oxidation by BiVO₄ modified using nickel hexacyanoferrate has been also reported [25]. Some of PBA, e.g., cobalt hexacyanoferrate can accumulate photoexcited holes from BiVO₄, positively affecting the efficiency of photocurrent generation [26]. Thus, PBAs which exhibit electrocatalytic properties in oxygen evolution reaction can be successfully utilized to modification of electrode/electrolyte interface in BiVO₄-based photoanodes.

In the present work, bismuth vanadate was modified in two different ways. The first one is based on a formation of the BiVO₄/V₂O₅ bulk heterojunction. The phase of V₂O₅ can be formed from the excess of the V-source during the synthesis of BiVO₄. The thin films were prepared using pulsed laser deposition technique. The role of V₂O₅ in the photoelectrochemical water oxidation is discussed here. In order to enhance photocurrent generation in a wider range of applied potential, BiVO₄/V₂O₅ photoanodes were modified by cobalt hexacyanocobaltate (Cohcc) synthesize during two-step method. The choice of Cohec is based on previous reports showing its catalytic activity towards water oxidation. Among tested cobalt hexacyanometalates, cobalt hexacyanocobaltate exhibited the lowest overpotential of water oxidation [24]. We have also reported previously that the presence of Cohcc on the BiVO4 affects its photoelectrochemical performance as well [27]. Both ways of BiVO₄ modification enhance the efficiency of water splitting in a different way. In both cases, significant enhancement of anodic photocurrent has been demonstrated.

Experimental

Chemicals

FTO (fluorine-doped tin oxide) glass slides were purchased from Sigma and used as electrode substrates. Chemicals of analytical grade, Bi(NO₃)₃·5H₂O, NH₄VO₃, V₂O₅, K₃[Co(CN)]₆ were supplied by Sigma-Aldrich. K₂SO₄, acetone, and isopropanol were supplied by POCH. The metallic Co target (TK 8900) for magnetron sputtering was purchased from Quorum. Triple distilled water was used for all electrochemical experiments.

Layer Preparation

The high-temperature solid-state chemical reaction using of Bi(NO₃)₃·5H₂O and an excess amount of NH₄VO₃ (the Bi:V molar ratio equals to 1:1.5) was used to obtain the BiVO₄/V₂O₅ powder via annealing at 500 °C for 5 h. The resulting mixture was homogenized and pressed into a pellet, and annealed again at 500 °C for 4 h. The BiVO₄/V₂O₅ pellet acted as a target for the pulsed laser deposition (PLD) technique. PLD was performed using a laser (Nd:YAG) equipped with a 4th harmonic generation module emitting 6 ns pulses at 266 nm (4 pulses per 1 s). The energy density of the laser was established at about 6.5 J cm². The deposition process was performed at room temperature in an oxygen atmosphere $(p_{O2} \sim 1 \ 10^{-2} \text{ mbar})$. Deposition took 60 min. Then, the samples were annealed in a tube furnace (PRC 55 L/1300 M, Czylok) for 2 h at 450 °C in air atmosphere (heating rate 2 °C/min) in order to obtain crystalline BiVO₄/V₂O₅ films. Deposition of FTO/V₂O₅ and FTO/BiVO₄ was performed for a comparison at the same conditions, but from separate targets, (a) V₂O₅ – pure oxide and (b) BiVO₄ (synthesized using equimolar amounts of Bi(NO₃)₃·5H₂O and NH₄VO₃).

The Cohcc deposition was performed according to the procedure reported in our previous report [27]. Briefly, it was a two-step method that consists (1) sputtering of a metallic cobalt using magnetron sputtering and (2) electrooxidation of Co film in 0.05 M $K_3[Co(CN)_6] + 0.1$ M KCl electrolyte. As a result, FTO/BiVO₄/V₂O₅/Cohcc electrodes were obtained.





Research methods

The surface morphology was examined using scanning electron microscopy (SU3500, Hitachi). Raman spectra were recorded by a confocal micro-Raman spectrometer (InVia, Renishaw) with sample excitation, by means of an argon ion laser emitting at 514 nm and operating at 5% of its total power (50 mW). The transmittance of the samples was measured by UV-Vis spectrometer (Lambda 35, Perkin-Elmer). The spectra were registered in the range of 300-700 nm, with a scanning speed of 120 nm min⁻¹. The calibration of UV-Vis spectrometer was performed using bare FTO substrate.

The electrochemical and photoelectrochemical studies of materials were conducted using the AutoLabPGStat 302 N potentiostat-galvanostat system (Methrom, AutoLab) in the one-compartment three-electrode cell with a quartz window, where photoanode served as a working electrode (geometric surface area of ~0.5 cm²). The Pt mesh with the high surface area was used as a counter electrode, while Ag/AgCl (3 M KCl) as a reference electrode. The electrochemical tests were carried out in deaerated 0.2 M K₂SO₄. Scan rate during photoelectrochemical measurements was equal to 20 mV s⁻¹ A xenon lamp (LOT-QuantumDesign) equipped with AM 1.5 filter with a light intensity of 100 mW cm⁻² was used as the light source.

Results and Discussion

Scanning Electron Microscopy

In order to investigate the surface morphology of annealed films, the SEM was employed. The SEM images of four films are presented in Fig.1. The film of V₂O₅ deposited using pulsed laser deposition technique crystallizes in the form of coarse, longitudinal crystals. Bismuth vanadate exhibits completely different morphology. The films are built from small grains with ~100-200 nm diameter. In the case of BiVO₄/V₂O₅ composite, the morphology is dominated by the presence of BiVO₄ grains. Elongated V₂O₅ crystals are not present in the SEM image. As can be observed, the grains forming the layer seem to be partially melted on the edges. It may be related to the presence of V₂O₅ that is characterized by the lower than BiVO₄ melting point (690 °C [28] and 940 °C [29], respectively). Films modified by cobalt hexacyanocobaltate are evenly covered by regular, cubic-shaped crystallites characteristic for metal hexacyanometallates [30]. The zoom of the ideal cube of Cohcc is shown in Fig. 1 inset.

Raman Spectroscopy

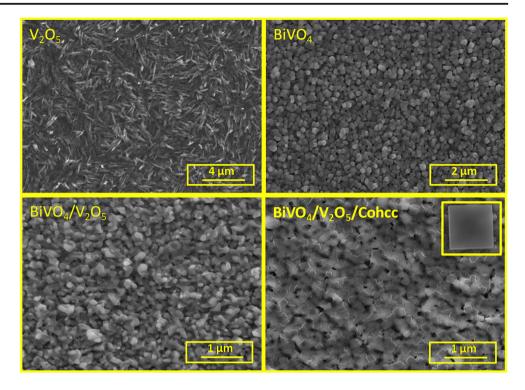
The Raman spectra of the V₂O₅, BiVO₄, BiVO₄/V₂O₅, and BiVO₄/V₂O₅/Cohec are shown in Fig. 2. Samples showed characteristic bands of monoclinic structure BiVO₄ in 828, 709, 367, and 327 cm⁻¹ [31]. A single band at 828 cm⁻¹ is assigned to symmetric stretching V-O while the band at 709 cm⁻¹ is attributed to the antisymmetric stretching V-O. The bands at 367 and 327 cm⁻¹ are assigned to the symmetric and antisymmetric deformation V-O vibrations in VO₄³⁻ units, respectively. The bands at 213 and 130 cm⁻¹ are attributed to external modes. The Raman spectrum of a film deposited from V₂O₅ target confirms that the sputtered layer consist only V_2O_5 [32]. In the case of the layer deposited from the BiVO₄ target with V-source excess, Raman spectrum of resulting layer consists of band characteristic for both bismuth vanadate and vanadium pentaoxide. No other phases are detected, confirming the formation of pure BiVO₄/V₂O₅ junction. Raman spectroscopy has been applied also in order to affirm the chemical structure of Cohcc. As it is shown in Fig. 2, the spectrum of BiVO₄/V₂O₅/Cohcc exhibits new bands in comparison of the film without Cohcc. The set of Raman bands at 2150–2230 cm⁻¹ is characteristic for $C \equiv N$ stretching vibration [33]. The presence of at least two bands at this region suggests that Co centers coexist at different oxidation states. The Raman band of Co-CN vibration at about 490 cm⁻¹ are not clearly detected [34], probably due to the overlapping with bands coming from V₂O₅.

UV-Vis spectroscopy

The influence of V_2O_5 on the optical properties of tested films was investigated using UV-Vis spectroscopy in a transmittance mode, see fig. 3. The spectrum of bismuth vanadate film is characterized by an absorption edge typical for monoclinic BiVO₄ deposited onto transparent conductive oxide substrate [35]. The energy bandgaps were estimated from the absorption edges. The Tauc plot was not utilized here due to the difficulty to choose an appropriate type of an electron transition. BiVO₄ is characterized by allowed direct and allowed indirect transitions [3], while V₂O₅ by direct forbidden transition [32]. The one function cannot be used in order to determine a bandgap of $BiVO_4/V_2O_5$ heterojunction. The E_g of BiVO₄ was estimated to be equal to 2.45 eV. The edge seen on UV-Vis spectrum of V₂O₅ containing film is clearly shifted towards higher wavelengths. The lower energy bandgap of V₂O₅ enhances the absorption ability of the film in a visible range of electromagnetic radiation. The shift of absorption edge of about 0.25 eV is achieved due to the presence of V₂O₅ phase. It is not a very significant change; however, it should positively affect the photoactivity of investigated photoanodes. As it was reported, the absorbance of Prussian blue analogues is negligible due to the very low absorption



Fig. 1 SEM images of FTO/ V₂O₅, FTO/BiVO₄/ FTO/BiVO₄/ V₂O₅, and FTO/BiVO₄/V₂O₅/ Cohcc



coefficient [36] in comparison with absorption coefficients of V₂O₅ and BiVO₄, and can be omitted in a discussion.

Photoelectrochemical Performance

In order to investigate the influence of the presence of V₂O₅ and Cohcc components on the photoelectrochemical properties of BiVO₄-based photoanodes, the linear sweep voltammograms were recorded under the illumination of the tested photoanodes. As it is shown in Fig. 4a, both BiVO₄ and V₂O₅ act as n-type semiconductors in contact with an aqueous

Normalized intensity **V**,**O**, BiVO BiVO₄/V₂O BiVO_/V_O_/Cohc 500 1000 1500 2000 Raman shift / cm⁻¹

Fig. 2 Raman spectra of V_2O_5 , $BiVO_4$, $BiVO_4/V_2O_5$, and $BiVO_4/V_2O_5/V_2O_5$ Cohcc

electrolyte. Registered photocurrent measured at 1.2 V vs Ag/AgCl was equal to 90 and 370 μ A cm⁻² for V_2O_5 and BiVO₄, respectively. There is a significant difference of onset potential, where photocurrent is detectable. Photocurrent of water oxidation was generated at the much lower potential in the case of bismuth vanadate. Such an effect was expected due to the differences of flat-band potential of two photoanode materials [37]. The curve of V₂O₅ consists also a dark current anodic hump at 0.2-0.6 V. It was previously reported that at this range of applied potential, electrochemical oxidation with simultaneous Na⁺ desorption occurs, what was confirmed using electrochemical quartz crystal microbalance and in situ Raman spectroscopy [38]. This phenomenon should be rather

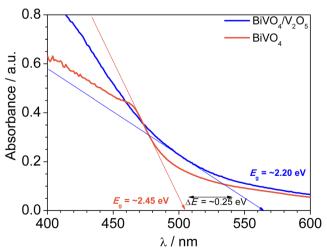


Fig. 3. UV-Vis spectra of $BiVO_4$ and $BiVO_4/V_2O_5$





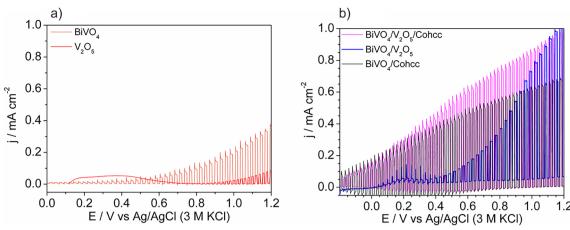
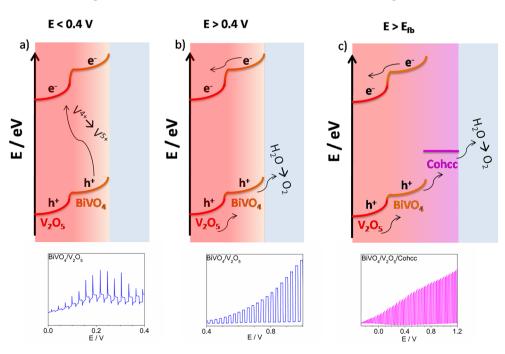


Fig. 4 The linear sweep voltammetry curves of a) FTO/BiVO₄ and FTO/V₂O₅, and b) FTO/BiVO₄/V₂O₅, FTO/BiVO₄/Cohcc (based on previous report [32]) and FTO/BiVO₄/V₂O₅/Cohcc recorded under intermittent simulated solar light illumination

confined to the surface or a few crystallographic planes of the studied thin film than to the bulk oxide because current response is recorded at relatively high sweep rates. The effect of cation intercalation/deintercalation in aqueous electrolytes is even more pronounced for K⁺ cations [39, 40]. It was reported that electroactivity of V₂O₅ due to the Li⁺ intercalation/ deintercalation is related to the partial reduction of V⁵⁺ to V^{4+} centers [41, 42], and it occurs in the aqueous electrolyte as well [43]. The similar effect of dark anodic hump was observed here in the case of BiVO₄/V₂O₅ junction; see Fig. 4b. Interestingly, the photocurrent at this range of potential was registered. It is very likely that part of photoexcited holes on BiVO₄ took part at V₂O₅ oxidation at a specific range of applied potential (~0-0.4 V), where BiVO₄ acts as photoanode (E > E_{fb}), but applied potential is lower than E_{fb} of V₂O₅. The LSV curve of FTO/BiVO₄/Cohec photoanode,

presented in Fig. 4b for comparison, does not exhibit anodic hump. As it is shown, photoanode without V₂O₅ generates lower photocurrent at a more anodic potential. Lack of heterojunction makes bulk e⁻/h⁺ recombination easier to occur. However, curves recorded for photoanodes with and without V₂O₅ almost overlapped at a low potential range that is strongly influenced by the presence of cobalt hexacyanocobaltate. Schematically, the effect of V₂O₅ oxidation taking into account band alignment is presented in Fig. 5a. More anodic potential makes V₂O₅ on the electrode already oxidized, and applied potential is higher than flat-band potential of V₂O₅, thus photoexcited holes from both components take part in water photooxidation as it is outlined in a Fig. 5b. There is a significant enhancement of photocurrent generation due to the bulk heterojunction formation and photocurrent reached ~1 mA cm⁻² at 1.2 V. There are three possible reasons for

Fig. 5 The schematic presentation of photocurrent generation by a), b) BiVO₄/V₂O₅ and c) BiVO₄/V₂O₅/Cohec



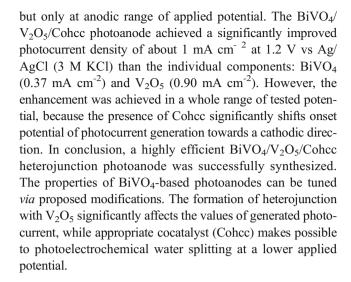


increase in photoactivity. The first one is related to the wider

range of absorbed radiation due to the narrower energy

bandgap of V₂O₅. As it was evidenced (Fig. 4a), photoexcited

Four different photoanode arrangements: FTO/BiVO₄, FTO/ V₂O₅, FTO/BiVO₄/V₂O₅, and FTO/BiVO₄/V₂O₅/Cohec were systematically examined using UV-Vis spectroscopy, Raman Spectroscopy, scanning electron microscopy, and electrochemical measurement under intermittent simulated solar light illumination. The phase of V₂O₅ can be formed from the excess of the V-source during the synthesis of BiVO₄, and no additional reagents are required. Pulsed laser deposition technique allows obtaining BiVO₄/V₂O₅ bulk heterojunction. The photoelectrochemical measurements showed that it is possible to photooxidize V₂O₅ component by photoexcited BiVO₄ at some range of applied potential. Applying a higher potential makes BiVO₄/V₂O₅ a more efficient photoanode for water oxidation (than bare BiVO₄ and V₂O₅) due to the wider utilization of the solar spectrum, improvement of hole mobility, and enhancement of bulk e⁻/h⁺ separation due to the formation of internal electric field on the BiVO₄/V₂O₅ heterojunction. Thus, V₂O₅ improve the photoelectrochemical water splitting,



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