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## **CAVITATION EROSION RESISTANCE INFLUENCE OF MATERIAL PROPERTIES**

### **ABSTRACT**

The cavitation erosion is the phenomena that causes degradation of fluid flow machinery components due to repetitive implosion of cavitation bubbles adjacent to the solid surface. Cavitation erosion is a complex phenomenon, which includes not only hydrodynamic factors of liquid, but also properties of erodible material e.g. microstructure, hardness or Young modulus. In order to reduce the negative impact of erosion on machine components, there are many methods to increase cavitation erosion resistance. The paper discusses the correlations between structural and mechanical properties and the resistance to cavitation erosion (CER) of pure materials, their alloys and coatings. Methods to increase CER have also been described - using heat / thermo-chemical treatment and application of coatings by various methods.

**Keywords:** *cavitation erosion, erosion, resistance, material properties*

### **INTRODUCTION**

According to the ASTM G32 standard, cavitation is defined as the formation and subsequent collapse, within a liquid, of cavities or bubbles that contain vapor, gas or a mixture of vapor and gas [ASTM G-32-10 Standard]. Bubbles filled with vapor or dissolved gases form in low pressure regions and implode violently in areas of higher pressure. During bubbles implosion the micro-jets and the shock waves are generated, which are called cavitation pulses due to their short duration. The repeated impacts of the micro-jets and the shock waves can cause e.g. mass loss of material, this phenomenon is known as cavitation erosion. This affects the life of the devices [1].

Cavitation erosion depends on the physical and chemical properties of liquid and material, and also on the development of cavitation called as cavitation intensity, which can be expressed by the equation [2]:

$$J = \frac{K}{T} \frac{1}{2\rho c} \sum_{k=1}^M n_k p_k^2$$

Where  $K = 10^{-5}$  (proportional coefficient),  $T$  total measurement time,  $\rho$  - density of the liquid,  $c$  - sound velocity,  $M$  - number of pressure intervals,  $k$  - the consecutive number of the interval,  $n_k$  - number of pulses in the interval,  $p_k$  - value of pressure amplitudes in  $k$ -th interval.

The resistance of material to cavitation erosion is presented by the curves of mass loss in exposure time. Instead of mass loss, volume loss or erosion rate in exposure time can be shown (Fig.1) (according ASTM G32-10).

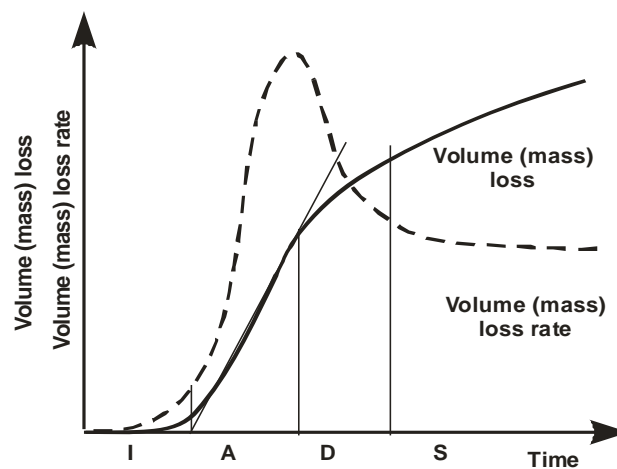


Fig.1. Cavitation curve with periods of degradation

The cavitation curve allows detecting following periods of material degradation: incubation (I), acceleration (A), deceleration (D) and steady state (S). During incubation period, material accumulates impact energy, undergoes elastic and plastic deformation, and the properties of surface layer are changed. The acceleration period is characterized by intensive material loss. At the end of this period erosion rate achieves its maximum value. Cavitation erosion extends over the entire area subjected to cavitation. Changes of surface geometry cause the start of the next period, which is the deceleration period and is characterized by progressive decrease of erosion rate. The impacts of cavitation pulses are amortized by a liquid filling the pits. As a result less impact energy is delivered to material surface layer. The last period of degradation is steady state period, during which the erosion rate is almost constant due to a slight changes in surface geometry structure [3].

The duration of the mentioned erosion periods depends on the material properties, the type of cavitation, cavitation number and cavitation intensity.

Cavitation damage is a serious problem not only in hydro turbines [4], but also in other hydraulic devices, e.g. propellers (the interaction between a ship's propeller and the non-uniform wake behind the ship) or pump impellers. Rayleigh was the first person who reported this type of damage on ship propellers. Degradation is mainly caused by the coherent collapse of a cloud of cavitation bubbles, which is more violent than that of individual bubbles and accompanied by intense noise [5], [6].

Although the cavitation phenomenon is very undesirable due to the risk of the occurrence of erosion, in some circumstances, some effects of cavitation are applied in advanced

oxidation of substances that pollute the natural environment [7], e.g. in decomposition of selected organic compounds, which hardly undergo biodegradation, intensification of biological wastewater treatment, ultrasonic emulsification and synthesis of biodiesel [8].

Due to harmful effects of cavitation, there are some ways to their prevention. Karimi and Martin [9] classified these methods into three groups. The first group includes improvement of the hydrodynamic designs, the second group includes the design of high cavitation erosion resistance materials and the last group includes simultaneous changes in material properties and hydrodynamic design to diminish cavitation intensity.

In order to predict the erosion rate, it is important to determine the relationship between the material properties and their cavitation resistance. However, the correlations found so far are not universal and usually cover a limited group of materials [10]. The aim of this paper is to present properties of eroded material affecting the cavitation erosion and their correlation with cavitation erosion resistance, and possible ways to increase resistance to cavitation erosion.

### CAVITATION EROSION OF PURE METALS AND METAL ALLOY STRUCTURES

The cavitation erosion resistance depends on material properties (mechanical, endurance and structural), liquid properties (e.g. temperature, chemical composition, chemical aggressiveness, viscosity) and cavitation intensity. According Fig. 2., the highest cavitation erosion resistance has material with homogeneous and fine-grained structure, high endurance and mechanical properties, and high corrosion resistance. To the key material properties belongs hardness, tensile strength, yield strength, strain energy, fracture toughness and fatigue strength. Therefore, their effect on erosion resistance is described below.

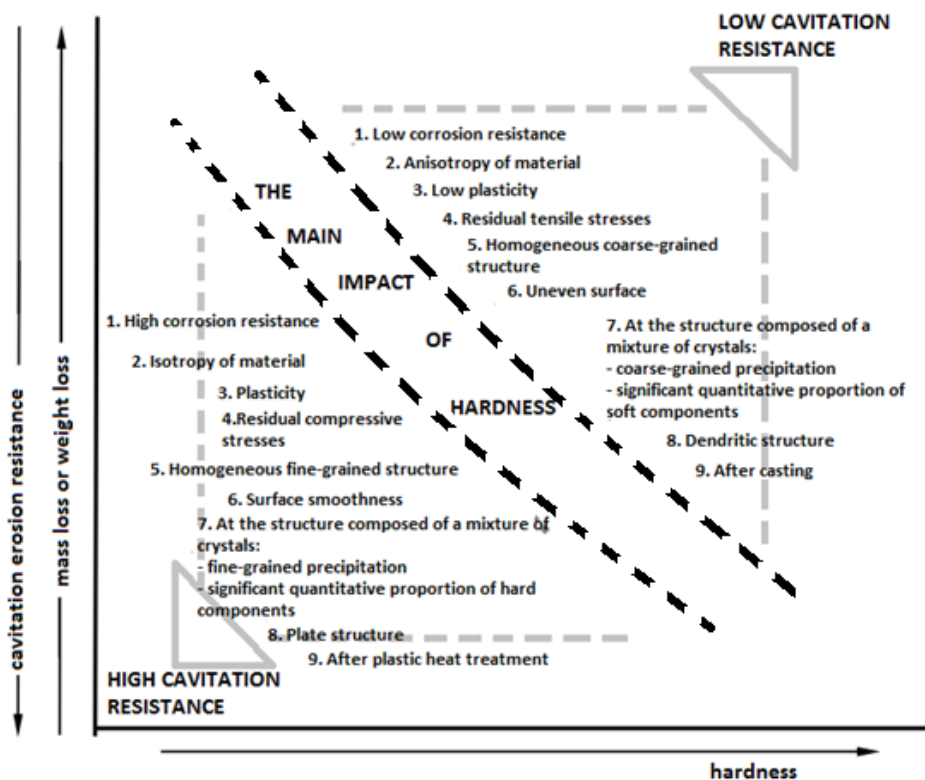


Fig. 2. Relation between cavitation erosion resistance mass or weight loss as function of hardness for different material properties [11]

## HARDNESS

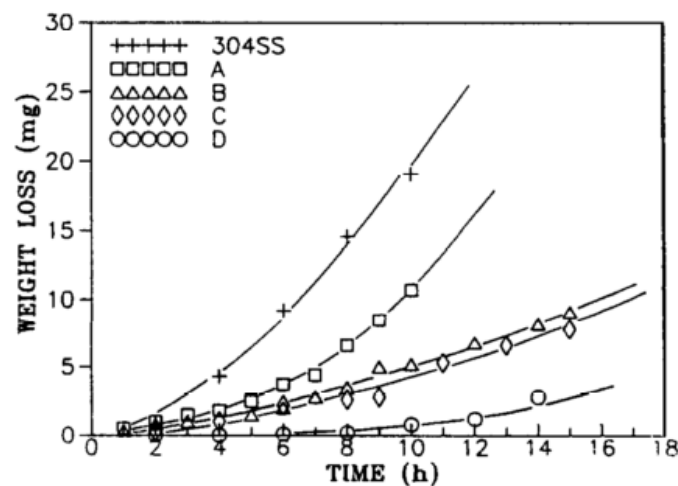
Hardness is the most often mentioned property, which has an influence on the erosion resistance [1][4]. Hardness is described as the resistance of material to withstand indentation. This property is related with the other material properties e.g. Young's modulus, and also plasticity and strength. Heymann noted the existence of an exponential correlation between hardness (HV) and the normalized resistance to cavitation erosion  $N_e$  defined as the ratio of the tested material's resistance to the resistance of the reference material tested at similar test parameters in a following form [3]:

$$N_e \sim HV^n \quad (1)$$

Eq. 1 works for various groups of materials: stainless steels, cast steel, bronzes, brass and aluminum alloys. In addition, Heymann [3] obtained that the normalized erosion resistance ( $N_e$ ) increases with the hardness of the material in exponent of  $n = \frac{5}{2}$ . His correlation between hardness and cavitation erosion resistance (Eq. 1) has been used in many works [9], [12]–[14].

Richman and McNaughton [15] analyzed results of Knapp's investigation also paying attention that the best correlation is the logarithmic correlation between hardness and MDP (mean depth penetration) with the correlation coefficient equals 0.813.

Investigations of four Fe-Mn-Al alloys and 304 stainless steel [16] in distilled water confirmed an increase the cavitation erosion resistance with material's hardness (Fig. 3).



**Fig. 3.** The cumulative weight loss of different Fe-Mn-Al alloys (in order of decreasing hardness and strength: C, D, B, A) after heat treatment and 304 stainless steel in distilled water [16]

However, tests performed in 3.5% NaCl solution showed that besides hardness also ability to passivation is beneficial for the cavitation erosion resistance. Kwok et al. [17] investigated an effect of heat treatment of  $\alpha+\beta$  brass and S30400 steel on the materials' properties and the cavitation erosion resistance. They obtained an exponential correlation between the cavitation erosion resistance ( $CER$ ) and following property: hardness, tensile strength, engineering strain energy and even grain size for both materials. They found that the correlations have the following form [17]:

$$CER = k(X)^a, \quad (2)$$

where  $k$  and  $a$  are constants different for each material property, including grain size,  $X$  is hardness, tensile strength and engineering strain energy.

Similar correlation of  $CER$  with hardness have been found by Tzanakis et al. [18]. According to Ref. [18], the normalized cavitation erosion resistance calculated as a ration of the cavitation erosion resistance ( $CER$ ) to the maximum cavitation erosion resistance ( $CER_{max}$ ) can be expressed in a following form:

$$CER/CER_{max} = a(H)^n \quad (3)$$

with a determination coefficient  $R^2 = 0.79$ .

Hattori et al. [19] basing on experimental data of 15 types carbon steels tested at various test methods have found a direct impact of hardness on the cavitation erosion resistance. The correlation is described by equation:

$$(N_e)_X = 2,1Exp(-6) \times HV^{2,4} \quad (4)$$

where  $N_e$  is normalized erosion resistance of arbitrary material  $X$ .

Statistical analysis of a database of 13 kinds of stainless steels with various structure (austenite, martensite, ferrite and duplex) tested at vibratory devices allows Hattori and Ishikura [20] to correlate  $CER$  with hardness by equation:

$$CER = 2,6Exp(-7) \times (HV \times F_{mat})^{2,4} \quad (5)$$

Where  $F_{mat}$  is material factor equals a ratio of hardness of eroded surface to that of original surface.

Despite many studies showing the positive effect of an increase in material hardness on an increase in cavitation resistance, a comparison of a wide range of materials, including austenitic steels, carbon steels, intermetallic alloys and cemented carbides, does not confirm the existence of such universal correlation [9][20][21]. The mentioned correlation is observed only within individual groups of materials, e.g. low-carbon steels, aluminium alloys and titanium alloys [22].

### TENSILE STRENGTH

Besides hardness, also tensile strength ( $\sigma_y$ ) and ultimate tensile stress ( $\sigma_u$ ) are used to correlate with the cavitation erosion resistance. Investigations of the cavitation erosion resistance of copper and nickel alloys allow to propose the correlation between ultimate tensile strength and Maximal Depth Penetration Rate (MDPR) for by equation [23]:

$$(MDPR)^{-1} = k(\sigma_u)^{2,3} \quad (6)$$

where  $k$  is constant.

Garcia and Hammitt [24] offered the correlation for several groups of materials (aluminum alloys, stainless steels, cooper alloys, nickel alloys) describing MDPR by equation:

$$MDPR = -1,570 + 6,48 \times 10^2 (\sigma)^{-1/2} \quad (7)$$

Hattori and Ishikura [20] confirmed an impact of tensile strength on the cavitation erosion resistance of steels (Fig. 4). The increase of the cavitation resistance together with the increase of tensile strength was also confirmed in Refs [14], [25]–[27]. For this reason, methods that improve ultimate tensile strength are considered beneficial for improving *CER*. Friction stir processed improves hardness and also ultimate tensile strength, which increase *CER* [28]. However, as emphasized in Ref. [27], tensile strength not alone, but together with other material properties, including fatigue strength, stacked energy consumption (SFE) determine *CER*.

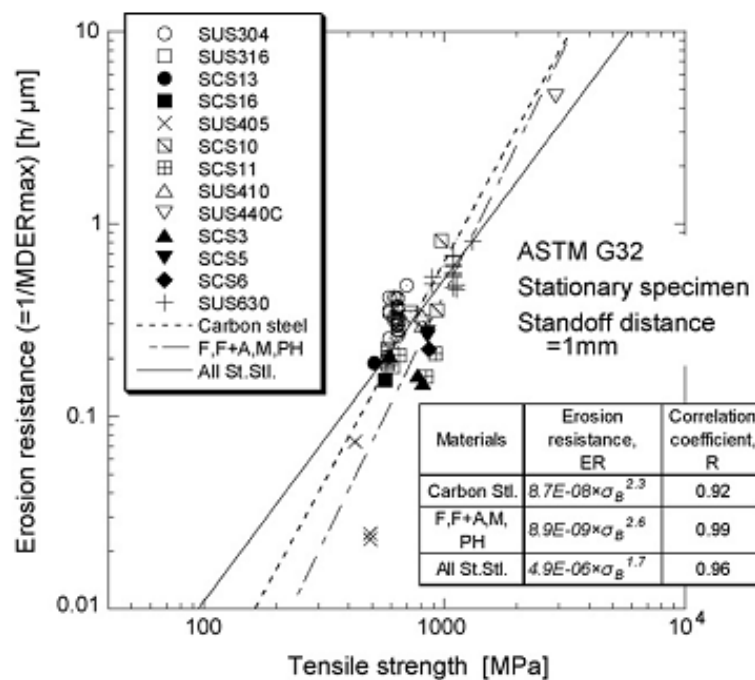


Fig. 4. Relation between tensile strength and erosion resistance for steels [20]

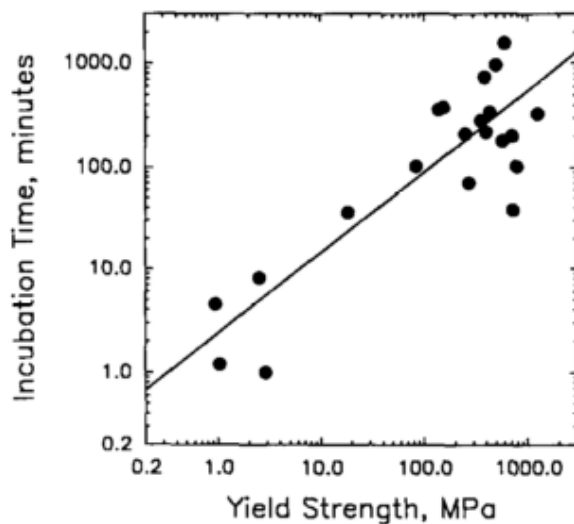
Investigation of Cr-Co-Ni-Mn austenitic stainless steels [14] proved that an increase of tensile strength as well as yield strength increase the incubation period and *CER*. In Ref. [29] was shown that high yield strength connected with high toughness and ductility improve *CER* of high-nitrogen austenitic stainless steel (HNAS) in comparison to two CrNiMo stainless steels even 3.3 times.

On the other hand, Sreedhar et al [30] described inversely proportional influence of yield strength to *CER* for commercial aluminum alloys.

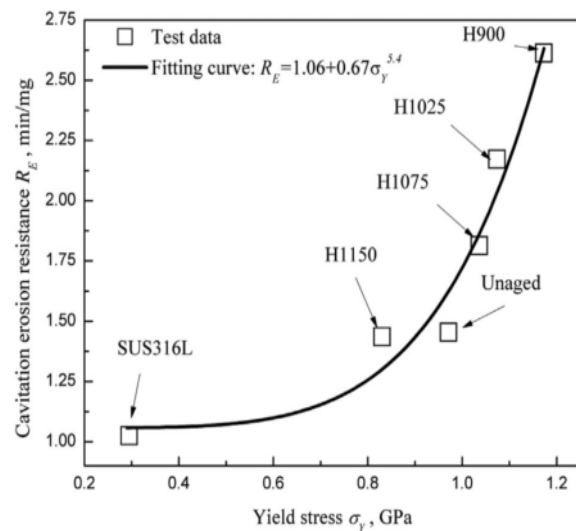
High yield strength affects the longer incubation time and better cavitation erosion resistance in steel Cr-Co-Ni-Mn compared to 304 stainless steel. High yield strength also has a beneficial effect on *CER* of titanium alloy and TiNiNb alloy [14]. Bregliozi et al. [25] noticed that an increase of yield strength as well as hardness and tensile strength influence the increase of *CER* of 304 austenitic stainless steel. Feller and Kharrazi [12] comparing an effect of tensile strength on incubation time for pure metals and their alloys, noticed that yield strength of material is not the dominant factor to describe its *CER*. Similar relationship was obtained by Thiruvengadam [13] who investigated different group of materials. However,

Richman and McNaughton [15] noticed that it is possible to correlate yield strength with *CER*, but it is important to remember that yield strength is roughly correlated with incubation time (Fig. 5).

The influence of yield strength on *CER* is confirmed also in Ref [31] (Fig. 6). On the other side, high entropy alloys [32] with low yield strength have better *CER* than 316L stainless steel.



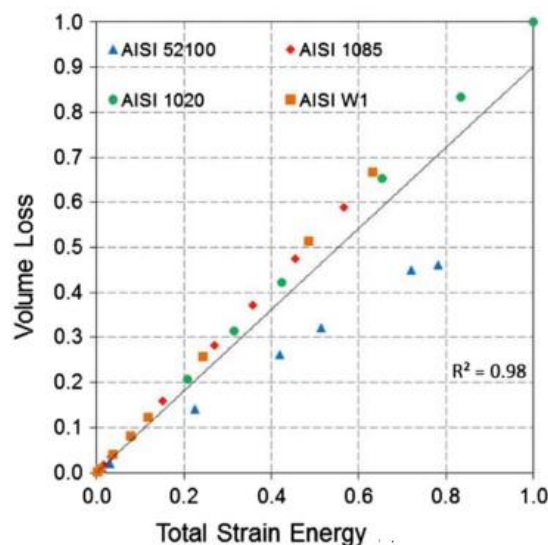
**Fig. 5.** Yield strength and incubation time correlation in vibratory apparatus in Feller and Kharrazi investigation [15]



**Fig. 6.** Correlation between cavitation erosion resistance and yield stress [25]

### STRAIN ENERGY

Strain energy ( $U$ ) is the energy stored in the material due to deformation caused by external forces. The accumulated energy caused by plastic deformation is called plastic strain energy or in case of elastic deformation – elastic strain energy [23].



**Fig. 7.** Relationship between total strain energy and volume loss for steel used in Ref. [18]

Thiruvengadam and Waring [13] proved that strain energy plays a significant role in *CER*. Comparing estimated strain energy ( $U_e$ ) and true strain energy ( $U_e'$ ) of various metals on the resistance to cavitation erosion, the linear relationship between volume loss rate and strain energy was found [13]. This relationship was better than for other mechanical properties.

Mann [33] reported that boronizing 13Cr–4Ni steel resulted high hardness and low elongations, and finally low *CER* in comparison to in as-received state or boronizing and tempering state of this steel. The best *CER* had this steel in as-received state due to the highest strain-energy. Similarly, excellent correlation of *CER* of three commercial steel grades (AISI 52100, AISI 1020, AISI 1085) with their strain energy has been found in Ref. [18] (Fig. 7). However, in Ref. [34] has been pointed that *CER* is better described by a combination of several properties, e.g. of ultimate tensile strength and elastic modulus or ultimate resilience and hardness, absorption energy and fatigue strength than the only one property.

### FATIGUE STRENGTH

Besides hardness, a property that has a dominant effect on *CER* is fatigue strength  $\sigma_f$  [15], [35]. This was affirmed by Richman and McNaughton [15] comparing and analyzing the results obtained by Feller and Kharrazi [12] and Knapp [36]. They noticed that fatigue strength is responsible for the material's resistance to the cavitation erosion phenomenon in 96%. The exponential relationship of fatigue strength and mean depth penetration (a) as well as the relationship of fatigue strength and maximum erosion rate (b) is shown in Fig. 8.

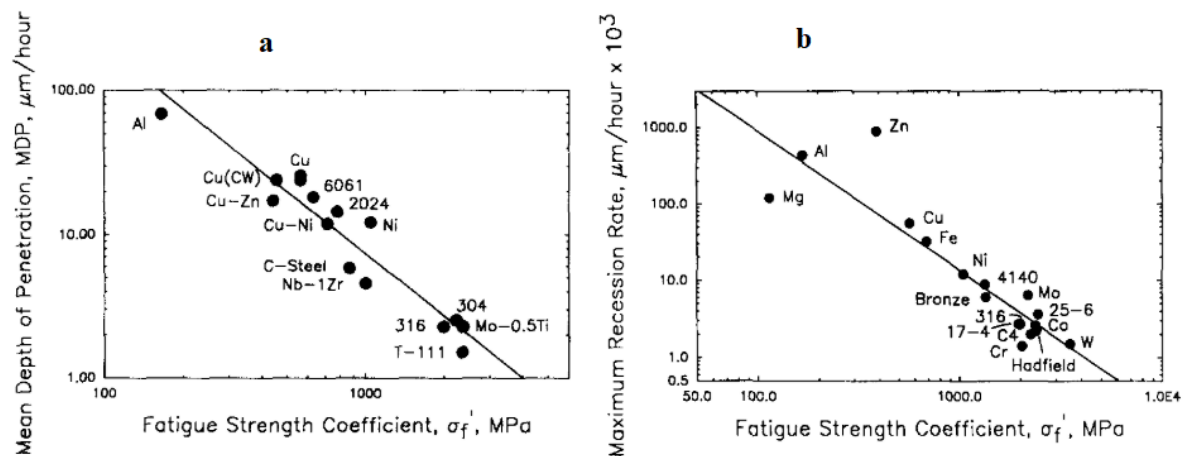


Fig. 8. Correlation between fatigue strength coefficient and mean depth penetration (a), maximum recession rate (b)[15]

Also in Ref. [37], there was confirmed significant impact of fatigue strength to *CER* and showed beneficent effect of explosive cladding of NiTi, which have high fatigue strength coefficient combined with a high cyclic strain-hardening exponent, on *CER*.

Będkowski et al. [35] performing fatigue and cavitation erosion tests have proved a correlation between *CER* of the steels (10HNAP, 18G2A, 15G2ANb) and their fatigue strength determined under uniaxial random tension–compression by following equation:

$$\log \sigma_{RMS} = 1,534 + 0,425 \log p_1 \quad (8)$$

where:  $\sigma_{RMS}$  - root-mean-square value of stress,  $p_1$  - pressure in the nozzle inlet in cavitation jet made by Lichtarowicz.



Chen et al. [38] proposed two parameters  $T$  and  $N$  for predicting  $CER$  based on an assessment that cavitation erosion is the result of strain fatigue failure. The parameter  $T$  (the uniform deformation strengthening modulus of material, [MPa]) and the parameter  $N$  (considered as the impact load property, [J/m]) are described by the formulas [38]:

$$T = \frac{\sigma - \sigma_{0.2}}{\varepsilon_f} \quad (9)$$

$$N = \frac{E_i}{S_i} + \frac{E_p}{S_p} \quad (10)$$

where:  $\varepsilon_f$  - true strain obtained in static tensile test,  $\varepsilon_f = \ln(1 - \Psi)$ ,  $\Psi$  - reduction of cross-sectional area,  $\sigma$  - tensile strength,  $\sigma_{0.2}$  - yield strength,  $E_i$  - crack formation work,  $E$  - total energy (work) that the specimens absorbed during fracture,  $S$  - the maximum offset at the point of fracture,  $E_p$  - crack extension work,  $S_i$ ,  $S_p$  - are the respective offset corresponding to  $E_i$  and  $E_p$ .

Based on the above mentioned parameters, the average weight loss rate of steel ( $W_c$  [mg/h]) was proposed to calculate from a following formula [38]:

$$W_c = \alpha \frac{v^2}{\sqrt{NT}}, \quad (11)$$

Where:  $v$  is water velocity [m/s],  $\alpha$  is a material constant.

Using the above formula, a comparative analysis with experimental data for four steel (16CrNi4Mo, Q235, M50NiL, 0Cr13Ni4Mo) was conducted (Fig. 8).

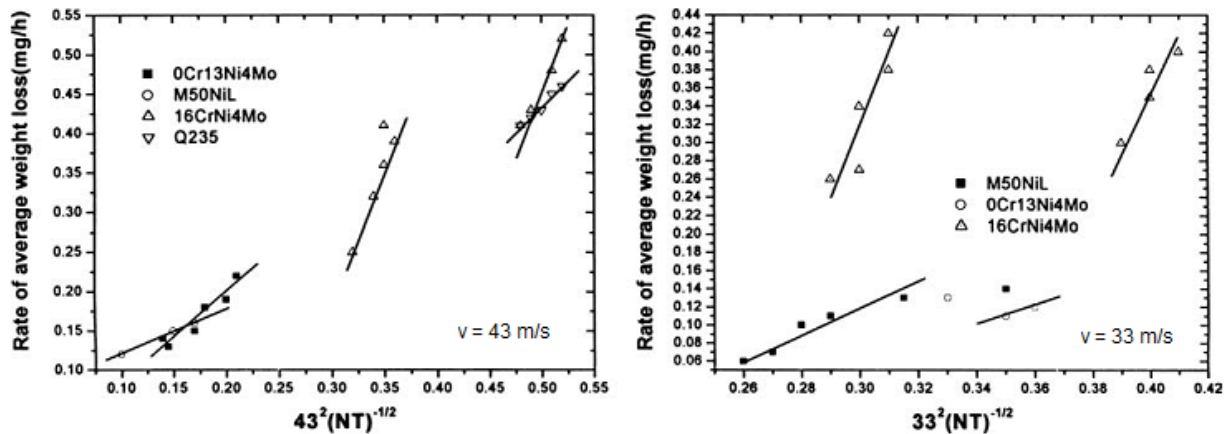


Fig. 8. Correlation of weight loss rate with  $W_c$  parameter [38]

The observed differences in Fig. 8. were associated with the  $\alpha$  parameter, which is related to other mechanical properties. It was also noticed an increase of  $CER$  with a simultaneous decrease in the value of  $\alpha$  parameter.

The fatigue strength determine also  $CER$  of plastics. Hattori and Takamoto [39] showed a linear relationship between  $CER$  and the impact energy and the fatigue strength. In addition, fatigue approach (the Palmgreen–Miner rule and the Paris- Erdogan law) have been also used for predicting mass or volume loss in cavitation erosion tests [40], [41].

## STRUCTURE

In addition to mechanical properties, *CER* depends on the structure of the material. According to Ref. [25], *CER* is related to the grain size. AISI 304 stainless steel and HN steel exhibited an increase in mechanical properties (hardness, tensile strength, yield strength) along with the decreasing grain size. This contributed to increase *CER*. Furthermore, fine grain steel characterized better corrosion resistance compared with medium and coarse grain steel. In Ref. [42], it is showed that in standard polycrystalline metals *CER* increases with decreasing grain size. Better mechanical properties due to fine grain for steel also confirmed in Ref. [32], however, coarse grain size in high entropy alloys or low nickel austenitic stainless steel [42] results in high ductility. Small grain sizes ensure good *CER* also for applying coatings [43], even 3 to 6 time better [44] than pure steel.

Another method to enhance *CER* of the steel AISI 316L was friction stir processing (FSP) [32]. Cavitation erosion tests made according to ASTM G-32 have shown that mass loss of 316L steel after FSP was significantly lower than for unprocessed steel. Moreover, the incubation time was even ten times longer after friction stir processing. Also, surface microhardness of the FSP samples was higher approximately two times than base material. This confirms that, grain size influences surface hardness and *CER* which is in accordance with previous mentioned impact of hardness on *CER*. Microscopic studies revealed that 316L steel exposed to the lowest rotation speed during FSP process exhibits more homogenous microstructure after CE tests than 316L steel exposed to high rotation speed during FSP and base material (large and deep erosion craters). The common feature of all microstructures gained for 316L steel exposed to cavitation is a surface morphology typical of ductile fracture. As highlighted in Ref. [32], fine-grained steel during cavitation impact indicates less damage than coarse-grained steel, as a result of higher work hardening rate due to micro-jet impingements.

Also it has been reported in the literature that usually, ferritic stainless steel has lower *CER* than austenitic and martensitic stainless steel [45]. The cavitation resistance of materials beyond microstructure is also influenced by e.g., roughness [46], composition and phase transformation [47].

The simplest method of improving *CER* is to increase content or to add component that affects the change of material properties. This method was applied in Ref. [48], in which is described a reinforcement *CER* of steel 316L by adding molybdenum. Samples were made by mixing and melting mixture of pure (99,9% wt) components desirable steel and hot rolling into plates. Using this method allowed to obtain samples with Mo content 2,5%, 4%, 6% and 8%. Cavitation erosion tests proved that with increasing Mo content decreases mass loss and mean depth of erosion (MDE). Microstructure analysis of all samples after cavitation erosion tests shown that, with the higher molybdenum content, the roughness decreased. In addition, the surface of the sample with the highest Mo content showed slight damage. This confirms that *CER* can also be elevated by addition to steel molybdenum. Allenstein et al [49] improved about 5-6 times cavitation erosion resistance Ca-6NM martensitic stainless steel by low-temperature plasma nitride in comparison to non-nitrided steel Ca-6NM. They also noticed that longer nitriding time translates to higher cavitation erosion resistance.

## COATINGS

Undoubted effect of hardness on *CER* contributed to searching for methods of increasing material hardness or surface hardness. The most popular is thermal or thermochemical treatment, especially nitriding [31], [49], [50], which increase hardness and also improves



other mechanical properties. The other way for increasing surface hardness is deposition of hard coatings (e.g. WC-Co [51]) by means of various methods e.g. High Velocity Oxy-Fuel Thermal Spray (HVOF) [52], [53], PVD [54], [55], [56]), Vacuum Plasma Spraying (VPS) [57], Atmospheric Plasma Spraying (APS) [58], arc spraying (ARC) [44], or Plasma Assisted Physical Vapour Deposition (PAPVD) [59]. Friction stir process (FSP) [28] is, as mentioned, another surface treatment used to reduce grain size and improve *CER*. Application of the coating allows not only to obtain the appropriate surface hardness, but also other mechanical properties that may effect on *CER*, e.g. high wear resistance, which can be beneficial in situations where surface wear or erosion can occur [51]. Kim and Lee [60] proved strongly dependence between hardness and *CER* of coatings produced by arc-spraying. However, as emphasized in Ref. [61], along with increasing hardness increases brittleness, which has an effect on decreasing *CER*. The active screen low temperature plasma nitriding increased both hardness and *CER* of martensitic stainless steel AISI 410 [10]. Mass loss was approx. 28 time less than this steel without mentioned treatment. Comparison two surface treatments - PAPVD and plasma nitriding proved that a combination of plasma nitriding followed by deposition of  $Cr_{1-x}N_x$  coating using PAPVD method is a better solution for increasing *CER* than just a single surface treatment [59].

As was mentioned, deposition of coatings is used to improve *CER*. HVOF coatings belong to coatings that increase *CER* [62]. Taillon et al. [63] used HVOF method to deposit  $Fe_3Al$  and  $Fe_3Al$  reinforced with nitride and boride phases coatings from pure powders. Their results were compared with coatings made from commercial powders e.g. WC-CoCr and  $Cr_2C_2-NiCr$ . As a substrate for HVOF coatings was used steel AISI 444. The researchers also investigated low carbon martensitic stainless steel, 13Cr-8Ni (S13800), 13Cr-4Ni (S41500), 15Cr-5Ni (S15500) and ferritic stainless steel AISI 444. All coatings exhibit lower erosion rates than bare steels, but no correlation of *CER* with hardness, young modulus or surface properties has been found. Microstructure analysis of eroded surfaces showed that cracks nucleation starts in the matrix for composite coatings and in grain boundaries of ferritic steel. The investigated coatings exhibit comparable erosion rates to commercial powders. Wu et al. [62] proved that WC-Co-Cr coating sprayed by HVOF method has 64% lower mass loss compared to stainless steel 1Cr18Ni9Ti.

The atmospheric plasma spraying (APS) [64] allows to create low porous and hard YSZ-NiCrBSi composite coatings, which also strengthen the surface of the material for cavitation erosion. Better *CER* is a result of denser composite microstructures, although cavitation damage can occur on YSZ-NiCrBSi splats interfaces, which may also influence on higher mass loss rates during cavitation exposure. The APS method also allows creating the CoMoCrSi coatings [34] in which we can observe hard precipitates (cobalt-molybdenum-silicon based intermetallic Laves phases) dispersed in a softer cobalt-based alloy matrix. This coating has also high cavitation erosion resistance, but it is higher after heat treatment and compared with HVOF coatings is lower.

High density and low porosity coating, with high adhesion to the substrate is also possible by arc spaying [46]. However, in the case of the applied coating, it was noticed that it was necessary to grind the surface, which affected the increasing *CER*. Stella et al [57] deposited NiTi coating using vacuum plasma spraying (VPS) on stainless steel 316L. For comparison was used pre-alloyed Ni-Ti powder prepared by the manufacturer by gas atomization (c1), and powder of Ni powder and Ti powder blended in tubular mixer (c2). Cavitation erosion tests showed pronounced bigger mass loss for coating c2 than coating c1. Also bulk material did not exhibit any mass loss in 30 hours of tests, after this time (to the end of tests) bulk material exhibit less than 4 mg, while for the coating c2 total mass loss was 12.8 mg and for the coating c1 was 5.6 mg, which was explained by low adherence particles in the

pores. Moreover, a high value of surface porosity was noted for tested coatings. EDX analysis revealed presence of additional  $\text{Ni}_3\text{Ti}$  and  $\text{Ni}_4\text{Ti}_3$  phases in c2 coating, and  $\text{NiTi}$  phase in c1 coating. Mean depth of erosion (MDER) had the lowest value for c2 coating. Analysis of coatings microstructure during the cavitation erosion tests were also performed. After 22 hours of cavitation tests, not all damages in c1 coating can correspond initial defects. According to Stella et al [57], the  $\text{NiTi}$  grains could be affected by stress generation or crack propagation. Thorough analysis of microstructure showed undeformed grains and intergranular cracks on grains boundaries in c1 coating. Microstructure of the coating c2 showed damage of the  $\text{Ni}_3\text{Ti}$  phase mostly located along phase boundaries, what was explained of the accumulated stresses between two phases characterized by various deformations.

Comparison of the cavitation erosion resistance of WC-10Co4Cr coatings, Co-based coatings, WC-10Co4Cr/Co-based composite coatings, and Fe-based amorphous / nanocrystalline coatings produced by HVOF method on 316L steel in artificial sea water and deionized water showed the highest *CER* of WC-10Co4Cr coatings while maintaining high corrosion resistance [65]. In addition, the coating with the highest *CER* had the highest surface hardness (1273 HV), which confirms earlier considerations regarding the high association of surface hardness with *CER*.

In Ref. [66], there was highlighted an influence of HVOF spraying parameters on composition, microstructure and mechanical properties of Co-based coatings, and their effects on *CER* of 316L stainless steel substrate. It has been shown that the higher the flame temperature during coating deposition, the hardness and toughness of the coatings were improved and decreased the roughness of surface.

According to Ref. [67], deposition of ZrC nanoceramic coating by the double glow discharge sputter technique on steel 316 also improved *CER* and increased corrosion resistance in comparison to uncoated steel.

Another example of an effect of coatings deposition on *CER* is shown in Ref. [68], in which 8 wt% yttria-stabilized zirconia (8YSZ) and 8YSZ-ER coatings deposited on 316 steel were investigated. In addition, NiCrAlY was introduced into the micro defects of the as-sprayed 8YSZ coating by vacuum impregnation. The addition of NiCrAlY to the 8YSZ coating enabled to improved cohesion strength, toughness, and hardness of the coating. This increased the *CER* three times compared to the 8YSZ coating. According to [68], the coating absorbs the shock wave resulting from bubble collapse, using it for recrystallization, which increased *CER*.

The magnetron sputtered AlTiN and TiAlN coatings deposited on SS304 stainless steel [67] showed better AlTiN thin film adhesion to the steel substrate than TiAlN thin film, which was correlated with surface hardness. The results obtained highlight the important impact of structural and mechanical properties (hardness, adhesion, and elastic modulus) on *CER*. With an increase in adhesion of the coating increased *CER*. *CER* of AlTiN coating was three times higher comparing with TiAlN coating, and ten times compared with uncoated steel. In addition, sliding wear of both coatings was 24 times lower than uncoated steel, and deposition of coating caused a decrease of the friction coefficient two times. Similar results of the role of adhesion of PVD coatings on *CER* were obtained in Refs [55], [56] and [69].

Rudolf et al. [70] compared WCCoCr,  $\text{Cr}_3\text{C}_2+25\text{NiC}$ , WC+CrC+Ni coatings with and without impregnation deposited on 13Cr4Ni steel with uncoated steel substrate using at two cavitation rigs: cavitating orifice and cavitating jet. The results showed that the tested coatings have very low *CER*. *CER* of the coatings were lower than that of uncoated steel.

The cited examples and other works [71-73] indicate that an important aspect in the case of coatings is not only the mechanical properties of the substrate or the coating itself, but also

the properties of the whole coating-substrate system, including structural properties [69]. Particular attention should also be paid to the coating application parameters and the application method.

## SUMMARY

In the paper correlations of selected mechanical properties (hardness, yield strength, tensile strength, strain energy and fatigue strength) with resistance to cavitation erosion have been presented. The influence of grain size on cavitation resistance of material and the possibilities of increasing cavitation erosion resistance is also shown. The exponential correlation between hardness, ultimate tensile stress, yield strength and *CER* is shown. However, with increasing hardness increases brittleness and brittle materials have low *CER*. The other property, which highly affects *CER*, is fatigue strength. The relationship between fatigue strength and mean depth penetration as well as maximum erosion rate has been proved several times. Fatigue approach has been used for predicting mass or volume loss of bulk steels exposed to cavitation.

Besides mechanical and endurance properties, structural properties, especially the grain size and the type of steel (austenitic, martensitic, ferritic), are also important for *CER*. The smaller the grain size, the higher the *CER*. Modification of surface layer by applying coatings or heat treatment allows increasing *CER*.

Deposition of hard coatings increase *CER* in most cases. However, the correlation between coating properties and *CER* of the coating-substrate system is not simple. The mechanical properties of the coatings depend on deposition parameters. Besides hardness, adhesion is the other key property highly affects *CER*. In case of HVOF coatings, the grain size of the powder used for the coating is also an equally important aspect for *CER*.

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