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Dielectric Spectroscopy Studies and Modelling of Piezoelectric Properties of Multiferroic Ceramics

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Abstract: Compounds and solid solutions of bismuth ferrite (BiFeO₃)—barium titanate (BaTiO₃) system are of great scientific and engineering interest as multiferroic and potential high-temperature lead-free piezoelectric materials. In the present paper, the results of research on the synthesis and characterisation of 0.67Bi_{1-x}FeO₃–0.33BaTiO₃ (67BFBT) ceramics in terms of crystal structure and dielectric and piezoelectric properties are reported. It was found that the produced 67BFBT ceramics were characterised by a tetragonal crystal structure described by the *P4mm* space group, an average crystallite size $\langle D \rangle \approx 80$ nm, and an average strain $\langle \epsilon \rangle = 0.01\%$. Broad-band dielectric spectroscopy (BBDS) was employed to characterise the dielectric response of polycrystalline ceramics. The frequency range from $\nu = 10^{-1}$ Hz to $\nu = 10^5$ Hz was used to characterise the influence of the electric field strength on dielectric response of the ceramic sample at room temperature. The dielectric spectra were checked for consistency with the Kramers–Kronig test, and the high quality of the measurements were confirmed. The electric equivalent circuit method was used to fit the dielectric spectra within the frequency range that corresponded to the occurrence of the resonant spectra of the radial mode for thin disk sample, i.e., from $\nu = 10^5$ Hz to $\nu = 10^7$ Hz and the temperature range from $T = -20$ °C to $T = 50$ °C. The electric equivalent circuit [R_sCPE₁([L₁R₁C₁]C₀)] was used, and good fitting quality was reached. The relevant calculations were performed, and it was found that the piezoelectric charge coefficient exhibited a value of $d_{31} = 35$ pC/N and the planar coupling factor was $k_p = 31\%$ at room temperature. Analysis of impedance spectra performed in terms of circumferential magnetic field made it possible to establish an influence of magnetic field on piezoelectric parameters of 67BFBT multiferroic ceramics. Additionally, the “magnetic” tunability of the modulus of the complex dielectric permittivity makes 67BFBT a sensing material with vast potential.

Keywords: electroceramics; structure; impedance spectroscopy; piezoelectric properties; modelling; “magnetic” tunability



Citation: Lisińska-Czekaj, A.; Czekaj, D.; Garbarz-Glos, B.; Bąk, W.; Zate, T.T.; Jeon, J.-H. Dielectric Spectroscopy Studies and Modelling of Piezoelectric Properties of Multiferroic Ceramics. *Appl. Sci.* **2023**, *13*, 7193. <https://doi.org/10.3390/app13127193>

Academic Editor: Theodore E. Matikas

Received: 17 April 2023

Revised: 9 June 2023

Accepted: 12 June 2023

Published: 16 June 2023



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1. Introduction

From a scientific point of view, multiferroic materials are a somewhat mysterious yet, at the same time, extremely interesting class of materials. Thanks to the unique properties of these materials, which are characteristic of both ferroelectric and ferromagnetic materials, they have a wide range of potential applications. Multiferroic materials can open new

horizons in the design of modern devices, especially those whose operation is based on controlling the electrical properties of materials (and thus influencing the parameters of the electrical circuit) by changing the magnetic field and vice versa, i.e., controlling the magnetic properties of materials by changing electric field strength [1].

Let us consider the terminal elements of bismuth ferrite (BiFeO_3)—barium titanate (BaTiO_3) system. It is well known that BiFeO_3 has an ABO_3 -type perovskite structure and is one of the few materials characterised by the coexistence of ferroelectric and (anti)ferromagnetic properties at room temperature. Because of this essential feature of multiferroics, BiFeO_3 (BFO) is considered the most suitable among all available multiferroic materials for the development of practical spintronic devices and related technologies [2]. BaTiO_3 (BT) belongs to the corner-sharing oxygen octahedral material family (structure), which was discovered during World War II. Since then, it has remained one of the most widely used and systematically studied non-linear dielectric materials and is considered the prototype for ferroelectric ceramics as well as being the first piezoelectric ceramics transducer ever developed [3]. Since the discovery of BT in the early 1940s, it has been continually used in new industrial and commercial applications. Its unique physical characteristics, e.g., values of piezoelectric ($d_{33} \sim 190$ pC/N) and dielectric ($\epsilon_r \sim 1700$ and low dielectric losses at room temperature) parameters, enable the use of BT ceramics for capacitors, multilayer capacitors, ultrasonic transducers, pyroelectric infrared sensors, positive temperature coefficient resistors, and energy storage devices [4,5].

However, neither BFO nor BT are actually used in their pure chemical form. To optimise properties for specific applications, they are often combined with various additives that adjust and enhance their basic properties. They are also combined with each other to form either compounds or ceramic–ceramic composites, which have received a lot of scientific interest [6–8]. To justify the above-mentioned points, let us point out a bulk ternary BiFeO_3 – BaTiO_3 and PbTiO_3 system [9] that was fabricated and studied as a potential high-temperature piezoelectric ceramics transducer. In case of Tm-doped $0.7\text{Bi}_{1-x}\text{Tm}_x\text{FeO}_3$ – 0.3BaTiO_3 ($x = 0$ – 0.05) ceramics, it was found that Tm-induced structural modifications yielded improvements in the dielectric constant, remnant polarization, and remnant magnetization values [10]. On the other hand, studies of Mn-modified $(1-x)\text{BiFeO}_3$ – $x\text{BaTiO}_3$ ceramics showed that the addition of Mn improved dielectric losses and increased dc resistivity in bulk BiFeO_3 – BaTiO_3 , greatly enhancing the piezoelectric response; however, the poor stability of the poled domain structure caused rapid ageing and a large difference between low- and high-field piezoelectric charge coefficient d_{33} [11].

The main motivation to carry out the present research was the high potential for practical application that the materials of the BiFeO_3 – BaTiO_3 system have. In addition, the issues of the physics and technology of multiferroic materials are extremely rich, interesting, and non-trivial from a scientific point of view. Novel materials of rich functionality can be created by means of combining and properly modifying bismuth ferrite and barium titanate. Therefore, the chemical composition $0.67\text{Bi}_{1.02}\text{FeO}_3$ – 0.33BaTiO_3 (67BFBT) was chosen as the research material due to its potential use as an alternative to lead-based piezoelectric materials [8,11,12].

The aim of the present research was to synthesise $0.67\text{Bi}_{1.02}\text{FeO}_3$ – 0.33BaTiO_3 (67BFBT) multiferroic ceramics and characterise the produced ceramics in terms of crystal structure, surface morphology, impedance spectroscopy, and piezoelectric properties, including modelling with the electric equivalent circuit method and the use of a circumferential magnetic field to reveal the effect of a weak magnetic field on the piezoelectric and dielectric properties of 67BFBT ceramics.

2. Materials and Methods

Ceramics of $0.67\text{Bi}_{1.02}\text{FeO}_3$ – 0.33BaTiO_3 composition were synthesised by a conventional solid-state reaction process. Chemically pure Bi_2O_3 (Sigma Aldrich, Burlington, MA, USA, 99.9%), Fe_2O_3 (Sigma Aldrich, USA, 99%), TiO_2 (Sigma Aldrich, USA, 99.8%) metal oxides, and BaCO_3 (Sigma Aldrich, USA, 99%) were used as raw materials. To obtain a

homogeneous mixture, stoichiometric amounts of oxide and carbonate precursors were first carefully weighted. Then, they were mixed thoroughly by ball-milling in a polyethene bottle with yttria-stabilised ZrO₂ (YSZ) balls for 24 h in ethanol dispersion reagent. After that, they were calcined at 900 °C for 6 h in a box furnace. Later on, the calcined powder was ball-milled again for 24 h, and then the dried powder was slightly pressed into compacts in a stainless steel die into disks with a diameter of 10 mm using a uniaxial press, followed by cold isostatic pressing (CIP) at 200 MPa for 5 min. The green pellets were sintered at 970 °C for 3 h in a box furnace.

The relative density, crystal structure, and microstructure of the sintered sample were examined by using the Archimedes method, an X-ray diffractometer (XRD, D/Max-2500, Rigaku, Japan), and a scanning electron microscope (SEM, JEOL JSM-7900F), respectively. Two-dimensional grain sizes of the sintered 67BFBT ceramics were calculated by measuring more than 100 grains area in the SEM image using ImageJ—a public domain Java image processing and analysis programme [13,14]. The obtained values were then converted to three-dimensional grain sizes by multiplying by a factor of 1.68 [15].

Broad-band dielectric spectroscopy measurements (BBDS) [16] were carried out using a system consisting of a high performance frequency analyser (Alpha-AN) and a temperature control system (Quatro Cryosystem) via Novocontrol. It is well known (see [17,18]) that, by applying the suitable strategy and alternative methods of analysing the experimental impedance data obtained as a result of the BBDS measurement, it is possible to characterise the electrical and dielectric properties of the ceramic material.

Spectroscopic measurements were carried out on disk-shaped pieces that were 7.52 mm in diameter and 0.6 mm thick. To ensure a good electric contact, the sample ends were cleaned with a soft acid solution and silver electrodes (silver paste) were deposited. Before the measurement, the system was cooled down to −20 °C with liquid nitrogen. The measurements were performed during the heating cycle up to a maximum temperature of 50 °C. Dielectric spectra were recorded at programmed temperatures with a temperature step of 10 °C 15 min after the temperature had been stabilised. The measurement software allowed a frequency run of 40 points per decade in the measuring frequency range. WinDATA Novocontrol software was used for the recording, visualization, and processing of experimental data. However, to check the consistency of the experimental data, a computer programme created by Boukamp was used [17,19]. The Kramers–Kronig test was carried out for experimental data measured at room temperature within a frequency range from $\nu = 10^{-1}$ Hz to $\nu = 10^5$ Hz. The amplitude of the test voltage and related electric field strength were as follows: $U_{ac} = 0.1$ V_{rms} ($E_{ac} = 166$ V/m), $U_{ac} = 0.2$ V_{rms} ($E_{ac} = 333$ V/m), $U_{ac} = 1.0$ V_{rms}, ($E_{ac} = 1666$ V/m), and $U_{ac} = 2.0$ V_{rms} ($E_{ac} = 3333$ V/m) (root mean square—rms). Practical application of the impedance data validation method in the study of electroceramics was described by us in detail elsewhere (in [20] and the references cited therein). Experimental data recorded within a frequency range from $\nu = 10^5$ Hz to $\nu = 10^7$ Hz, which corresponded to the occurrence of radial mode resonance spectra for a sample in the form of a thin disk, were fitted to the electric equivalent circuit [R_s CPE₁([L_1 R₁C₁]C₀)]. Complex non-linear least squares method (CNLS) was employed for fitting [17,19].

Piezoelectric parameters were calculated according to resonance (anti-resonance method) [21,22]. For this method, an accurate impedance analyser is a crucial requirement for use in testing (a Novocontrol high-performance frequency analyser was used for the current study's experiments). The resonant frequencies (ν_r) and anti-resonant frequencies (ν_a) were detected based on spectroscopic characteristics measured in the course of BBDS experiments, and the free capacitance C^T at 1 kHz was measured [23].

Calculations were performed using the classical formulas for radial vibrations of a disk-shape sample [21–25].

3. Results and Discussion

3.1. Morphology Studies of $0.67\text{Bi}_{1.02}\text{FeO}_3\text{--}0.33\text{BaTiO}_3$ Ceramics

The morphology of $0.67\text{Bi}_{1.02}\text{FeO}_3\text{--}0.33\text{BaTiO}_3$ ceramics sintered at $970\text{ }^\circ\text{C}$ is shown in Figure 1a, whereas a histogram of the grain size distribution is shown in Figure 1b.

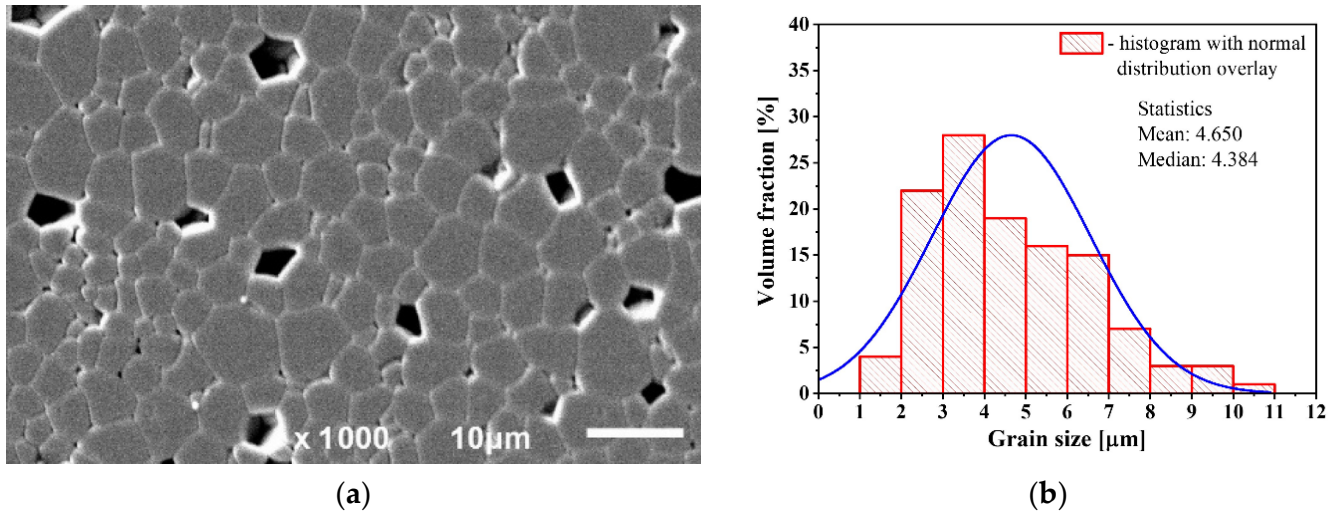


Figure 1. (a) Morphology of 67BFBT ceramics sintered at $970\text{ }^\circ\text{C}$; (b) grain size distribution.

One can see from Figure 1a that several polygonal voids are present in the cross-section of the ceramic sample subjected to the SEM investigation preparation procedure. It is worth noting that all the voids visible in the photo (Figure 1a) exhibit regular, mostly pentagonal shapes that resemble the shape of the grains of the ceramics under study. Therefore, it is reasonable to suppose that the voids were “man-made” and created during the mechanical treatment of the cross-section surface of the ceramic sample (i.e., sample preparation for SEM investigation). It is reasonable to suppose that the grains that initially took their sites were removed from their original positions by mechanical treatment.

The grain size distribution was measured with the help of an image processing and analysis programme (ImageJ [13,14]) on the basis of SEM photos of the ceramic sample (Figure 1a), and the results are given in Figure 1b. The mean and median values of the average grain size were approximately 4.7 and 4.4 μm, respectively.

3.2. X-ray Diffraction Studies

As an example, an X-ray diffraction pattern of 67BFBT ceramics sintered at $970\text{ }^\circ\text{C}$ is shown in Figure 2. The search-match process was performed using Match! software (Crystal Impact, Bonn, Germany) [26]. Phase analysis was performed, and it was found that the reference pattern of $\text{Ba}_{0.3}\text{Bi}_{0.7}\text{FeO}_{2.85}$ [27] matches all peaks shown within the measuring range 2Θ with Figure-of-Merit (FoM) parameter $FoM = 0.96$.

It is worth noting that the reference pattern (source of entry: Crystallography Open Database; COD ID 4341652) exhibited a tetragonal structure (space group $P4mm$) with the following unit cell parameters: $a = 3.9963\text{ \AA}$ and $c = 4.0032\text{ \AA}$. Based on Archimedes’ method and calculations of theoretical density, it was found that 67BFBT ceramics reached a relative density of 94.5%.

Line profile analysis was performed, and the resulting Williamson–Hall plot is shown in Figure 3.

One can see from the Williamson–Hall plot that the average size of the crystallites is $\langle D \rangle = 801\text{ \AA}$. The average strain ($\langle \epsilon \rangle$), which is a measure of micro-deformations, is $\langle \epsilon \rangle = 0.01\%$.

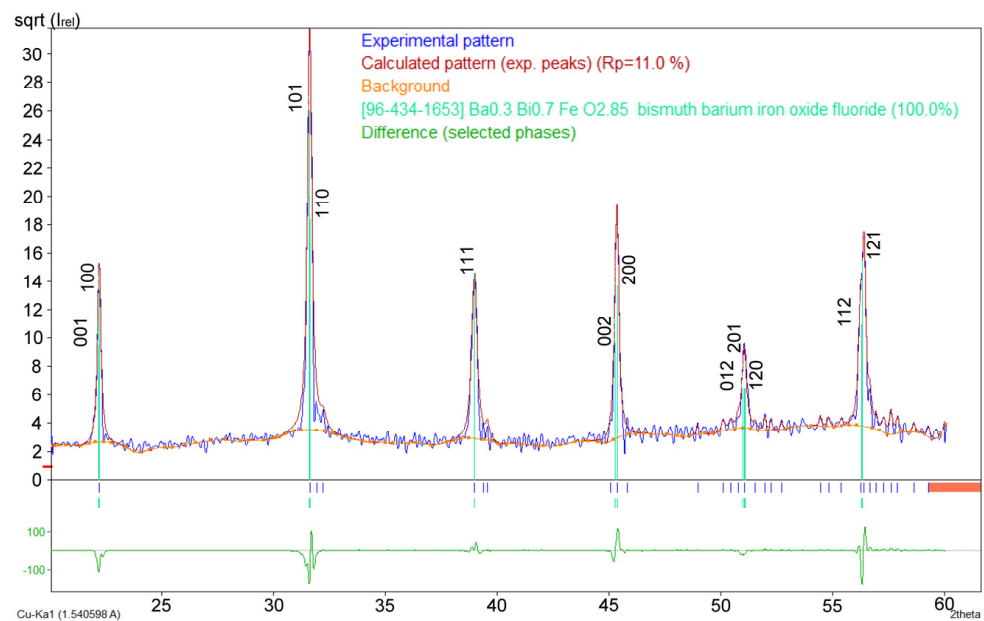


Figure 2. X-ray diffraction pattern of 67BFBT ceramics. $\text{Ba}_{0.3}\text{Bi}_{0.7}\text{FeO}_{2.85}$ was taken as a standard; COD ID:4341652.

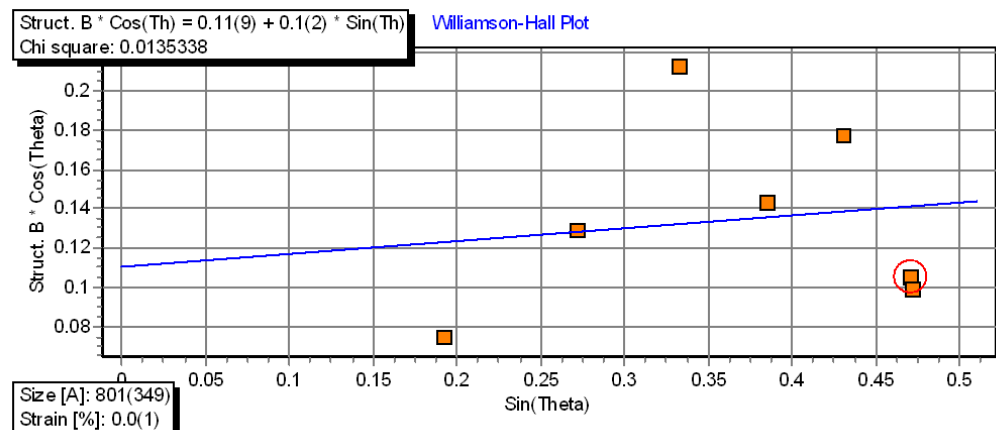


Figure 3. Williamson–Hall plot for 67BFBT ceramics calculated based on X-ray diffraction pattern.

3.3. Data Validation of Impedance Measurements

It is commonly known (e.g., see [16–19]) that impedance spectroscopy is extremely susceptible to random disturbances. Therefore, knowing the quality of the measured impedance data is extremely important to facilitate correct analysis. Kramers–Kronig (K-K) relations are very helpful for data validation [17,19]. The Kramers–Kronig rule states that the imaginary part of a dispersion is fully determined by the form of the real part of dispersion over the frequency range $\infty \geq \nu \geq 0$. Similarly, the real part of dispersion is determined by the form of the imaginary part of dispersion [17]. In the present study, an analysis based on the K-K relations was performed with the use of the computer programme created by Boukamp [17,19].

Kramers–Kronig transform test results of impedance data measured for 67BFBT ceramics at room temperature (RT) within the frequency range $\Delta\nu = (10^{-1}–10^5)$ Hz are shown in Figures 4a and 5a. A complex impedance diagram combined with K-K transform test results measured at a signal voltage of $U = 0.1 V_{\text{rms}}$ is shown in Figure 4b. An Impedance diagram with K-K transform test results measured at a signal voltage of $U = 2.0 V_{\text{rms}}$ is shown in Figure 5b. One can see in Figures 4a and 5a that the data recorded at RT exhibit a small deviation from K-K behaviour (residuals are less than 0.4%). Upon inspection of the

results given in Figures 4b and 5b, it can be determined that there is very good agreement between the measurements (blue circles) and K-K calculations (red crosses).

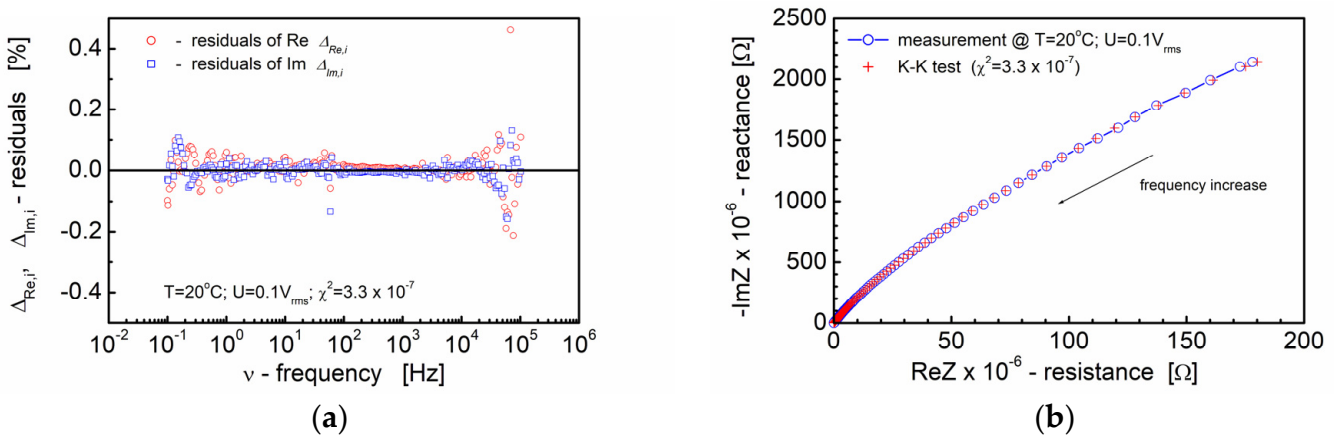


Figure 4. Results of the Kramers–Kronig validation test for admittance measurements of 67BFTO ceramics at RT and measuring signal voltage $U = 0.1 \text{ V}_{\text{rms}}$. Panel (a) shows the residuals plot; panel (b) shows the comparison of measured (blue circles) and calculated values (red crosses) for complex impedance plot ($-Z''$ vs. Z' ; free scale).

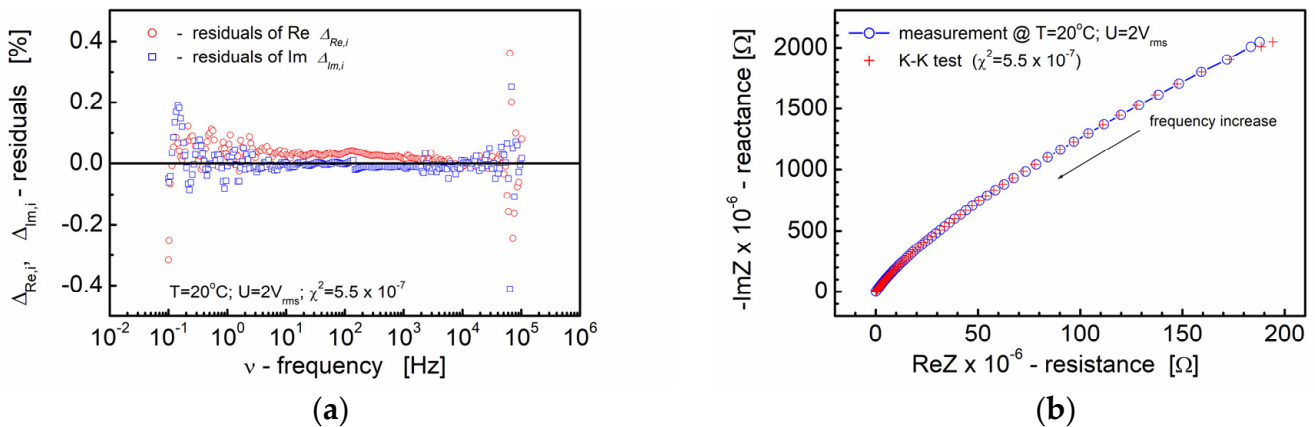


Figure 5. Results of the Kramers–Kronig validation test for dielectric measurements of 67BFTO ceramics at RT and measuring signal voltage $U = 2 \text{ V}_{\text{rms}}$. Panel (a) shows the residuals plot; panel (b) shows the comparison of measured and calculated values for complex impedance plot ($-Z''$ vs. Z' ; free scale).

The quality parameter “chi-squared” reached a value of $\chi^2 = 3.3 \times 10^{-7}$ – 5.5×10^{-7} for room temperature measurements taken at $U = 0.1 \text{ V}_{\text{rms}}$ ($E = 166 \text{ V/m}$) and $U = 2.0 \text{ V}_{\text{rms}}$ ($E = 3333 \text{ V/m}$), respectively. The above-mentioned results proved the high quality of the measurements and fully justified further analysis of the impedance data.

An alternative representation of the impedance data showing the influence increasing electric field intensity on the spectroscopic dependence of the reactance (reactance times pulsation: $-Z''\omega$) of the piezoelectric ceramic sample 67BFTO at room temperature is shown in Figure 6a. It can be seen that the spectroscopic plots shift towards higher frequencies as the electric field strength increases. Additionally, the dependence of the real part of complex impedance (Z') on the imaginary part of complex impedance times pulsation ($-Z''\omega$) (Figure 6b) shows the substantial dependence on the electric field strength.

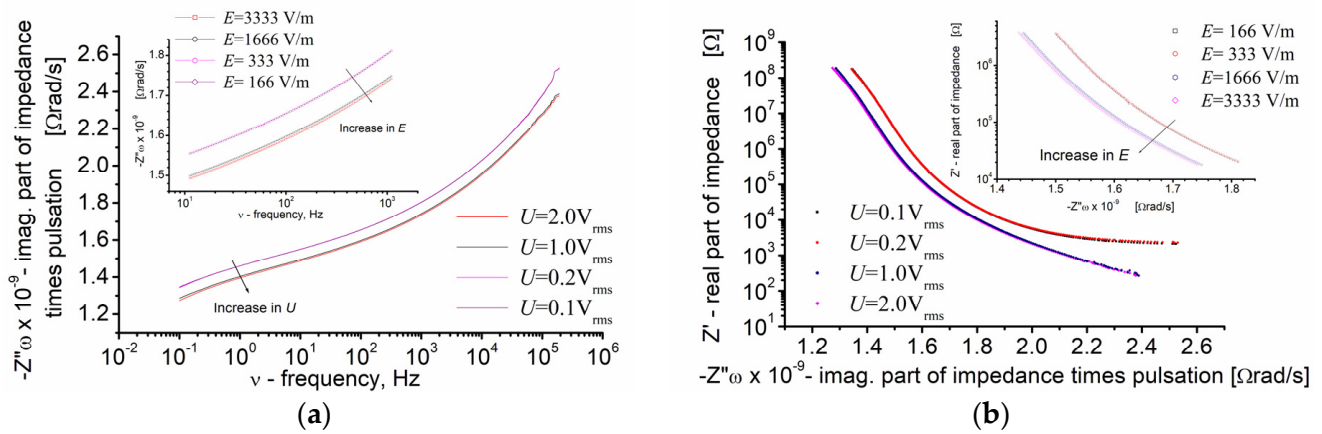


Figure 6. Alternative representation of the impedance data collected at room temperature and signal amplitude $U = 0.1$ V_{rms}, $U = 0.2$ V_{rms}, $U = 1.0$ V_{rms}, and $U = 2.0$ V_{rms}. Panel (a) shows the dependence of reactance times pulsation ($-Z''\omega$) on frequency (ν); panel (b) shows the dependence of resistance (Z') on reactance times pulsation ($-Z''\omega$).

As shown in the next paragraph of this paper (Section 3.4), an increase in the electric field strength causes an increase in the circumferential magnetic field strength. In turn, the dependence of electric (and dielectric) parameters on magnetic field strength (and vice versa) is a key feature of multiferroic materials, especially lead-free piezoelectric ceramics such as 67BFTO.

3.4. Piezoelectric Ceramics Characterisation with the Resonant Method

It is commonly known (e.g., see [21,24]) that values of the piezoelectric properties of a material can be derived from the resonance behaviour of suitably shaped specimens subjected to a sinusoidally varying electric field. Therefore, the impedance measurements were performed for 67BFBT ceramics within the following frequency range: $\nu = 100$ kHz–10 MHz, which corresponded to the frequency ranges of resonant spectra of the radial mode and the thickness extension mode for thin disk sample [21,22].

To reveal the influence of a weak circumferential magnetic field on the resonance behaviour of multiferroic lead-free material, the measuring voltage $U = 0.1$ V_{rms} and $U = 2.0$ V_{rms} was taken. Taking into account all the impedances in the circuit, the measuring sinusoidal signal leads to electric currents (i_{ac}) through the sample of 67BFBT ceramics in the range following range: $i_{ac} = 1.2$ mA (at $U_{ac} = 0.1$ V_{rms}, $\nu = 10$ MHz)–11.7 mA (at $U = 2.0$ V_{rms}, $\nu = 10$ MHz). The ac field amplitude (rms) generated by these currents in a radial point r on the sample can be calculated as:

$$H_{ac} = i_{ac}r/2\pi a^2 \quad (1)$$

where r is the radial point considered on the sample cross-section, and a is its total radius. The rms range of ac-measuring fields leads to circumferential magnetic fields (i.e., on the sample edges—the lateral surface of the disk-shaped sample—is where it is highest) between 0.05 and 0.49 A/m (at $U = 0.1$ V_{rms} and $U = 2.0$ V_{rms}, $\nu = 10$ MHz, respectively) [28]. The results of the calculated radial magnetic field intensity H_{ac} are given in Figure 7. It is worth noting that the resonance behaviour of the 67BFBT ceramic sample is also reflected in Figure 7. Positions of the resonances are closely related to the piezoelectric properties of the material.

Figure 8 shows the frequency response of a 0.67Bi_{1.02}FeO₃–0.33BaTiO₃ ceramic thin disk that was 7.52 mm in diameter and 0.6 mm thick. Electrodes were deposited onto both faces of the disk, and then the disk was poled in the direction perpendicular to the faces of the disk. The measurements were taken within a temperature range from -20 °C to 50 °C. It should be noted that the frequency peaks visible at about $3\text{--}4 \times 10^5$ Hz (Figure 8)

are radial resonances, whereas the frequency peaks visible at about $3\text{--}5 \times 10^6$ Hz (also in Figure 8) are related to the thickness mode resonance.

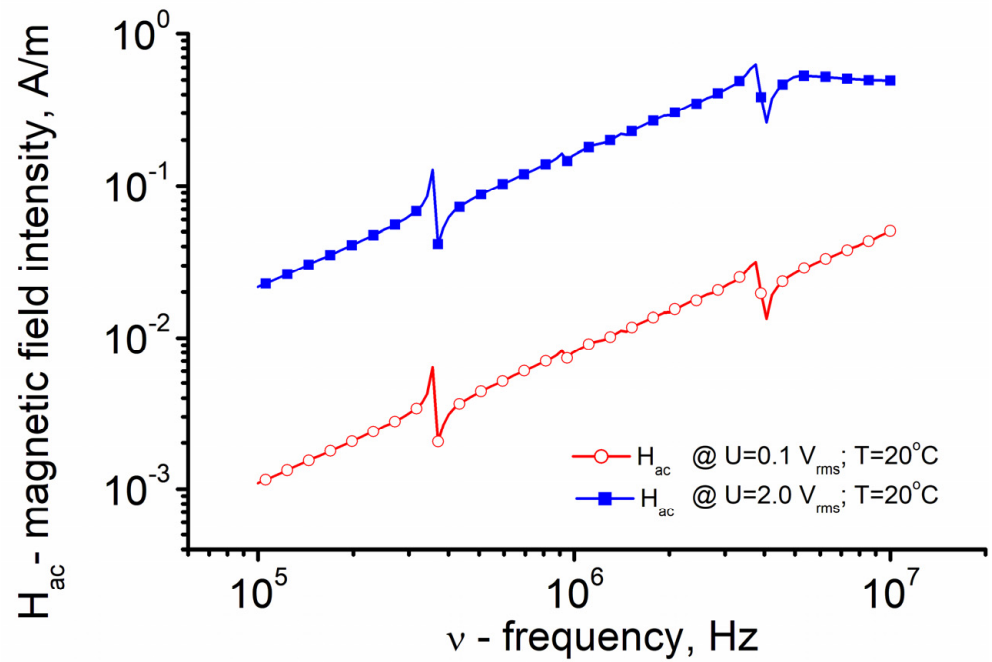


Figure 7. Dependence of circumferential magnetic field intensity H_{ac} on frequency of the measuring field for 67BFBT ceramics studied at temperature 20°C ; measuring voltage 0.1 V_{rms} and 2.0 V_{rms} .

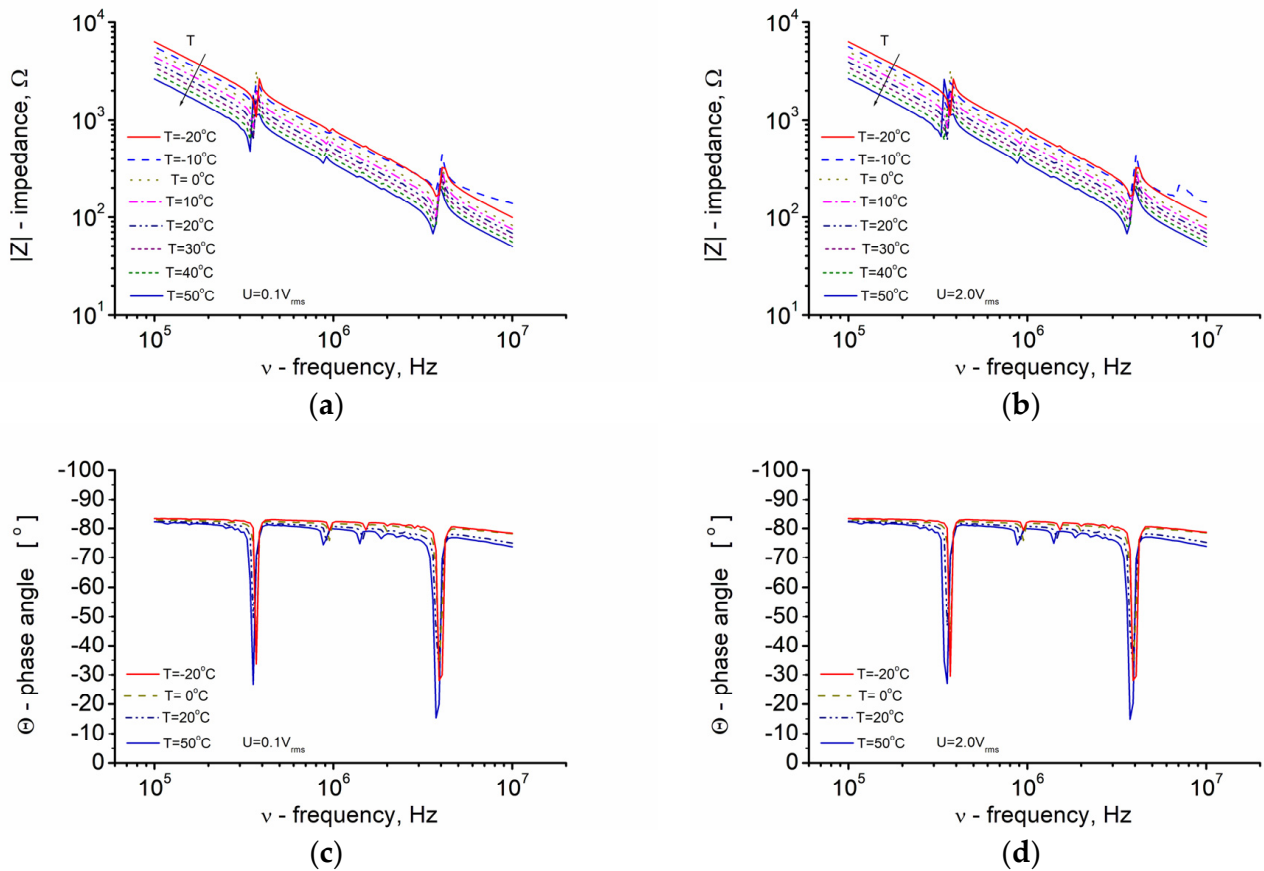


Figure 8. Dependence of impedance modulus $|Z|$ as well as the phase angle (Θ) on frequency for 67BFBT ceramics; measuring voltages: $U = 0.1\text{ V}_{rms}$ (a,c) and $U = 2.0\text{ V}_{rms}$ (b,d).

Spectroscopic plots of modulus of complex impedance $|Z|$ exhibit the characteristic frequencies, namely ν_{\min} when the impedance $|Z|$ is at its minimum ($|Z|_{\min}$) and ν_{\max} when the impedance is at its maximum ($|Z|_{\max}$) (Figure 8a,b). At the same time, the phase angle (Θ) given in Figure 8c,d tends to have a value of “zero.”

3.5. Modelling of Impedance—Frequency Characteristics of the Piezoelectric Equivalent Circuit

Let us first explain the notation used. In the adopted notation, square or box brackets [] denote that elements are in series-connected, whereas round brackets () denote the parallel connection of electric elements. According to the adopted notation, (RC) is a parallel circuit, while [RC] is a series connection of the elements R and C.

To accurately approximate the behaviour of the piezoelectric specimen close to its fundamental resonance, it can be represented by the electric equivalent circuit ([L₁R₁C₁]C₀) consisting of a “mechanical arm” (L, C, and R connected in series) and C₀ (which corresponds to the electrical capacitance of the specimen) connected in parallel [24]. In this connection, it is worth remembering that the impedance of the parallel circuit can be represented by the equivalent series circuit consisting of equivalent in series-connected resistance and reactance values.

In the case of our simulation and fitting, the modified electric equivalent circuit [RsCPE₁([L₁R₁C₁]C₀)] including resistance Rs and constant phase element CPE₁ connected in series with the “piezoelectric” equivalent circuit was used. Figure 9 shows the electric equivalent circuit used for the simulation and fitting of the impedance response of the ceramic specimen vibrating close to its fundamental resonance.

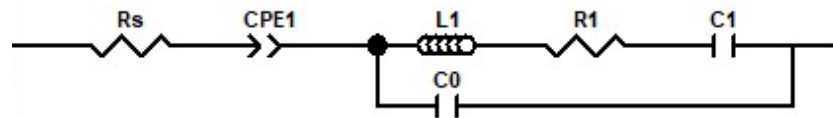


Figure 9. Electric equivalent circuit of a piezoelectric specimen used for simulation and fitting of the dielectric response.

It should be noted that the resonance frequency ν_r and antiresonance frequency ν_a correspond to the “zero” value of reactance for the electric equivalent circuit (Figure 9). The reactance of the “mechanical arm” is zero at the series resonant frequency ν_s when:

$$\omega = 1/\sqrt{L_1C_1} \tag{2}$$

where ω is the angular frequency ($\omega = 2\pi\nu$).

The reactance of the parallel circuit is zero at the parallel resonant frequency (ν_p). The parallel resonance ν_p occurs when the currents flowing in the two arms are in antiphase, which is when:

$$\omega = (C_0 + C_1/L_1C_0C_1)^{1/2} \tag{3}$$

In this connection, it must be pointed out that the relation between the above-mentioned characteristic frequencies of the equivalent circuit is as follows: $\nu_{\min} < \nu_s < \nu_r$. However, the difference between them is very small ($\nu_{\min} \sim \nu_s \sim \nu_r$) [24]. Similar relation exists between antiresonance, parallel resonance, and $|Z|_{\max}$ frequencies: $\nu_a < \nu_p < \nu_{\max}$ ($\nu_a \sim \nu_p \sim \nu_{\max}$). What is important is that values of ν_{\min} and ν_{\max} can be readily measured using an impedance analyser (Alpha-AN High Performance Frequency Analyzer).

An example of the fitting results obtained for impedance characteristics measured under the influence of a weak circumferential magnetic field is shown in Figure 10. Figure 10a shows the spectra measured at $H_{ac} = 0.05$ A/m, ($E = 166$ V/m; $U = 0.1$ V_{rms}) at 20 °C, whereas Figure 10b shows the results obtained for spectrum measured under the following conditions: $H_{ac} = 0.49$ A/m, ($E = 1333$ V/m; $U = 2$ V_{rms}) at 20 °C.

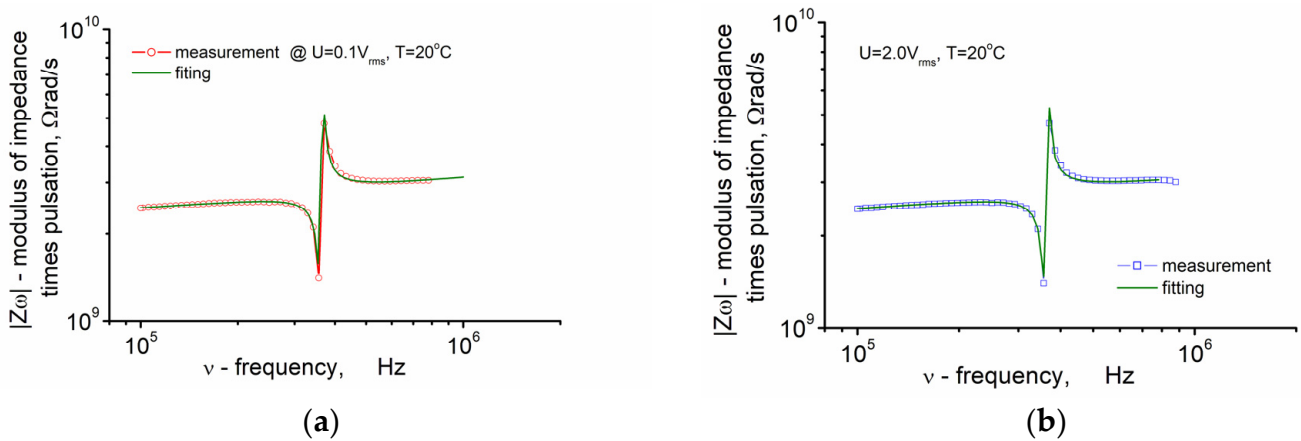


Figure 10. Impedance/frequency characteristics of 67BFBT ceramics (empty symbols) studied under the following conditions: temperature—20 °C and measuring voltages—(a) $U = 0.1 V_{rms}$; (b) $U = 2.0 V_{rms}$. Fit result (green solid line) was limited to the frequency range of the first resonance.

Experimental data were fitted to the electric equivalent circuit using the ZView programme (Scribner Associates, Inc. Southern Pines, NC, USA). Complex non-linear least squares method (CNLS) was employed for the analysis of the impedance/frequency data of the electroceramics [17]. The fitting procedure was limited to the frequency range of the radial resonances $\Delta\nu = (2-6) \times 10^5$ Hz. The quality of the fitting procedure was estimated according to the following parameters: “chi-squared” (χ^2) and weighted sum of squares (WSS) [29]. In the case of the fitting results shown in Figure 10, the parameters were as follows: “chi-squared” was $\chi^2 = 2.78 \times 10^{-3}$ and $\chi^2 = 3.23 \times 10^{-3}$ for $U = 0.1 V_{rms}$ and $U = 2.0 V_{rms}$, respectively. The weighted sum of squares was $WSS = 0.314$ and $WSS = 0.177$ for $U = 0.1 V_{rms}$ and $U = 2.0 V_{rms}$, respectively.

Figure 11 shows the dependence of the fitting quality parameters, namely “chi-squared” (χ^2) and weighted sum of squares (WSS), on the temperatures at which the impedance/frequency characteristics of 67BFBT ceramics were recorded.

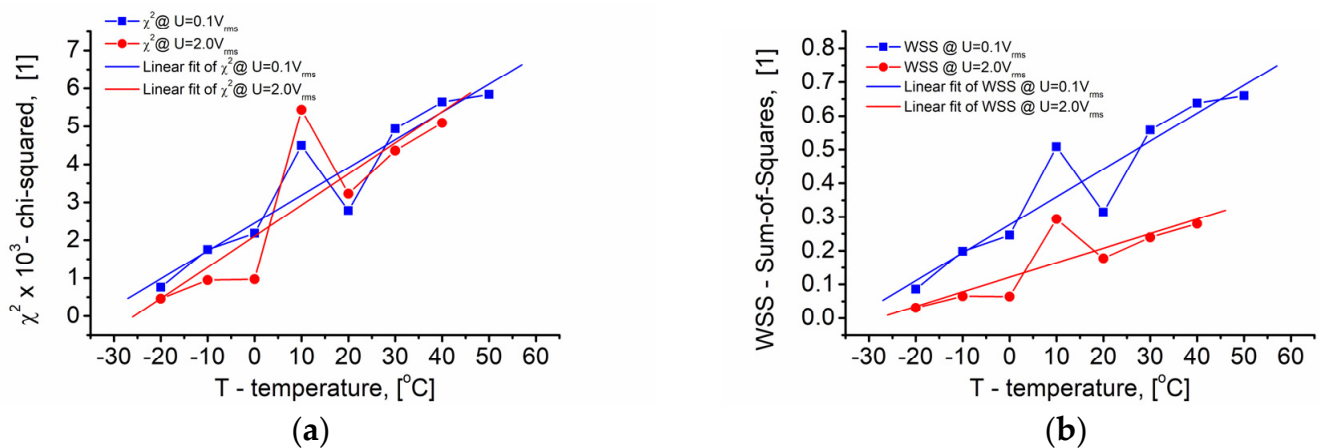


Figure 11. Fitting quality parameters χ^2 (“chi-squared”) (a) and weighted sum of squares (WSS) (b) obtained as a result of fitting the experimental resonance spectra and response of the electric equivalent circuit given in Figure 8; amplitude of the measuring signal: $U = 0.1 V_{rms}$ (blue squares) and $U = 2.0 V_{rms}$ (red circles).

One can see in Figure 11 that both χ^2 (“chi-squared”) fitting quality parameter (Figure 11a) and WSS parameter (Figure 11b) change within one order of magnitude. Upon visual inspection, it can be seen that the linear approximations used for the χ^2 and WSS data approximation show that a higher electric field strength of the measuring signal (and, at the same time, a higher value of the circumferential magnetic field strength) improves the

quality of the fitting procedure. This also means that the data is less susceptible to external interference. A comparison of the statistical characteristics of the obtained “chi-squared” fitting quality parameters showed that the standard deviation was $SD = 0.00192$ and $SD = 0.00212$ for the low value of the electric field strength $E = 166 \text{ V/m}$ ($U = 0.1 \text{ V}_{\text{rms}}$) and high electric field strength $E = 3333 \text{ V/m}$ ($U = 2.0 \text{ V}_{\text{rms}}$), respectively. The WSS parameter showed that the standard deviation was $SD = 0.21743$ and $SD = 0.11077$ for $E = 166 \text{ V/m}$ and $E = 3333 \text{ V/m}$, respectively. An increase in the intensity of the measuring field led to a substantial improvement in the quality of further data simulation.

3.6. Calculation of Piezoelectric Parameters of 67BFTO Ceramics

The entry parameters used to calculate the piezoelectric parameters of BFBT ceramics, namely resonant frequencies ν_r , anti-resonant frequencies ν_a , impedance, and free capacitance C^T on 1 kHz were measured by using the aforementioned impedance analyser. As shown in Figure 8, a thorough analysis of the experimental data (i.e., measured impedance spectra) was also performed. The procedure for the calculation of single coefficients is described in detail in classical textbooks (e.g., see [21,24,25]), scientific papers (e.g., see [22,23]), or standards (e.g., see [30]). The results of the calculations are given in Figures 12 and 13.

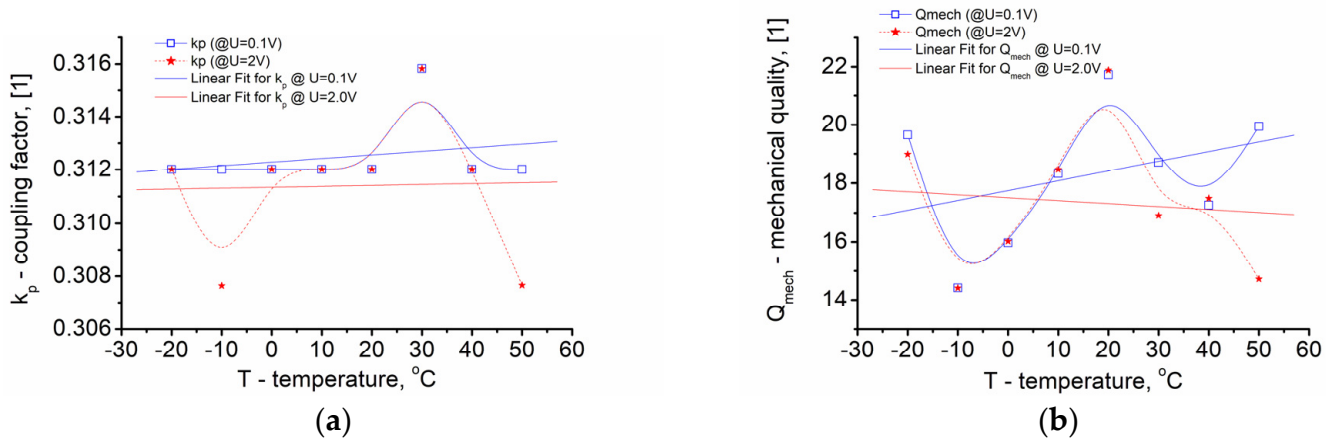


Figure 12. (a) Dependence of the planar coupling factor k_p for the vibration along the radial direction in a circle-shaped disk of 67BFTO ceramics on temperature; (b) dependence of mechanical quality factor on temperature; calculations were performed based on experimental impedance data measured under the influence of radial magnetic field $H_{\text{ac}} = 0.05 \text{ A/m}$ ($U = 0.1 \text{ V}_{\text{rms}}$) and $H_{\text{ac}} = 0.49 \text{ A/m}$ ($U = 2.0 \text{ V}_{\text{rms}}$).

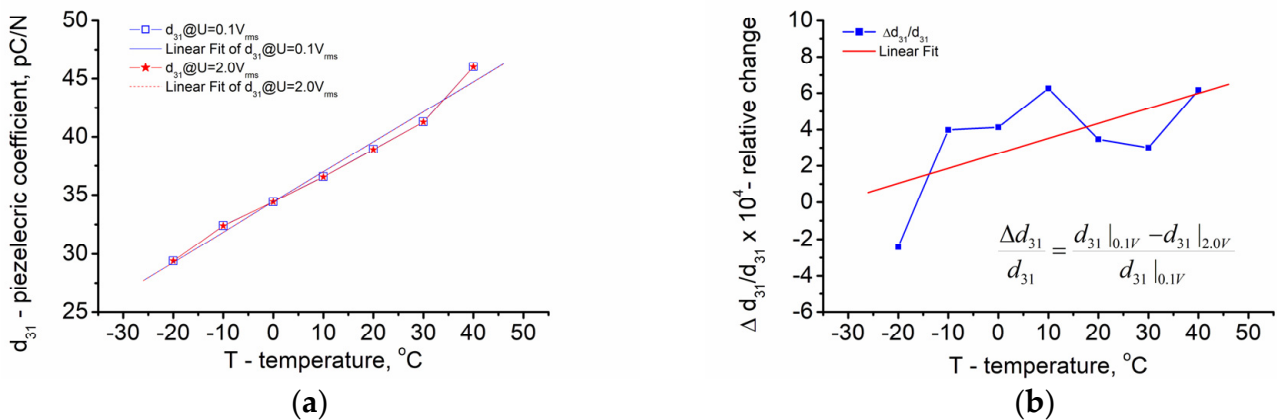


Figure 13. (a) Dependence of piezoelectric charge coefficient d_{31} on temperature; (b) dependence of the relative change in d_{31} on temperature; the inset in Figure 13b shows the formula for calculation of the relative change in piezoelectric modulus d_{31} .

Figure 12a shows dependence of the planar coupling factor k_p for the vibration along the radial direction in a circle-shaped disk of 67BFTO ceramics on temperature. The results of the calculations of the mechanical quality factor Q_m are given in Figure 12b.

One can see from Figure 12a that the coupling factor for both magnetic circumferential fields has a value of about $k_p = 31\%$. Local extremes visible on the plots at $-10\text{ }^\circ\text{C}$ and $-30\text{ }^\circ\text{C}$ differ from the average value by about 1%. Therefore, it is reasonable to use linear regression fit for data analysis. The difference between the linear fit plots is rather small but can easily be discerned. Moreover, the linear regression fit for a higher magnetic field ($H_{ac} = 0.49\text{ A/m}$) shows better stability with temperature within the measured temperature range.

One can see from Figure 12b that the mechanical quality factor Q_m exhibits non-monotonic behaviour with increasing temperature for both of the used values of magnetic circumferential field intensity. The local extremes are clearly visible. Within the temperature range $\Delta T = (-30\text{--}+20)\text{ }^\circ\text{C}$, the courses of the Q_m curves are almost identical. The influence of magnetic field intensity on Q_m becomes noticeable at about room temperature ($20\text{ }^\circ\text{C}$). The difference between the linear fit plots can easily be discerned in Figure 12b. One can see that the linear regression fit for higher magnetic field intensity ($H_{ac} = 0.49\text{ A/m}$; red line in Figure 12b) shows a negative slope with temperature within the measured temperature range, whereas the linear regression fit for Q_m behaviour at a smaller magnetic circumferential field exhibits a positive one ($H_{ac} = 0.05\text{ A/m}$; blue squares; blue line in Figure 12b).

Figure 13a shows the dependence of piezoelectric charge coefficient d_{31} on the radial vibration mode of a thin disk (excited through the piezoelectric effect across the thickness of the disk). One can see from Figure 13a that the piezoelectric charge coefficient d_{31} increases from $d_{31} \approx 30\text{ pC/N}$ to $d_{31} \approx 45\text{ pC/N}$ at the following temperature range: $\Delta T = (-20\text{--}+40)\text{ }^\circ\text{C}$. The courses of the d_{31} curves calculated for data measured at radial magnetic field $H_{ac} = 0.05\text{ A/m}$ (blue squares; solid line; $U = 0.1\text{ V}_{\text{rms}}$) and $H_{ac} = 0.49\text{ A/m}$ ($U = 2.0\text{ V}_{\text{rms}}$; red stars; dashed line) are almost identical. Additionally, the linear regression fits for the experimental data cannot easily be discerned in Figure 13a. Linear regression ($Y = A + B \times X$) for the data measured at $H_{ac} = 0.05\text{ A/m}$ ($U = 0.1\text{ V}_{\text{rms}}$) exhibited the following parameters: $A = 3.44221 \times 10^{-11}$; $B = 2.57759 \times 10^{-13}$; and $R = 0.99097$. Alternatively, linear regression for the data measured at $H_{ac} = 0.49\text{ A/m}$ ($U = 2.0\text{ V}_{\text{rms}}$) showed the following values: $A = 3.4412 \times 10^{-11}$; $B = 2.57385 \times 10^{-13}$; and $R = 0.99102$. Therefore, to show the possible influence of the circumferential magnetic field on the piezoelectric charge coefficient d_{31} , the relative change in the d_{31} piezoelectric modulus was introduced, and the results are plotted in Figure 13b.

One can see from Figure 13b that the plot of the relative change in the d_{31} piezoelectric modulus exhibits non-monotonic behaviour with increasing temperature. Within the measuring temperature range, one local maximum and one local minimum are clearly visible in the plot. One can see from Figure 13b that, according to linear fit, the relative change in the piezoelectric modulus has a positive sign within the whole measuring temperature range. This means that an increase in the intensity of the circumferential magnetic field generated by electric currents through the sample causes the suppression of the piezoelectric response of the multiferroic ceramic sample under study.

The above-mentioned influence of circumferential magnetic field on the piezoelectric charge coefficient d_{31} can explain the dielectric properties of multiferroic ceramics. The point is that the magnetic field influences both real (ϵ') and imaginary (ϵ'') parts of the complex dielectric permittivity. It was found that the higher the radial magnetic field, the lower the dielectric permittivity (at a given temperature). As an example, the dependence of the real and imaginary parts of the complex dielectric permittivity on frequency (below resonance) for radial magnetic field $H_{ac} = 0.023\text{ A/m}$ ($U = 2.0\text{ V}_{\text{rms}}$; $\nu = 10^5\text{ Hz}$) and $H_{ac} = 0.012\text{ A/m}$ ($U = 1.0\text{ V}_{\text{rms}}$; $\nu = 10^5\text{ Hz}$) at $20\text{ }^\circ\text{C}$ are shown in Figure 14.

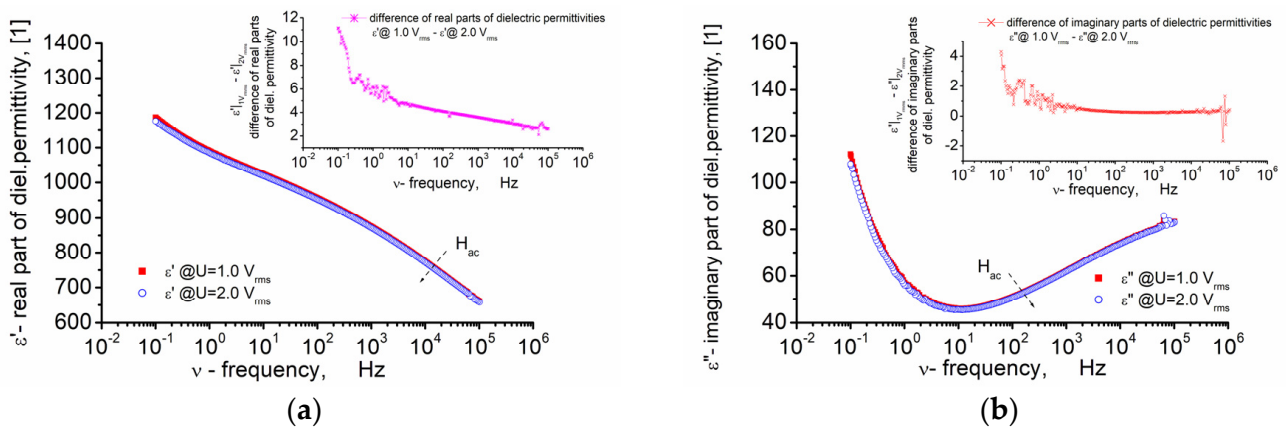


Figure 14. (a) Dependence of the real part of complex dielectric permittivity (ϵ') on frequency. (b) Dependence of the imaginary part of complex dielectric permittivity (ϵ'') on frequency; measurements were performed under electric field strength $E = 1666 \text{ V/m}$ ($U = 1.0 \text{ V}_{\text{rms}}$) and $E = 3333 \text{ V/m}$ ($U = 2.0 \text{ V}_{\text{rms}}$) corresponding to radial magnetic field $H_{\text{ac}} = 0.012 \text{ A/m}$ and $H_{\text{ac}} = 0.023 \text{ A/m}$ (at $\nu = 100 \text{ kHz}$, respectively).

The influence of the circumferential magnetic field on the dielectric properties of 67BFBT ceramics was also revealed when the difference of the modulus of dielectric permittivity was plotted against the difference of circumferential magnetic field caused by the electric currents (i_{ac}) flowing through the ceramic sample (Figure 15).

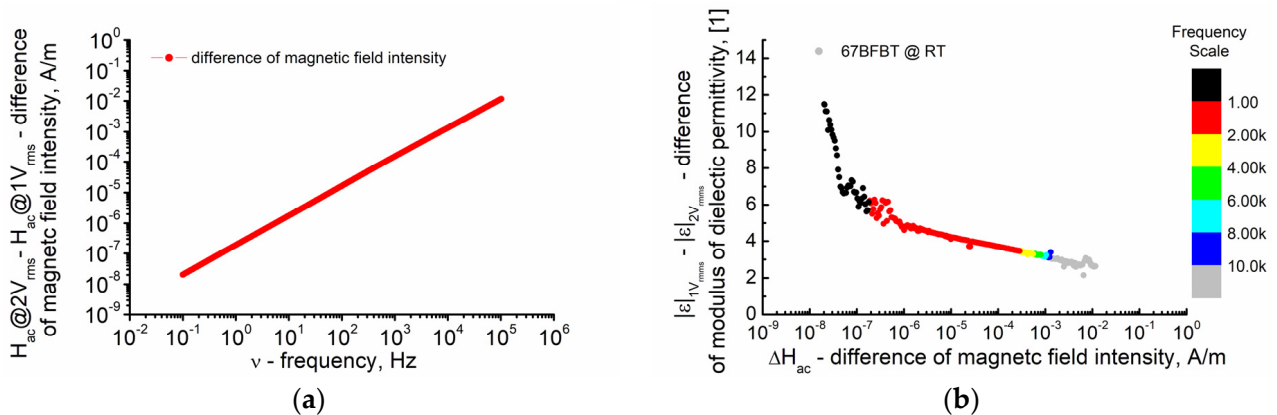


Figure 15. Dependence of the difference of circumferential magnetic field strength on frequency of the measuring electric field (a). Dependence of modulus of complex dielectric permittivity on circumferential magnetic field strength (b).

One can see in Figure 15a that the difference in circumferential magnetic fields generated by the electric currents flowing through the sample linearly depends (in log-log scale) on the frequency of the measuring sinusoidal signal. On the other hand, Figure 15b shows that the change in magnetic field strength causes a change in the modulus of the complex dielectric permittivity. Thus, the possibility of adjusting the dielectric permittivity (and therefore capacitance value) via changes in magnetic field intensity was obtained for 67BFBT ceramics. It is worth noting that changing the capacitance of a capacitor in an electric circuit has a predictable effect on the complex impedance and phase angle in the circuit. These parameter changes can be exploited to yield tuneable impedance-matching networks, tuneable filters, phase shifters, and other functional multiferroic devices.

4. Conclusions

Multiferroic $0.67\text{Bi}_{1.02}\text{FeO}_3\text{-}0.33\text{BaTiO}_3$ (67BFBT) ceramics were fabricated via a solid-state reaction process and subsequent sintering in a box furnace. They were studied in terms

of structure and dielectric and piezoelectric properties. The fabricated 67BFBT ceramics adopted the tetragonal crystal structure described by the $P4mm$ space group. The average crystallite size was $\langle D \rangle \approx 80$ nm, and the average strain was $\langle \epsilon \rangle = 0.01\%$. Broad-band dielectric spectroscopy (BBDS) was used for measurements whereas the electric equivalent circuit method was used to fit the impedance-frequency spectra within the frequency range of the radial resonances. Complex non-linear least squares method (CNLS) was employed to fit the parameters of the electric equivalent circuit $[RsCPE_1([L_1R_1C_1]C_0)]$. The high quality of the fitting procedure was confirmed by “chi-squared” (χ^2) and weighted sum of squares (WSS) parameters. Measurements and subsequent fitting and simulations have shown that the planar coupling factor was $k_p \approx 31\%$, and the mechanical quality was within the range $Q_{mech} \approx 21$. The piezoelectric charge coefficient for the radial vibration mode of a thin disk (excited through the piezoelectric effect across the thickness of the disk) exhibited a value of $d_{31} \approx 35$ pC/N at room temperature.

The idea of using a measuring signal of different amplitudes (BBDS measurements) was to check the possible influence of the radial magnetic field caused by electric currents flowing through the sample on the piezoelectric properties of multiferroic 67BFBT ceramics. The rms range of ac measuring fields led to circumferential magnetic fields ranging from 0.05 A/m to 0.49 A/m. It was established that an increase in the intensity of the circumferential magnetic field caused the suppression of the real (ϵ') and imaginary (ϵ'') parts of the complex dielectric permittivity and piezoelectric charge coefficient (d_{31}) of the multiferroic 67BFBT ceramics within the range of the measurement parameters used (i.e., frequency, temperature, and radial magnetic field). The “magnetic” tunability of the dielectric properties of 67BFBT ceramics was also found, as evidenced by the possibility of adjusting the modulus of the complex dielectric permittivity (and thus the capacitance value) by means of the magnetic field strength. Changes in dielectric parameters can be used to obtain “magnetically” tuneable functional multiferroic devices, e.g., tuneable filters or phase shifters. Taking into account that the strength of the magnetic field used in the current study was about 6 mOe (in CGS units), one can conclude that multiferroic 67BFBT ceramics are very sensitive to magnetic field influence, which makes 67BFBT sensing material with vast potential.

Author Contributions: Conceptualization, D.C. and A.L.-C.; methodology, A.L.-C. and D.C.; software, D.C. and W.B.; validation, A.L.-C. and J.-H.J.; formal analysis, A.L.-C., D.C. and J.-H.J.; investigation, A.L.-C., D.C., T.T.Z., J.-H.J., B.G.-G. and W.B.; resources, A.L.-C., D.C., B.G.-G. and W.B.; data curation, A.L.-C., T.T.Z., J.-H.J., B.G.-G. and W.B.; writing—original draft preparation, A.L.-C. and D.C.; writing—review and editing, D.C., A.L.-C. and J.-H.J.; visualization, D.C.; supervision, D.C.; project administration, A.L.-C.; funding acquisition, A.L.-C. All authors have read and agreed to the published version of the manuscript.

Funding: This research was funded by POLISH NATIONAL SCIENCE CENTRE (NCN), grant number N N507 446934.

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Not applicable.

Data Availability Statement: The data presented in this paper are available on request from the corresponding author.

Acknowledgments: The present research was supported by Gdańsk University of Technology as part of the EINSTEINIUM SHORT-TERM INCOMING VISITS programme.

Conflicts of Interest: The authors declare no conflict of interest.

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